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(21) International Application Number: PCT/US99/19582 (22) International Filing Date: 26 August 1999 (26.08.99) (30) Priority Data: <table border="0"> <tr> <td>09/141,386</td> <td>27 August 1998 (27.08.98)</td> <td>US</td> </tr> <tr> <td>09/141,393</td> <td>27 August 1998 (27.08.98)</td> <td>US</td> </tr> <tr> <td>09/141,405</td> <td>27 August 1998 (27.08.98)</td> <td>US</td> </tr> <tr> <td>09/140,525</td> <td>27 August 1998 (27.08.98)</td> <td>US</td> </tr> </table> (71) Applicant: SUPERIOR MICROPOWDERS LLC [US/US]; 3740 Hawkins N.E., Albuquerque, NM 87109 (US). (72) Inventors: KODAS, Toivo, T.; 11102 San Rafael Drive NE, Albuquerque, NM 87122 (US). HAMPDEN-SMITH, Mark, J.; 2901 Maximillian NW, Albuquerque, NM 87131 (US). POWELL, Quint, H.; 14336 Grand Avenue NE, Albuquerque, NM 87123 (US). CARUSO, James; 7721 Cedar Canyon Court, NE, Albuquerque, NM 87122 (US). SKAMSER, Daniel, J.; 10327 Hackamore Place, SW, Albuquerque, NM 87121 (US). KUNZE, Klaus; 800 Oro Real Drive, NE, Albuquerque, NM 87123 (US).	09/141,386	27 August 1998 (27.08.98)	US	09/141,393	27 August 1998 (27.08.98)	US	09/141,405	27 August 1998 (27.08.98)	US	09/140,525	27 August 1998 (27.08.98)	US	(74) Agent: BREYFOGLE, Ross, E.; Holme Roberts & Owen LLP, Suite 4100, 1700 Lincoln Street, Denver, CO 80203-4541 (US). (81) Designated States: AE, AL, AM, AT, AU, AZ, BA, BB, BG, BR, BY, CA, CH, CN, CR, CU, CZ, DE, DK, DM, EE, ES, FI, GB, GD, GE, GH, GM, HR, HU, ID, IL, IN, IS, JP, KE, KG, KP, KR, KZ, LC, LK, LR, LS, LT, LU, LV, MD, MG, MK, MN, MW, MX, NO, NZ, PL, PT, RO, RU, SD, SE, SG, SI, SK, SL, TJ, TM, TR, TT, UA, UG, UZ, VN, YU, ZA, ZW, ARIPO patent (GH, GM, KE, LS, MW, SD, SL, SZ, UG, ZW), Eurasian patent (AM, AZ, BY, KG, KZ, MD, RU, TJ, TM), European patent (AT, BE, CH, CY, DE, DK, ES, FI, FR, GB, GR, IE, IT, LU, MC, NL, PT, SE), OAPI patent (BF, BJ, CF, CG, CI, CM, GA, GN, GW, ML, MR, NE, SN, TD, TG). Published <i>With international search report.</i>
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(54) Title: PHOSPHOR POWDERS, METHODS FOR MAKING PHOSPHOR POWDERS AND DEVICES INCORPORATING SAME													
(57) Abstract Phosphor powders and a method for making phosphor powders. The phosphor powders have a small particle size, a narrow particle size distribution and are substantially spherical. The method of the invention advantageously permits the economic production of such powders. The invention also relates to improved devices, such as display devices, incorporating the phosphor powders.													

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**PHOSPHOR POWDERS,
METHODS FOR MAKING PHOSPHOR POWDERS
AND DEVICES INCORPORATING SAME**

BACKGROUND OF THE INVENTION

1. Field of the Invention

The present invention relates to phosphor powders, methods for producing phosphor powders and devices, such as display devices, incorporating the powders. In particular, the present invention is directed to phosphor powders having well-controlled chemical and physical properties. The present invention also relates to a method for producing such powders by spray-conversion.

2. Description of Related Art

Phosphors are compounds that are capable of emitting useful quantities of radiation in the visible and/or ultraviolet spectrums upon excitation of the phosphor by an external energy source. Due to this property, phosphor compounds have long been utilized in cathode ray tube (CRT) screens for televisions, in lighting elements and in other devices. Typically, inorganic phosphor compounds include a host material doped with a small amount of an activator ion.

There are a number of requirements for phosphor powders, which can vary dependent upon the specific application of the powder. Generally, phosphor powders should have one or more of the following properties: high purity; high crystallinity; small particle size; narrow particle size distribution; spherical morphology; controlled surface chemistry; homogenous distribution of the activator ion; good dispersibility; and controlled porosity. The proper combination of the foregoing properties will result in a phosphor powder with high luminescent intensity and long lifetime that can be used in many applications. It is also advantageous for many applications to provide phosphor powders that are surface passivated or coated, such as with a thin, uniform dielectric or semiconducting coating.

Phosphor powders can be categorized by the excitation mechanism, which causes luminescence in the phosphor. In this regard, phosphor powders can be categorized as cathodoluminescent, photoluminescent, electroluminescent or x-ray phosphors.

5 Cathodoluminescent phosphor powders luminesce when the phosphor powder is excited by an electron source, usually a high-energy electron source. For example, cathodoluminescent phosphors are utilized in CRT devices. More recently, cathodoluminescent phosphors have been utilized in advanced display devices that provide illuminated text, graphics or video output. In particular,
10 there has been significant growth in the field of flat panel displays such as field emission displays (FEDs). FEDs are similar in principal to CRTs, wherein electrons emitted from a tip stimulate the phosphors, which then emit light of a preselected color.

Photoluminescent phosphor powders are phosphor powders, which emit
15 light when excited by light (photons) of a different wavelength. For example, photoluminescent phosphor powders are used in fluorescent lighting elements including common industrial lights. Photoluminescent phosphors are also used as the back lights for display screens.

More recently, photoluminescent phosphor powders have been used in
20 advance display devices such as plasma displays. Plasma display panels utilize a gas trapped between transparent layers wherein the gas emits ultraviolet light when excited by an electric field. The ultraviolet light stimulates the photoluminescent phosphors on the screen to emit visible light. Plasma displays are particularly useful for larger displays, such as greater than about 20 diagonal
25 inches. In addition, photoluminescent phosphor powders can also be used as a means of identifying an article of manufacture, such as a confidential document, currency, postage and the like.

Electroluminescent phosphor powders emit light when excited by an electric field. Electroluminescent phosphor powders can also be used in flat
30 panel display devices such as thick film and thin film electroluminescent displays.

Thin film and thick film electroluminescent displays (TFEL) utilize a film of electroluminescent material trapped between glass plates and electrodes that emit light in an electric field. Electroluminescent phosphor powders are also used in electroluminescent lamps (EL), which include an electroluminescent phosphor powder deposited on a polymer substrate which emits light when an electric field is applied.

X-ray phosphor powders are used in image intensifiers, particularly for medical devices.

Numerous methods have been proposed for producing phosphor particles. One such method is referred to as the solid-state method. In this process, the phosphor precursor materials are mixed in the solid state and are heated so that the precursors react and form a powder of the phosphor material. For example, U.S. Patent No. 4,925,703 by Kasenga et al. discloses a method for the production of a manganese activated zinc silicate phosphor ($\text{ZnSiO}_4:\text{Mn}$). The method includes a step of dry blending a mixture of starting components such as zinc oxide, silicic acid and manganese carbonate and firing the blended mixture at about 1250°C. The resulting phosphor is broken up or crushed into smaller particles. Solid-state routes, and many other production methods, utilize a grinding step to reduce the particle size of the powders. The mechanical grinding damages the surface of the phosphor, forming dead layers, which inhibit the brightness of the phosphor, powders.

Phosphor powders have also been made by liquid precipitation. In these methods, a solution, which includes phosphor particle precursors, is chemically treated to precipitate phosphor particles or phosphor particle precursors. These particles are typically calcined at an elevated temperature to produce the phosphor compound. The particles must often be further crushed, as is the case with solid-state methods.

In yet another method, phosphor particle precursors or phosphor particles are dispersed in a solution, which is then spray dried to evaporate the liquid. The phosphor particles are thereafter sintered in the solid state at an elevated

temperature to crystallize the powder and form a phosphor. For example, U.S. Patent No. 4,948,527 by Ritsko et al. discloses a process for producing $Y_2O_3:Eu$ phosphors by dispersing yttrium oxide in a europium citrate solution to form a slurry which is then spray dried. The spray-dried powder was then converted to an oxide by firing at about 1000°C for two hours and then at 1600°C for about four hours. The fired powder was then lightly crushed and cleaned to recover useful phosphor particles.

U.S. Patent No. 5,644,193 by Matsuda et al. discloses phosphor powders having an average particle size of up to 20 μm . The phosphors can include rare earth oxides, rare earth oxysulfides and tungstates. The particles are produced by fusing phosphor particles in a thermal plasma and rapidly cooling the particles.

Despite the foregoing, there remains a need for phosphor powders having well-controlled properties for various applications. Desirable properties typically include a high luminescent intensity, particles having a substantially spherical morphology, narrow particle size distribution, a high degree of crystallinity and good homogeneity. The powder should have good dispersibility and the ability to be fabricated into thin layers having uniform thickness. Phosphor powders having these properties will be particularly useful in display devices and other advanced applications.

SUMMARY OF THE INVENTION

The present invention provides improved phosphor powder batches including phosphors having a small particle size, spherical morphology and good crystallinity. The present invention also provides methods for forming phosphor powder batches and devices, such as advanced display devices and lighting elements, incorporating the powder batches.

BRIEF DESCRIPTION OF THE DRAWINGS

Fig. 1 is a process block diagram showing one embodiment of the process of the present invention.

5 Fig. 2 is a side view in cross section of one embodiment of aerosol generator of the present invention.

Fig. 3 is a top view of a transducer mounting plate showing a 49 transducer array for use in an aerosol generator of the present invention.

Fig. 4 is a top view of a transducer mounting plate for a 400 transducer array for use in an ultrasonic generator of the present invention.

10 Fig. 5 is a side view of the transducer mounting plate shown in Fig. 4.

Fig. 6 is a partial side view showing the profile of a single transducer mounting receptacle of the transducer mounting plate shown in Fig. 4.

Fig. 7 is a partial side view in cross-section showing an alternative embodiment for mounting an ultrasonic transducer.

15 Fig. 8 is a top view of a bottom retaining plate for retaining a separator for use in an aerosol generator of the present invention.

Fig. 9 is a top view of a liquid feed box having a bottom retaining plate to assist in retaining a separator for use in an aerosol generator of the present invention.

20 Fig. 10 is a side view of the liquid feed box shown in Fig. 9.

Fig. 11 is a side view of a gas tube for delivering gas within an aerosol generator of the present invention.

25 Fig. 12 shows a partial top view of gas tubes positioned in a liquid feed box for distributing gas relative to ultrasonic transducer positions for use in an aerosol generator of the present invention.

Fig. 13 shows one embodiment for a gas distribution configuration for the aerosol generator of the present invention.

Fig. 14 shows another embodiment for a gas distribution configuration for the aerosol generator of the present invention.

Fig. 15 is a top view of one embodiment of a gas distribution plate/gas tube assembly of the aerosol generator of the present invention.

Fig. 16 is a side view of one embodiment of the gas distribution plate/gas tube assembly shown in Fig. 15.

5 Fig. 17 shows one embodiment for orienting a transducer in the aerosol generator of the present invention.

Fig. 18 is a top view of a gas manifold for distributing gas within an aerosol generator of the present invention.

Fig. 19 is a side view of the gas manifold shown in Fig. 18.

10 Fig. 20 is a top view of a generator lid of a hood design for use in an aerosol generator of the present invention.

Fig. 21 is a side view of the generator lid shown in Fig. 20.

Fig. 22 is a process block diagram of one embodiment of the process of the present invention including a droplet classifier.

15 Fig. 23 is a top view in cross section of an impactor of the present invention for use in classifying an aerosol.

Fig. 24 is a front view of a flow control plate of the impactor shown in Fig. 23.

20 Fig. 25 is a front view of a mounting plate of the impactor shown in Fig. 23.

Fig. 26 is a front view of an impactor plate assembly of the impactor shown in Fig. 23.

Fig. 27 is a side view of the impactor plate assembly shown in Fig. 26.

25 Fig. 28 is a process block diagram of one embodiment of the present invention including a particle cooler.

Fig. 29 is a top view of a gas quench cooler of the present invention.

Fig. 30 is an end view of the gas quench cooler shown in Fig. 29.

Fig. 31 is a side view of a perforated conduit of the quench cooler shown in Fig. 29.

Fig. 32 is a side view showing one embodiment of a gas quench cooler of the present invention connected with a cyclone.

Fig. 33 is a process block diagram of one embodiment of the present invention including a particle coater.

5 Fig. 34 is a block diagram of one embodiment of the present invention including a particle modifier.

Fig. 35 shows cross sections of various particle morphologies of some composite particles manufacturable according to the present invention.

10 Fig. 36 is a block diagram of one embodiment of the process of the present invention including the addition of a dry gas between the aerosol generator and the furnace.

Fig. 37 illustrates pixel regions on a display device according to the prior art.

15 Fig. 38 illustrates pixel regions on a display device according to an embodiment of the present invention.

Fig. 39 illustrates a schematic view of a CRT device according to an embodiment of the present invention.

Fig. 40 illustrates a schematic representation of pixels on a viewing screen of a CRT device according to an embodiment of the present invention.

20 Fig. 41 schematically illustrates a field emission display according to an embodiment of the present invention.

Fig. 42 schematically illustrates a cross-section of an electroluminescent display device according to an embodiment of the present invention.

25 Fig. 43 schematically illustrates an exploded view of an electroluminescent display device according to an embodiment of the present invention.

Fig. 44 illustrates an electroluminescent lamp according to an embodiment of the present invention.

30 Fig. 45 schematically illustrates a plasma display panel according to an embodiment of the present invention.

Fig. 46 schematically illustrates a fluorescent lighting element according to an embodiment of the present invention.

Fig. 47 schematically illustrates a photoluminescent phosphor powder layer dispersed on a lighting element tube according to an embodiment of the present invention.

Fig. 48 illustrates the use of an x-ray image intensifier according to an embodiment of the present invention.

Fig. 49 illustrates a cross-section of an x-ray image intensifier according to an embodiment of the present invention.

Fig. 50 illustrates an SEM photomicrograph of a phosphor powder according to an embodiment of the present invention.

Fig. 51 illustrates the particle size distribution of a phosphor powder according to an embodiment of the present invention.

Fig. 52 illustrates an x-ray diffraction pattern of a phosphor powder according to an embodiment of the present invention.

Fig. 53 illustrates an SEM photomicrograph of a phosphor powder according to an embodiment of the present invention.

Fig. 54 illustrates the particle size distribution of a phosphor powder according to an embodiment of the present invention.

Fig. 55 illustrates an x-ray diffraction pattern of a phosphor powder according to an embodiment of the present invention.

Fig. 56 illustrates an SEM photomicrograph of a phosphor powder according to an embodiment of the present invention.

Fig. 57 illustrates an SEM photomicrograph of a phosphor powder according to an embodiment of the present invention.

Fig. 58 illustrates the particle size distribution of a phosphor powder according to an embodiment of the present invention.

Fig. 59 illustrates an SEM photomicrograph of a phosphor powder according to an embodiment of the present invention.

Fig. 60 illustrates the particle size distribution of a phosphor powder according to an embodiment of the present invention.

Fig. 61 illustrates an x-ray diffraction pattern of a phosphor powder according to an embodiment of the present invention.

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DESCRIPTION OF THE INVENTION

The present invention is generally directed to phosphor powders and methods for producing the powders, as well as devices which incorporate the powders. Phosphors emit light, typically visible light, upon stimulation by an energy source such as electrons (cathodoluminescent phosphors), photons (photoluminescent phosphors), an electric field (electroluminescent phosphors), or x-ray energy (x-ray phosphors). Examples of particular phosphor compounds are listed in detail below. It will be appreciated that some phosphor compounds will fall into two or more of these categories.

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In one aspect, the present invention provides a method for preparing a particulate product. A feed of liquid-containing, flowable medium, including at least one precursor for the desired particulate product, is converted to aerosol form, with droplets of the medium being dispersed in and suspended by a carrier gas. Liquid from the droplets in the aerosol is then removed to permit formation in a dispersed state of the desired particles. In one embodiment, the particles are subjected, while still in a dispersed state, to compositional or structural modification, if desired. Compositional modification may include, for example, coating the particles. Structural modification may include, for example, crystallization, recrystallization or morphological alteration of the particles. The term powder is often used herein to refer to the particulate product of the present invention. The use of the term powder does not indicate, however, that the particulate product must be dry or in any particular environment. Although the particulate product is typically manufactured in a dry state, the particulate product may, after manufacture, be placed in a wet environment, such as in a paste or slurry.

The process of the present invention is particularly well suited for the production of particulate products of finely divided particles having a small weight average size. In addition to making particles within a desired range of weight average particle size, with the present invention the particles may be produced with a desirably narrow size distribution, thereby providing size uniformity that is desired for many applications.

In addition to control over particle size and size distribution, the method of the present invention provides significant flexibility for producing phosphor particles of varying composition, crystallinity and morphology. For example, the present invention may be used to produce homogeneous particles involving only a single phase or multi-phase particles including multiple phases. In the case of multi-phase particles, the phases may be present in a variety of morphologies. For example, one phase may be uniformly dispersed throughout a matrix of another phase. Alternatively, one phase may form an interior core while another phase forms a coating that surrounds the core. Other morphologies are also possible, as discussed more fully below.

Referring now to Fig. 1, one embodiment of the process of the present invention is described. A liquid feed 102, including at least one precursor for the desired particles, and a carrier gas 104 are fed to an aerosol generator 106 where an aerosol 108 is produced. The aerosol 108 is then fed to a furnace 110 where liquid in the aerosol 108 is removed to produce particles 112 that are dispersed in and suspended by gas exiting the furnace 110. The particles 112 are then collected in a particle collector 114 to produce a particulate product 116.

As used herein, the liquid feed 102 is a feed that includes one or more flowable liquids as the major constituent(s), such that the feed is a flowable medium. The liquid feed 102 need not comprise only liquid constituents. The liquid feed 102 may comprise only constituents in one or more liquid phase, or it may also include particulate material suspended in a liquid phase. The liquid feed 102 must, however, be capable of being atomized to form droplets of sufficiently small size for preparation of the aerosol 108. Therefore, if the liquid

feed 102 includes suspended particles, those particles should be relatively small in relation to the size of droplets in the aerosol 108. Such suspended particles should typically be smaller than about 1 μm in size, preferably smaller than about 0.5 μm in size, and more preferably smaller than about 0.3 μm in size and most preferably smaller than about 0.1 μm in size. Most preferably, the suspended particles should be colloidal. The suspended particles could be finely divided particles, or could be agglomerate masses comprised of agglomerated smaller primary particles. For example, 0.5 μm particles could be agglomerates of nanometer-sized primary particles. When the liquid feed 102 includes suspended particles, the particles typically comprise no greater than about 10 weight percent of the liquid feed.

As noted, the liquid feed 102 includes at least one precursor for preparation of the particles 112. The precursor may be a substance in either a liquid or solid phase of the liquid feed 102. Frequently, the precursor will be a material, such as a salt, dissolved in a liquid solvent of the liquid feed 102. The precursor may undergo one or more chemical reactions in the furnace 110 to assist in production of the particles 112. Alternatively, the precursor material may contribute to formation of the particles 112 without undergoing chemical reaction. This could be the case, for example, when the liquid feed 102 includes, as a precursor material, suspended particles that are not chemically modified in the furnace 110. In any event, the particles 112 comprise at least one component originally contributed by the precursor.

The liquid feed 102 may include multiple precursor materials, which may be present together in a single phase or separately in multiple phases. For example, the liquid feed 102 may include multiple precursors in solution in a single liquid vehicle. Alternatively, one precursor material could be in a solid particulate phase and a second precursor material could be in a liquid phase. Also, one precursor material could be in one liquid phase and a second precursor material could be in a second liquid phase, such as could be the case when the liquid feed 102 comprises an emulsion. Different components

contributed by different precursors may be present in the particles together in a single material phase, or the different components may be present in different material phases when the particles 112 are composites of multiple phases. Specific examples of preferred precursors for the phosphor particles of the present invention are discussed more fully below.

The carrier gas 104 may comprise any gaseous medium in which droplets produced from the liquid feed 102 may be dispersed in aerosol form. Also, the carrier gas 104 may be inert, in that the carrier gas 104 does not participate in formation of the particles 112. Alternatively, the carrier gas may have one or more active component(s) that contribute to formation of the particles 112. In that regard, the carrier gas may include one or more reactive components that react in the furnace 110 to contribute to formation of the particles 112. Preferred carrier gas compositions for the phosphor particles of the present invention are discussed more fully below.

The aerosol generator 106 atomizes the liquid feed 102 to form droplets in a manner to permit the carrier gas 104 to sweep the droplets away to form the aerosol 108. The droplets comprise liquid from the liquid feed 102. The droplets may, however, also include nonliquid material, such as one or more small particles held in the droplet by the liquid. For example, when the particles 112 are composite particles, one phase of the composite may be provided in the liquid feed 102 in the form of suspended precursor particles and a second phase of the composite may be produced in the furnace 110 from one or more precursors in the liquid phase of the liquid feed 102. Furthermore the precursor particles could be included in the liquid feed 102, and therefore also in droplets of the aerosol 108, for the purpose only of dispersing the particles for subsequent compositional or structural modification during or after processing in the furnace 110.

An important aspect of the present invention is generation of the aerosol 108 with droplets of a small average size and narrow size distribution. In this

manner, the particles 112 may be produced at a desired small size with a narrow size distribution, which are advantageous for many applications.

The aerosol generator 106 is capable of producing the aerosol 108 such that it includes droplets having a weight average size in a range having a lower
5 limit of about 1 μm and preferably about 2 μm ; and an upper limit of about 10 μm ; preferably about 7 μm , more preferably about 5 μm and most preferably about 4 μm . A weight average droplet size in a range of from about 2 μm to about 4 μm is more preferred for most applications, with a weight average droplet size of about 3 μm being particularly preferred for some applications.

10 The aerosol generator is also capable of producing the aerosol 108 such that it includes droplets in a narrow size distribution. Preferably, the droplets in the aerosol are such that at least about 70 percent (more preferably at least about 80 weight percent and most preferably at least about 85 weight percent) of the droplets are smaller than about 10 μm and more preferably at least about 70
15 weight percent (more preferably at least about 80 weight percent and most preferably at least about 85 weight percent) are smaller than about 5 μm . Furthermore, preferably no greater than about 30 weight percent, more preferably no greater than about 25 weight percent and most preferably no greater than about 20 weight percent, of the droplets in the aerosol 108 are
20 larger than about twice the weight average droplet size.

Another important aspect of the present invention is that the aerosol 108 may be generated without consuming excessive amounts of the carrier gas 104. The aerosol generator 106 is capable of producing the aerosol 108 such that it has a high loading, or high concentration, of the liquid feed 102 in droplet form.
25 In that regard, the aerosol 108 preferably includes greater than about 1×10^6 droplets per cubic centimeter of the aerosol 108, more preferably greater than about 5×10^6 droplets per cubic centimeter, still more preferably greater than about 1×10^7 droplets per cubic centimeter, and most preferably greater than about 5×10^7 droplets per cubic centimeter. That the aerosol generator 106 can
30 produce such a heavily loaded aerosol 108 is particularly surprising considering

the high quality of the aerosol 108 with respect to small average droplet size and narrow droplet size distribution. Typically, droplet loading in the aerosol is such that the volumetric ratio of liquid feed 102 to carrier gas 104 in the aerosol 108 is larger than about 0.04 milliliters of liquid feed 102 per liter of carrier gas 104 in the aerosol 108, preferably larger than about 0.083 milliliters of liquid feed 102 per liter of carrier gas 104 in the aerosol 108, more preferably larger than about 0.167 milliliters of liquid feed 102 per liter of carrier gas 104, still more preferably larger than about 0.25 milliliters of liquid feed 102 per liter of carrier gas 104, and most preferably larger than about 0.333 milliliters of liquid feed 102 per liter of carrier gas 104.

This capability of the aerosol generator 106 to produce a heavily loaded aerosol 108 is even more surprising given the high droplet output rate of which the aerosol generator 106 is capable, as discussed more fully below. It will be appreciated that the concentration of liquid feed 102 in the aerosol 108 will depend upon the specific components and attributes of the liquid feed 102 and, particularly, the size of the droplets in the aerosol 108. For example, when the average droplet size is from about 2 μm to about 4 μm , the droplet loading is preferably larger than about 0.15 milliliters of aerosol feed 102 per liter of carrier gas 104, more preferably larger than about 0.2 milliliters of liquid feed 102 per liter of carrier gas 104, even more preferably larger than about 0.2 milliliters of liquid feed 102 per liter of carrier gas 104, and most preferably larger than about 0.3 milliliters of liquid feed 102 per liter of carrier gas 104. When reference is made herein to liters of carrier gas 104, it refers to the volume that the carrier gas 104 would occupy under conditions of standard temperature and pressure.

The furnace 110 may be any suitable device for heating the aerosol 108 to evaporate liquid from the droplets of the aerosol 108 and thereby permit formation of the particles 112. The maximum average stream temperature, or conversion temperature, refers to the maximum average temperature that an aerosol stream attains while flowing through the furnace. This is typically determined by a temperature probe inserted into the furnace. Preferred

conversion temperatures for the phosphor particles according to the present invention are discussed more fully below.

Although longer residence times are possible, for many applications, residence time in the heating zone of the furnace 110 of shorter than about 4
5 seconds is typical, with shorter than about 2 seconds being preferred, such as from about 1 to 2 seconds. The residence time should be long enough, however, to assure that the particles 112 attain the desired maximum stream temperature for a given heat transfer rate. In that regard, with extremely short residence times, higher furnace temperatures could be used to increase the rate
10 of heat transfer so long as the particles 112 attain a maximum temperature within the desired stream temperature range. That mode of operation, however, is not preferred. Also, it is preferred that, in most cases, the maximum stream temperature not be attained in the furnace 110 until substantially at the end of the heating zone in the furnace 110. For example, the heating zone will often
15 include a plurality of heating sections that are each independently controllable. The maximum stream temperature should typically not be attained until the final heating section, and more preferably until substantially at the end of the last heating section. This is important to reduce the potential for thermophoretic losses of material. Also, it is noted that as used herein, residence time refers to
20 the actual time for a material to pass through the relevant process equipment. In the case of the furnace, this includes the effect of increasing velocity with gas expansion due to heating.

Typically, the furnace 110 will be a tube-shaped furnace, so that the aerosol 108 moving into and through the furnace does not encounter sharp
25 edges on which droplets could collect. Loss of droplets to collection at sharp surfaces results in a lower yield of particles 112. More important, however, the accumulation of liquid at sharp edges can result in re-release of undesirably large droplets back into the aerosol 108, which can cause contamination of the particulate product 116 with undesirably large particles. Also, over time, such

liquid collection at sharp surfaces can cause fouling of process equipment, impairing process performance.

The furnace 110 may include a heating tube made of any suitable material. The tube material may be a ceramic material, for example, mullite, silica (quartz) or alumina. Alternatively, the tube may be metallic. Advantages of using a metallic tube are low cost, ability to withstand steep temperature gradients and large thermal shocks, machinability and weldability, and ease of providing a seal between the tube and other process equipment. Disadvantages of using a metallic tube include limited operating temperature and increased reactivity in some reaction systems. One type of tube that is particularly useful according to the present invention is a lined metallic tube, such as a metal tube whose interior surface is lined with alumina.

When a metallic tube is used in the furnace 110, it is preferably a high nickel content stainless steel alloy, such as a 330 stainless steel, or a nickel-based super alloy. As noted, one of the major advantages of using a metallic tube is that the tube is relatively easy to seal with other process equipment. In that regard, flange fittings may be welded directly to the tube for connecting with other process equipment. Metallic tubes are generally preferred for spray-converting phosphor particles that do not require a maximum tube wall temperature of higher than about 1100°C during particle manufacture, which is the case for the phosphor particles according to the present invention.

Also, although the present invention is described with primary reference to a furnace reactor, which is preferred, it should be recognized that, except as noted, any other thermal reactor, including a flame reactor or a plasma reactor, could be used instead. A furnace reactor is, however, preferred, because of the generally even heating characteristic of a furnace for attaining a uniform stream temperature.

The particle collector 114, may be any suitable apparatus for collecting particles 112 to produce the particulate product 116. One preferred embodiment of the particle collector 114 uses one or more filter to separate the particles 112

from gas. Such a filter may be of any type, including a bag filter. Another preferred embodiment of the particle collector uses one or more cyclone to separate the particles 112. Other apparatus that may be used in the particle collector 114 includes an electrostatic precipitator. Also, collection should normally occur at a temperature above the condensation temperature of the gas stream in which the particles 112 are suspended. Also, collection should normally be at a temperature that is low enough to prevent significant agglomeration of the particles 112.

Of significant importance to the operation of the process of the present invention is the aerosol generator 106, which must be capable of producing a high quality aerosol with high droplet loading, as previously noted. With reference to Fig. 2, one embodiment of an aerosol generator 106 of the present invention is described. The aerosol generator 106 includes a plurality of ultrasonic transducer discs 120 that are each mounted in a transducer housing 122. The transducer housings 122 are mounted to a transducer mounting plate 124, creating an array of the ultrasonic transducer discs 120. Any convenient spacing may be used for the ultrasonic transducer discs 120. Center-to-center spacing of the ultrasonic transducer discs 120 of about 4 centimeters is often adequate. The aerosol generator 106, as shown in Fig. 2, includes forty-nine transducers in a 7 x 7 array. The array configuration is as shown in Fig. 3, which depicts the locations of the transducer housings 122 mounted to the transducer mounting plate 124.

With continued reference to Fig. 2, a separator 126, in spaced relation to the transducer discs 120, is retained between a bottom retaining plate 128 and a top retaining plate 130. Gas delivery tubes 132 are connected to gas distribution manifolds 134, which have gas delivery ports 136. The gas distribution manifolds 134 are housed within a generator body 138 that is covered by generator lid 140. A transducer driver 144, having circuitry for driving the transducer discs 120, is electronically connected with the transducer discs 120 via electrical cables 146.

During operation of the aerosol generator 106, as shown in Fig. 2, the transducer discs 120 are activated by the transducer driver 144 via the electrical cables 146. The transducers preferably vibrate at a frequency of from about 1 MHz to about 5 MHz, more preferably from about 1.5 MHz to about 3 MHz. Commonly used frequencies are at about 1.6 MHz and about 2.4 MHz. Furthermore, all of the transducer discs 110 should be operating at substantially the same frequency when an aerosol with a narrow droplet size distribution is desired. This is important because commercially available transducers can vary significantly in thickness, sometimes by as much as 10%. It is preferred, however, that the transducer discs 120 operate at frequencies within a range of 5% above and below the median transducer frequency, more preferably within a range of 2.5%, and most preferably within a range of 1%. This can be accomplished by careful selection of the transducer discs 120 so that they all preferably have thicknesses within 5% of the median transducer thickness, more preferably within 2.5%, and most preferably within 1%.

Liquid feed 102 enters through a feed inlet 148 and flows through flow channels 150 to exit through feed outlet 152. An ultrasonically transmissive fluid, typically water, enters through a water inlet 154 to fill a water bath volume 156 and flow through flow channels 158 to exit through a water outlet 160. A proper flow rate of the ultrasonically transmissive fluid is necessary to cool the transducer discs 120 and to prevent overheating of the ultrasonically transmissive fluid. Ultrasonic signals from the transducer discs 120 are transmitted, via the ultrasonically transmissive fluid, across the water bath volume 156, and ultimately across the separator 126, to the liquid feed 102 in flow channels 150.

The ultrasonic signals from the ultrasonic transducer discs 120 cause atomization cones 162 to develop in the liquid feed 102 at locations corresponding with the transducer discs 120. Carrier gas 104 is introduced into the gas delivery tubes 132 and delivered to the vicinity of the atomization cones 162 via gas delivery ports 136. Jets of carrier gas exit the gas delivery ports 136

in a direction so as to impinge on the atomization cones 162, thereby sweeping away atomized droplets of the liquid feed 102 that are being generated from the atomization cones 162 and creating the aerosol 108, which exits the aerosol generator 106 through an aerosol exit opening 164.

5 Efficient use of the carrier gas 104 is an important aspect of the aerosol generator 106. The embodiment of the aerosol generator 106 shown in Fig. 2 includes two gas exit ports per atomization cone 162, with the gas ports being positioned above the liquid medium 102 over troughs that develop between the atomization cones 162, such that the exiting carrier gas 104 is horizontally
10 directed at the surface of the atomization cones 162, thereby efficiently distributing the carrier gas 104 to critical portions of the liquid feed 102 for effective and efficient sweeping away of droplets as they form about the ultrasonically energized atomization cones 162. Furthermore, it is preferred that at least a portion of the opening of each of the gas delivery ports 136, through
15 which the carrier gas exits the gas delivery tubes, should be located below the top of the atomization cones 162 at which the carrier gas 104 is directed. This relative placement of the gas delivery ports 136 is very important to efficient use of carrier gas 104. Orientation of the gas delivery ports 136 is also important. Preferably, the gas delivery ports 136 are positioned to horizontally direct jets of
20 the carrier gas 104 at the atomization cones 162. The aerosol generator 106 permits generation of the aerosol 108 with heavy loading with droplets of the carrier liquid 102, unlike aerosol generator designs that do not efficiently focus gas delivery to the locations of droplet formation.

 Another important feature of the aerosol generator 106, as shown in Fig.
25 2, is the use of the separator 126, which protects the transducer discs 120 from direct contact with the liquid feed 102, which is often highly corrosive. The height of the separator 126 above the top of the transducer discs 120 should normally be kept as small as possible, and is often in the range of from about 1 centimeter to about 2 centimeters. The top of the liquid feed 102 in the flow
30 channels above the tops of the ultrasonic transducer discs 120 is typically in a

range of from about 2 centimeters to about 5 centimeters, whether or not the aerosol generator includes the separator 126, with a distance of about 3 to 4 centimeters being preferred. Although the aerosol generator 106 could be made without the separator 126, in which case the liquid feed 102 would be in direct contact with the transducer discs 120, the highly corrosive nature of the liquid feed 102 can often cause premature failure of the transducer discs 120. The use of the separator 126, in combination with use of the ultrasonically transmissive fluid in the water bath volume 156 to provide ultrasonic coupling, significantly extending the life of the ultrasonic transducers 120. One disadvantage of using the separator 126, however, is that the rate of droplet production from the atomization cones 162 is reduced, often by a factor of two or more, relative to designs in which the liquid feed 102 is in direct contact with the ultrasonic transducer discs 102. Even with the separator 126, however, the aerosol generator 106 used with the present invention is capable of producing a high quality aerosol with heavy droplet loading, as previously discussed. Suitable materials for the separator 126 include, for example, polyamides (such as Kapton? membranes from DuPont) and other polymer materials, glass, and plexiglass. The main requirements for the separator 126 are that it be ultrasonically transmissive, corrosion resistant and impermeable.

One alternative to using the separator 126 is to bind a corrosion-resistant protective coating onto the surface of the ultrasonic transducer discs 120, thereby preventing the liquid feed 102 from contacting the surface of the ultrasonic transducer discs 120. When the ultrasonic transducer discs 120 have a protective coating, the aerosol generator 106 will typically be constructed without the water bath volume 156 and the liquid feed 102 will flow directly over the ultrasonic transducer discs 120. Examples of such protective coating materials include platinum, gold, TEFLON?, epoxies and various plastics. Such a coating typically significantly extends transducer life. Also, when operating without the separator 126, the aerosol generator 106 will typically produce the

aerosol 108 with a much higher droplet loading than when the separator 126 is used.

One surprising finding with operation of the aerosol generator 106 of the present invention is that the droplet loading in the aerosol may be affected by the temperature of the liquid feed 102 as well as the temperature of the water bath volume 156. It has been found that when the liquid feed 102 and/or the water bath volume includes an aqueous liquid at an elevated temperature, the droplet loading increases significantly. The temperature of the liquid feed 102 and/or the water bath volume 156 is preferably higher than about 30°C, and more preferably higher than about 35°C. If the temperature becomes too high, however, it can have a detrimental effect on droplet loading in the aerosol 108. Therefore, the temperature of the liquid feed 102 and/or the water bath volume should generally be lower than about 50°C, and preferably lower than about 45°C. Either the liquid feed 102 or the water bath volume 156 may be maintained at the desired temperature in any suitable fashion. For example, the portion of the aerosol generator 106 where the liquid feed 102 is converted to the aerosol 108 could be maintained at a constant elevated temperature. Alternatively, the liquid feed 102 could be delivered to the aerosol generator 106 from a constant temperature bath maintained separate from the aerosol generator 106.

The design for the aerosol generator 106 based on an array of ultrasonic transducers is versatile and is easily modified to accommodate different generator sizes for different specialty applications. The aerosol generator 106 may be designed to include a plurality of ultrasonic transducers in any convenient number. Even for smaller scale production, however, the aerosol generator 106 preferably has at least nine ultrasonic transducers, more preferably at least 16 ultrasonic transducers, and even more preferably at least 25 ultrasonic transducers. For larger scale production, however, the aerosol generator 106 includes at least 40 ultrasonic transducers, more preferably at least 100 ultrasonic transducers, and even more preferably at least 400

ultrasonic transducers. In some large volume applications, the aerosol generator may have at least 1000 ultrasonic transducers.

Figs. 4-21 show component designs for an aerosol generator 106 including an array of 400 ultrasonic transducers. Referring first to Figs. 4 and 5, the transducer mounting plate 124 is shown with a design to accommodate an array of 400 ultrasonic transducers, arranged in four subarrays of 100 ultrasonic transducers each. The transducer mounting plate 124 has integral vertical walls 172 for containing the ultrasonically transmissive fluid, typically water, in a water bath similar to the water bath volume 156 described previously with reference to Fig. 2.

As shown in Figs. 4 and 5, four hundred transducer mounting receptacles 174 are provided in the transducer mounting plate 124 for mounting ultrasonic transducers for the desired array. With reference to Fig. 6, the profile of an individual transducer mounting receptacle 174 is shown. A mounting seat 176 accepts an ultrasonic transducer for mounting, with a mounted ultrasonic transducer being held in place via screw holes 178. Opposite the mounting receptacle 176 is a flared opening 180 through which an ultrasonic signal may be transmitted for the purpose of generating the aerosol 108, as previously described with reference to Fig. 2.

A preferred transducer mounting configuration, however, is shown in Fig. 7 for another configuration for the transducer mounting plate 124. As seen in Fig. 7, an ultrasonic transducer disc 120 is mounted to the transducer mounting plate 124 by use of a compression screw 177 threaded into a threaded receptacle 179. The compression screw 177 bears against the ultrasonic transducer disc 120, causing an o-ring 181, situated in an o-ring seat 182 on the transducer mounting plate, to be compressed to form a seal between the transducer mounting plate 124 and the ultrasonic transducer disc 120. This type of transducer mounting is particularly preferred when the ultrasonic transducer disc 120 includes a protective surface coating, as discussed previously, because the seal of the o-ring to the ultrasonic transducer disc 120 will be inside of the

outer edge of the protective seal, thereby preventing liquid from penetrating under the protective surface coating from the edges of the ultrasonic transducer disc 120.

Referring now to Fig. 8, the bottom retaining plate 128 for a 400 transducer array is shown having a design for mating with the transducer mounting plate 124 (shown in Figs. 4-5). The bottom retaining plate 128 has eighty openings 184, arranged in four subgroups 186 of twenty openings 184 each. Each of the openings 184 corresponds with five of the transducer mounting receptacles 174 (shown in Figs. 4 and 5) when the bottom retaining plate 128 is mated with the transducer mounting plate 124 to create a volume for a water bath between the transducer mounting plate 124 and the bottom retaining plate 128. The openings 184, therefore, provide a pathway for ultrasonic signals generated by ultrasonic transducers to be transmitted through the bottom retaining plate.

Referring now to Figs. 9 and 10, a liquid feed box 190 for a 400 transducer array is shown having the top retaining plate 130 designed to fit over the bottom retaining plate 128 (shown in Fig. 8), with a separator 126 (not shown) being retained between the bottom retaining plate 128 and the top retaining plate 130 when the aerosol generator 106 is assembled. The liquid feed box 190 also includes vertically extending walls 192 for containing the liquid feed 102 when the aerosol generator is in operation. Also shown in Figs. 9 and 10 is the feed inlet 148 and the feed outlet 152. An adjustable weir 198 determines the level of liquid feed 102 in the liquid feed box 190 during operation of the aerosol generator 106.

The top retaining plate 130 of the liquid feed box 190 has eighty openings 194 therethrough, which are arranged in four subgroups 196 of twenty openings 194 each. The openings 194 of the top retaining plate 130 correspond in size with the openings 184 of the bottom retaining plate 128 (shown in Fig. 8). When the aerosol generator 106 is assembled, the openings 194 through the top retaining plate 130 and the openings 184 through the bottom retaining plate 128

are aligned, with the separator 126 positioned therebetween, to permit transmission of ultrasonic signals when the aerosol generator 106 is in operation.

Referring now to Figs. 9-11, a plurality of gas tube feed-through holes 202 extend through the vertically extending walls 192 to either side of the assembly including the feed inlet 148 and feed outlet 152 of the liquid feed box 190. The gas tube feed-through holes 202 are designed to permit insertion therethrough of gas tubes 208 of a design as shown in Fig. 11. When the aerosol generator 106 is assembled, a gas tube 208 is inserted through each of the gas tube feed-through holes 202 so that gas delivery ports 136 in the gas tube 208 will be properly positioned and aligned adjacent the openings 194 in the top retaining plate 130 for delivery of gas to atomization cones that develop in the liquid feed box 190 during operation of the aerosol generator 106. The gas delivery ports 136 are typically holes having a diameter of from about 1.5 millimeters to about 3.5 millimeters.

Referring now to Fig. 12, a partial view of the liquid feed box 190 is shown with gas tubes 208A, 208B and 208C positioned adjacent to the openings 194 through the top retaining plate 130. Also shown in Fig. 12 are the relative locations that ultrasonic transducer discs 120 would occupy when the aerosol generator 106 is assembled. As seen in Fig. 12, the gas tube 208A, which is at the edge of the array, has five gas delivery ports 136. Each of the gas delivery ports 136 is positioned to divert carrier gas 104 to a different one of atomization cones that develop over the array of ultrasonic transducer discs 120 when the aerosol generator 106 is operating. The gas tube 208B, which is one row in from the edge of the array, is a shorter tube that has ten gas delivery ports 136, five each on opposing sides of the gas tube 208B. The gas tube 208B, therefore, has gas delivery ports 136 for delivering gas to atomization cones corresponding with each of ten ultrasonic transducer discs 120. The third gas tube, 208C, is a longer tube that also has ten gas delivery ports 136 for delivering gas to atomization cones corresponding with ten ultrasonic transducer discs 120. The

design shown in Fig. 12, therefore, includes one gas delivery port per ultrasonic transducer disc 120. Although this is a lower density of gas delivery ports 136 than for the embodiment of the aerosol generator 106 shown in Fig. 2, which includes two gas delivery ports per ultrasonic transducer disc 120, the design shown in Fig. 12 is, nevertheless, capable of producing a dense, high-quality aerosol without unnecessary waste of gas.

Referring now to Fig. 13, the flow of carrier gas 104 relative to atomization cones 162 during operation of the aerosol generator 106 having a gas distribution configuration to deliver carrier gas 104 from gas delivery ports on both sides of the gas tubes 208, as was shown for the gas tubes 208A, 208B and 208C in the gas distribution configuration shown in Fig. 11. The carrier gas 104 sweeps both directions from each of the gas tubes 208.

An alternative, and preferred, flow for carrier gas 104 is shown in Fig. 14. As shown in Fig. 14, carrier gas 104 is delivered from only one side of each of the gas tubes 208. This results in a sweep of carrier gas from all of the gas tubes 208 toward a central area 212. This results in a more uniform flow pattern for aerosol generation that may significantly enhance the efficiency with which the carrier gas 104 is used to produce an aerosol. The aerosol that is generated, therefore, tends to be more heavily loaded with liquid droplets.

Another configuration for distributing carrier gas in the aerosol generator 106 is shown in Figs. 15 and 16. In this configuration, the gas tubes 208 are hung from a gas distribution plate 216 adjacent gas flow holes 218 through the gas distribution plate 216. In the aerosol generator 106, the gas distribution plate 216 would be mounted above the liquid feed, with the gas flow holes positioned to correspond with an underlying ultrasonic transducer. Referring specifically to Fig. 16, when the ultrasonic generator 106 is in operation, atomization cones 162 develop through the gas flow holes 218, and the gas tubes 208 are located such that carrier gas 104 exiting from ports in the gas tubes 208 impinge on the atomization cones and flow upward through the gas flow holes. The gas flow holes 218, therefore, act to assist in efficiently

distributing the carrier gas 104 about the atomization cones 162 for aerosol formation. It should be appreciated that the gas distribution plates 218 can be made to accommodate any number of the gas tubes 208 and gas flow holes 218. For convenience of illustration, the embodiment shown in Figs. 15 and 16 shows a design having only two of the gas tubes 208 and only 16 of the gas flow holes 218. Also, it should be appreciated that the gas distribution plate 216 could be used alone, without the gas tubes 208. In that case, a slight positive pressure of carrier gas 104 would be maintained under the gas distribution plate 216 and the gas flow holes 218 would be sized to maintain the proper velocity of carrier gas 104 through the gas flow holes 218 for efficient aerosol generation. Because of the relative complexity of operating in that mode, however, it is not preferred.

Aerosol generation may also be enhanced through mounting of ultrasonic transducers at a slight angle and directing the carrier gas at resulting atomization cones such that the atomization cones are tilting in the same direction as the direction of flow of carrier gas. Referring to Fig. 17, an ultrasonic transducer disc 120 is shown. The ultrasonic transducer disc 120 is tilted at a tilt angle 114 (typically less than 10 degrees), so that the atomization cone 162 will also have a tilt. It is preferred that the direction of flow of the carrier gas 104 directed at the atomization cone 162 is in the same direction as the tilt of the atomization cone 162.

Referring now to Figs. 18 and 19, a gas manifold 220 is shown for distributing gas to the gas tubes 208 in a 400 transducer array design. The gas manifold 220 includes a gas distribution box 222 and piping stubs 224 for connection with gas tubes 208 (shown in Fig. 11). Inside the gas distribution box 222 are two gas distribution plates 226 that form a flow path to assist in distributing the gas equally throughout the gas distribution box 222, to promote substantially equal delivery of gas through the piping stubs 224. The gas manifold 220, as shown in Figs. 18 and 19, is designed to feed eleven gas tubes

208. For the 400 transducer design, a total of four gas manifolds 220 are required.

Referring now to Figs 23 and 24, the generator lid 140 is shown for a 400 transducer array design. The generator lid 140 mates with and covers the liquid feed box 190 (shown in Figs. 9 and 10). The generator lid 140, as shown in Figs. 20 and 21, has a hood design to permit easy collection of the aerosol 108 without subjecting droplets in the aerosol 108 to sharp edges on which droplets may coalesce and be lost, and possibly interfere with the proper operation of the aerosol generator 106. When the aerosol generator 106 is in operation, the aerosol 108 would be withdrawn via the aerosol exit opening 164 through the generator cover 140.

It is important that the aerosol stream that is fed to the furnace 110 have a high droplet flow rate and high droplet loading as would be required for most industrial applications. With the present invention, the aerosol stream fed to the furnace preferably includes a droplet flow of greater than about 0.5 liters per hour, more preferably greater than about 2 liters per hour, still more preferably greater than about 5 liters per hour, even more preferably greater than about 10 liters per hour, particularly greater than about 50 liters per hour and most preferably greater than about 100 liters per hour; and with the droplet loading being typically greater than about 0.04 milliliters of droplets per liter of carrier gas, preferably greater than about 0.083 milliliters of droplets per liter of carrier gas 104, more preferably greater than about 0.167 milliliters of droplets per liter of carrier gas 104, still more preferably greater than about 0.25 milliliters of droplets per liter of carrier gas 104, particularly greater than about 0.33 milliliters of droplets per liter of carrier gas 104 and most preferably greater than about 0.83 milliliters of droplets per liter of carrier gas 104.

In addition to the foregoing, it has been found to be advantageous according to the present invention to provide means for adjusting the concentration of precursor in the liquid feed. More specifically, it has been found that during aerosol production, the precursor solution can concentrate due to the

preferential evaporation of water from the liquid. As a result, it is desirable to provide water to the liquid either on a substantially continuous basis or periodically to maintain the concentration of the precursors within an acceptable range. In some instances, it may also be necessary to add small amounts of precursors if there is any preferential evaporation of precursor materials from the liquid.

The aerosol generator 106 of the present invention produces a concentrated, high quality aerosol of micro-sized droplets having a relatively narrow size distribution. It has been found, however, that for many applications the process of the present invention is significantly enhanced by further classifying by size the droplets in the aerosol 108 prior to introduction of the droplets into the furnace 110. In this manner, the size and size distribution of particles in the particulate product 116 are further controlled.

Referring now to Fig. 22, a process flow diagram is shown for one embodiment of the process of the present invention including such droplet classification. As shown in Fig. 22, the aerosol 108 from the aerosol generator 106 goes to a droplet classifier 280 where oversized droplets are removed from the aerosol 108 to prepare a classified aerosol 282. Liquid 284 from the oversized droplets that are being removed is drained from the droplet classifier 280. This drained liquid 284 may advantageously be recycled for use in preparing additional liquid feed 102.

Any suitable droplet classifier may be used for removing droplets above a predetermined size. For example, a cyclone could be used to remove over-size droplets. A preferred droplet classifier for many applications, however, is an impactor. One embodiment of an impactor for use with the present invention will now be described with reference to Figs. 23-27.

As seen in Fig. 23, an impactor 288 has disposed in a flow conduit 286 a flow control plate 290 and an impactor plate assembly 292. The flow control plate 290 is conveniently mounted on a mounting plate 294.

The flow control plate 290 is used to channel the flow of the aerosol stream toward the impactor plate assembly 292 in a manner with controlled flow characteristics that are desirable for proper impaction of oversize droplets on the impactor plate assembly 292 for removal through the drains 296 and 314. One
5 embodiment of the flow control plate 290 is shown in Fig. 24. The flow control plate 290 has an array of circular flow ports 296 for channeling flow of the aerosol 108 towards the impactor plate assembly 292 with the desired flow characteristics.

Details of the mounting plate 294 are shown in Fig. 25. The mounting
10 plate 294 has a mounting flange 298 with a large diameter flow opening 300 passing therethrough to permit access of the aerosol 108 to the flow ports 296 of the flow control plate 290 (shown in Fig. 24).

Referring now to Figs. 26 and 27, one embodiment of an impactor plate assembly 292 is shown. The impactor plate assembly 292 includes an impactor
15 plate 302 and mounting brackets 304 and 306 used to mount the impactor plate 302 inside of the flow conduit 286. The impactor plate 302 and the flow channel plate 290 are designed so that droplets larger than a predetermined size will have momentum that is too large for those particles to change flow direction to navigate around the impactor plate 302.

During operation of the impactor 288, the aerosol 108 from the aerosol
20 generator 106 passes through the upstream flow control plate 290. Most of the droplets in the aerosol navigate around the impactor plate 302 and exit the impactor 288 through the downstream flow control plate 290 in the classified aerosol 282. Droplets in the aerosol 108 that are too large to navigate around
25 the impactor plate 302 will impact on the impactor plate 302 and drain through the drain 296 to be collected with the drained liquid 284 (as shown in Fig. 23).

The configuration of the impactor plate 302 shown in Fig. 22 represents only one of many possible configurations for the impactor plate 302. For example, the impactor 288 could include an upstream flow control plate 290
30 having vertically extending flow slits therethrough that are offset from vertically

extending flow slits through the impactor plate 302, such that droplets too large to navigate the change in flow due to the offset of the flow slits between the flow control plate 290 and the impactor plate 302 would impact on the impactor plate 302 to be drained away. Other designs are also possible.

5 In a preferred embodiment of the present invention, the droplet classifier 280 is typically designed to remove droplets from the aerosol 108 that are larger than about 15 μm in size, more preferably to remove droplets larger than about 10 μm in size, even more preferably to remove droplets of a size larger than about 8 μm in size and most preferably to remove droplets larger than about 5
10 μm in size. The droplet classification size in the droplet classifier is preferably smaller than about 15 μm , more preferably smaller than about 10 μm , even more preferably smaller than about 8 μm and most preferably smaller than about 5 μm . The classification size, also called the classification cut point, is that size at which half of the droplets of that size are removed and half of the droplets of
15 that size are retained. Depending upon the specific application, however, the droplet classification size may be varied, such as by changing the spacing between the impactor plate 302 and the flow control plate 290 or increasing or decreasing aerosol velocity through the jets in the flow control plate 290.

Because the aerosol generator 106 of the present invention initially
20 produces a high quality aerosol 108, having a relatively narrow size distribution of droplets, typically less than about 30 weight percent of liquid feed 102 in the aerosol 108 is removed as the drain liquid 284 in the droplet classifier 288, with preferably less than about 25 weight percent being removed, even more preferably less than about 20 weight percent being removed and most preferably
25 less than about 15 weight percent being removed. Minimizing the removal of liquid feed 102 from the aerosol 108 is particularly important for commercial applications to increase the yield of high quality particulate product 116. It should be noted, however, that because of the superior performance of the aerosol generator 106, it is frequently not required to use an impactor or other
30 droplet classifier to obtain a desired absence of oversize droplets to the furnace.

This is a major advantage, because the added complexity and liquid losses accompanying use of an impactor may often be avoided with the process of the present invention.

With some applications of the process of the present invention, it may be possible to collect the particles 112 directly from the output of the furnace 110. More often, however, it will be desirable to cool the particles 112 exiting the furnace 110 prior to collection of the particles 112 in the particle collector 114. Referring now to Fig. 28, one embodiment of the process of the present invention is shown in which the particles 112 exiting the furnace 110 are sent to a particle cooler 320 to produce a cooled particle stream 322, which is then feed to the particle collector 114. Although the particle cooler 320 may be any cooling apparatus capable of cooling the particles 112 to the desired temperature for introduction into the particle collector 114, traditional heat exchanger designs are not preferred. This is because a traditional heat exchanger design ordinarily directly subjects the aerosol stream, in which the hot particles 112 are suspended, to cool surfaces. In that situation, significant losses of the particles 112 occur due to thermophoretic deposition of the hot particles 112 on the cool surfaces of the heat exchanger. According to the present invention, a gas quench apparatus is provided for use as the particle cooler 320 that significantly reduces thermophoretic losses compared to a traditional heat exchanger.

Referring now to Figs. 29-31, one embodiment of a gas quench cooler 330 is shown. The gas quench cooler includes a perforated conduit 332 housed inside of a cooler housing 334 with an annular space 336 located between the cooler housing 334 and the perforated conduit 332. In fluid communication with the annular space 336 is a quench gas inlet box 338, inside of which is disposed a portion of an aerosol outlet conduit 340. The perforated conduit 332 extends between the aerosol outlet conduit 340 and an aerosol inlet conduit 342. Attached to an opening into the quench gas inlet box 338 are two quench gas feed tubes 344. Referring specifically to Fig. 31, the perforated tube 332 is shown. The perforated tube 332 has a plurality of openings 345. The openings

345, when the perforated conduit 332 is assembled into the gas quench cooler 330, permit the flow of quench gas 346 from the annular space 336 into the interior space 348 of the perforated conduit 332. Although the openings 345 are shown as being round holes, any shape of opening could be used, such as slits. Also, the perforated conduit 332 could be a porous screen. Two heat radiation shields 347 prevent downstream radiant heating from the furnace. In most instances, however, it will not be necessary to include the heat radiation shields 347, because downstream radiant heating from the furnace is normally not a significant problem. Use of the heat radiation shields 347 is not preferred due to particulate losses that accompany their use.

With continued reference to Figs. 29-31, operation of the gas quench cooler 330 will now be described. During operation, the particles 112, carried by and dispersed in a gas stream, enter the gas quench cooler 330 through the aerosol inlet conduit 342 and flow into the interior space 348 of perforated conduit 332. Quench gas 346 is introduced through the quench gas feed tubes 344 into the quench gas inlet box 338. Quench gas 346 entering the quench gas inlet box 338 encounters the outer surface of the aerosol outlet conduit 340, forcing the quench gas 346 to flow, in a spiraling, swirling manner, into the annular space 336, where the quench gas 346 flows through the openings 345 through the walls of the perforated conduit 332. Preferably, the gas 346 retains some swirling motion even after passing into the interior space 348. In this way, the particles 112 are quickly cooled with low losses of particles to the walls of the gas quench cooler 330. In this manner, the quench gas 346 enters in a radial direction into the interior space 348 of the perforated conduit 332 around the entire periphery, or circumference, of the perforated conduit 332 and over the entire length of the perforated conduit 332. The cool quench gas 346 mixes with and cools the hot particles 112, which then exit through the aerosol outlet conduit 340 as the cooled particle stream 322. The cooled particle stream 322 can then be sent to the particle collector 114 for particle collection. The temperature of the cooled particle stream 322 is controlled by introducing more

or less quench gas. Also, as shown in Fig. 29, the quench gas 346 is fed into the quench cooler 330 in counter flow to flow of the particles. Alternatively, the quench cooler could be designed so that the quench gas 346 is fed into the quench cooler in concurrent flow with the flow of the particles 112. The amount of quench gas 346 fed to the gas quench cooler 330 will depend upon the specific material being made and the specific operating conditions. The quantity of quench gas 346 used, however, must be sufficient to reduce the temperature of the aerosol stream including the particles 112 to the desired temperature. Typically, the particles 112 are cooled to a temperature at least below about 200°C, and often lower. The only limitation on how much the particles 112 are cooled is that the cooled particle stream 322 must be at a temperature that is above the condensation temperature for water as another condensable vapor in the stream. The temperature of the cooled particle stream 322 is often at a temperature of from about 50°C to about 120°C.

Because of the entry of quench gas 346 into the interior space 348 of the perforated conduit 322 in a radial direction about the entire circumference and length of the perforated conduit 322, a buffer of the cool quench gas 346 is formed about the inner wall of the perforated conduit 332, thereby significantly inhibiting the loss of hot particles 112 due to thermophoretic deposition on the cool wall of the perforated conduit 332. In operation, the quench gas 346 exiting the openings 345 and entering into the interior space 348 should have a radial velocity (velocity inward toward the center of the circular cross-section of the perforated conduit 332) of larger than the thermophoretic velocity of the particles 112 inside the perforated conduit 332 in a direction radially outward toward the perforated wall of the perforated conduit 332.

As seen in Figs. 29-31, the gas quench cooler 330 includes a flow path for the particles 112 through the gas quench cooler of a substantially constant cross-sectional shape and area. Preferably, the flow path through the gas quench cooler 330 will have the same cross-sectional shape and area as the flow path through the furnace 110 and through the conduit delivering the aerosol

108 from the aerosol generator 106 to the furnace 110. In one embodiment, however, it may be necessary to reduce the cross-sectional area available for flow prior to the particle collector 114. This is the case, for example, when the particle collector includes a cyclone for separating particles in the cooled particle stream 322 from gas in the cooled particle stream 322. This is because of the high inlet velocity requirements into cyclone separators.

Referring now to Fig. 32, one embodiment of the gas quench cooler 330 is shown in combination with a cyclone separator 392. The perforated conduit 332 has a continuously decreasing cross-sectional area for flow to increase the velocity of flow to the proper value for the feed to cyclone separator 392. Attached to the cyclone separator 392 is a bag filter 394 for final clean-up of overflow from the cyclone separator 392. Separated particles exit with underflow from the cyclone separator 392 and may be collected in any convenient container. The use of cyclone separation is particularly preferred for powder having a weight average size of larger than about 1 μm , although a series of cyclones may sometimes be needed to get the desired degree of separation. Cyclone separation is particularly preferred for powders having a weight average size of larger than about 1.5 μm . Also, cyclone separation is best suited for high density materials. Preferably, when particles are separated using a cyclone, the particles are of a composition with specific gravity of greater than about 5.

In an additional embodiment, the process of the present invention can also incorporate compositional modification of the particles 112 exiting the furnace. Most commonly, the compositional modification will involve forming on the particles 112 a material phase that is different than that of the particles 112, such as by coating the particles 112 with a coating material. One embodiment of the process of the present invention incorporating particle coating is shown in Fig. 33. As shown in Fig. 33, the particles 112 exiting from the furnace 110 go to a particle coater 350 where a coating is placed over the outer surface of the particles 112 to form coated particles 352, which are then sent to the particle collector 114 for preparation of the particulate product 116. Coating

methodologies employed in the particle coater 350 are discussed in more detail below.

With continued reference primarily to Fig. 33, in a preferred embodiment, when the particles 112 are coated according to the process of the present invention, the particles 112 are also manufactured via the aerosol process of the present invention, as previously described. The process of the present invention can, however, be used to coat particles that have been premanufactured by a different process, such as by a liquid precipitation route. When coating particles that have been premanufactured by a different route, such as by liquid precipitation, it is preferred that the particles remain in a dispersed state from the time of manufacture to the time that the particles are introduced in slurry form into the aerosol generator 106 for preparation of the aerosol 108 to form the dry particles 112 in the furnace 110, which particles 112 can then be coated in the particle coater 350. Maintaining particles in a dispersed state from manufacture through coating avoids problems associated with agglomeration and redispersion of particles if particles must be redispersed in the liquid feed 102 for feed to the aerosol generator 106. For example, for particles originally precipitated from a liquid medium, the liquid medium containing the suspended precipitated particles could be used to form the liquid feed 102 to the aerosol generator 106. It should be noted that the particle coater 350 could be an integral extension of the furnace 110 or could be a separate piece of equipment.

In a further embodiment of the present invention, following preparation of the particles 112 in the furnace 110, the particles 112 may then be structurally modified to impart desired physical and chemical properties. Referring now to Fig. 34, one embodiment of the process of the present invention is shown including such structural particle modification. The particles 112 exiting the furnace 110 go to a particle modifier 360 where the particles are structurally modified to form modified particles 362, which are then sent to the particle collector 114 for preparation of the particulate product 116. The particle modifier 360 is typically a furnace, such as an annealing furnace, which may be integral

with the furnace 110 or may be a separate heating device. Regardless, it is important that the particle modifier 360 have temperature control that is independent of the furnace 110, so that the proper conditions for particle modification may be provided separate from conditions required of the furnace 110 to prepare the particles 112. The particle modifier 360, therefore, typically provides a temperature controlled environment and necessary residence time to effect the desired structural and/or chemical modification of the particles 112.

The structural modification that occurs in the particle modifier 360 may be any modification to the crystalline structure or morphology of the particles 112. Preferably, the particles 112 are heat treated in the particle modifier 360 to further convert and densify the particles 112 or to recrystallize the particles 112 into a polycrystalline or single crystalline phosphor form. Also, especially in the case of composite particles 112, the particles may be annealed for a sufficient time to permit redistribution within the particles 112 of different material phases. Particularly preferred parameters for such processes are discussed in more detail below.

The initial morphology of composite particles made in the furnace 110, according to the present invention, could take a variety of forms, depending upon the specified materials involved and the specific processing conditions. Examples of some possible composite particle morphologies, manufacturable according to the present invention are shown in Fig. 35. These morphologies could be of the particles as initially produced in the furnace 110 or that result from structural modification in the particle modifier 360. Furthermore, the composite particles could include a mixture of the morphological attributes shown in Fig. 35.

Aerosol generation with the process of the present invention has thus far been described with respect to the ultrasonic aerosol generator. Use of the ultrasonic generator is preferred for the process of the present invention because of the extremely high quality and dense aerosol generated. In some instances, however, the aerosol generation for the process of the present invention may

have a different design depending upon the specific application. For example, when larger particles are desired, such as those having a weight average size of larger than about 3 μm , a spray nozzle atomizer may be preferred. For smaller-particle applications, however, and particularly for those applications
5 utilizing particles smaller than about 3 μm , and preferably smaller than about 2 μm in size, as is often desired with the phosphor particles of the present invention, the ultrasonic generator, as described herein, is particularly preferred. In that regard, the ultrasonic generator of the present invention is particularly preferred for when making particles with an average size of from about 0.2 μm
10 to about 3 μm .

Although ultrasonic aerosol generators have been used for medical applications and home humidifiers, the use of ultrasonic generators for spray pyrolysis particle manufacture has largely been confined to small-scale, experimental situations. The ultrasonic aerosol generator of the present
15 invention described with reference to Figures 5-24, however, is well suited for commercial production of high quality powders with a small average size and a narrow size distribution. In that regard, the aerosol generator produces a high quality aerosol, with heavy droplet loading and at a high rate of production. Such a combination of small droplet size, narrow size distribution, heavy droplet
20 loading, and high production rate provide significant advantages over existing aerosol generators that usually suffer from at least one of inadequately narrow size distribution, undesirably low droplet loading, or unacceptably low production rate.

Through the careful and controlled design of the ultrasonic generator of
25 the present invention, an aerosol may be produced typically having greater than about 70 weight percent (and preferably greater than about 80 weight percent) of droplets in the size range of from about 1 μm to about 10 μm , preferably in a size range of from about 1 μm to about 5 μm and more preferably from about 2 μm to about 4 μm .

Also, the ultrasonic generator of the present invention is capable of delivering high output rates of liquid feed in the aerosol. The rate of liquid feed, at the high liquid loadings previously described, is preferably greater than about 25 milliliters per hour per transducer, more preferably greater than about 37.5 milliliters per hour per transducer, even more preferably greater than about 50 milliliters per hour per transducer and most preferably greater than about 100 milliliters per hour per transducer. This high level of performance is desirable for commercial operations and is accomplished with the present invention with a relatively simple design including a single precursor bath over an array of ultrasonic transducers. The ultrasonic generator is made for high aerosol production rates at a high droplet loading, and with a narrow size distribution of droplets. The generator preferably produces an aerosol at a rate of greater than about 0.5 liter per hour of droplets, more preferably greater than about 2 liters per hour of droplets, still more preferably greater than about 5 liters per hour of droplets, even more preferably greater than about 10 liters per hour of droplets and most preferably greater than about 40 liters per hour of droplets. For example, when the aerosol generator has a 400 transducer design, as described with reference to Figures 7-24, the aerosol generator is capable of producing a high quality aerosol having high droplet loading as previously described, at a total production rate of preferably greater than about 10 liters per hour of liquid feed, more preferably greater than about 15 liters per hour of liquid feed, even more preferably greater than about 20 liters per hour of liquid feed and most preferably greater than about 40 liters per hour of liquid feed.

Under most operating conditions, when using such an aerosol generator, total particulate product produced is preferably greater than about 0.5 gram per hour per transducer, more preferably greater than about 0.75 gram per hour per transducer, even more preferably greater than about 1.0 gram per hour per transducer and most preferably greater than about 2.0 grams per hour per transducer. The mass of powder produced per unit time will be influenced by the molecular weight of the compound.

One significant aspect of the process of the present invention for manufacturing particulate materials is the unique flow characteristics encountered in the furnace relative to laboratory scale systems. The maximum Reynolds number attained for flow in the furnace 110 with the present invention is very high, typically in excess of 500, preferably in excess of 1,000 and more preferably in excess of 2,000. In most instances, however, the maximum Reynolds number for flow in the furnace will not exceed 10,000, and preferably will not exceed 5,000. This is significantly different from lab-scale systems where the Reynolds number for flow in a reactor is typically lower than 50 and rarely exceeds 100.

The Reynolds number is a dimensionless quantity characterizing flow of a fluid, which, for flow through a circular cross sectional conduit is defined as:

$$Re = \frac{\rho v d}{\mu}$$

where: ρ = fluid density;
 v = fluid mean velocity;
 d = conduit inside diameter; and
 μ = fluid viscosity.

It should be noted that the values for density, velocity and viscosity will vary along the length of the furnace 110. The maximum Reynolds number in the furnace 110 is typically attained when the average stream temperature is at a maximum, because the gas velocity is at a very high value due to gas expansion when heated.

One problem with operating under flow conditions at a high Reynolds number is that undesirable volatilization of components is much more likely to occur than in systems having flow characteristics as found in laboratory-scale systems. The volatilization problem occurs with the present invention, because the furnace is typically operated over a substantial section of the heating zone in a constant wall heat flux mode, due to limitations in heat transfer capability. This is significantly different than operation of a furnace at a laboratory scale,

which typically involves operation of most of the heating zone of the furnace in a uniform wall temperature mode, because the heating load is sufficiently small that the system is not heat transfer limited.

5 With the present invention, it is typically preferred to heat the aerosol stream in the heating zone of the furnace as quickly as possible to the desired temperature range for particle manufacture. Because of flow characteristics in the furnace and heat transfer limitations, during rapid heating of the aerosol the wall temperature of the furnace can significantly exceed the maximum average target temperature for the stream. This is a problem because, even though the
10 average stream temperature may be within the range desired, the wall temperature may become so hot that components in the vicinity of the wall are subjected to temperatures high enough to undesirably volatilize the components. This volatilization near the wall of the furnace can cause formation of significant quantities of ultrafine particles that are outside of the size range desired.

15 Therefore, with the present invention, it is preferred that when the flow characteristics in the furnace are such that the Reynolds number through any part of the furnace exceeds 500, more preferably exceeds 1,000, and most preferably exceeds 2,000, the maximum wall temperature in the furnace should be kept at a temperature that is below the temperature at which a desired
20 component of the final particles would exert a vapor pressure not exceeding about 200 millitorr, more preferably not exceeding about 100 millitorr, and most preferably not exceeding about 50 millitorr. Furthermore, the maximum wall temperature in the furnace should also be kept below a temperature at which an intermediate component, from which a final component is to be at least partially
25 derived, should also have a vapor pressure not exceeding the magnitudes noted for components of the final product.

30 In addition to maintaining the furnace wall temperature below a level that could create volatilization problems, it is also important that this not be accomplished at the expense of the desired average stream temperature. The maximum average stream temperature must be maintained at a high enough

level so that the particles will have a desired high density. The maximum average stream temperature should, however, generally be a temperature at which a component in the final particles, or an intermediate component from which a component in the final particles is at least partially derived, would exert
5 a vapor pressure not exceeding about 100 millitorr, preferably not exceeding about 50 millitorr, and most preferably not exceeding about 25 millitorr.

So long as the maximum wall temperature and the average stream temperature are kept below the point at which detrimental volatilization occurs, it is generally desirable to heat the stream as fast as possible and to remove
10 resulting particles from the furnace immediately after the maximum stream temperature is reached in the furnace. With the present invention, the average residence time in the heating zone of the furnace may typically be maintained at shorter than about 4 seconds, such as from about 1 to 2 seconds.

Another significant issue with respect to operating the process of the present invention, which includes high aerosol flow rates, is loss within the
15 system of materials intended for incorporation into the final particulate product. Material losses in the system can be quite high if the system is not properly operated. If system losses are too high, the process would not be practical for use in the manufacture of particulate products of many materials. This has
20 typically not been a major consideration with laboratory-scale systems.

One significant potential for loss with the process of the present invention is thermophoretic losses that occur when a hot aerosol stream is in the presence of a cooler surface. In that regard, the use of the quench cooler, as previously
25 described, with the process of the present invention provides an efficient way to cool the particles without unreasonably high thermophoretic losses. There is also, however, significant potential for losses occurring near the end of the furnace and between the furnace and the cooling unit.

It has been found that thermophoretic losses in the back end of the furnace can be significantly controlled if the heating zone of the furnace is
30 operated such that the maximum stream temperature is not attained until near

the end of the heating zone in the furnace, and at least not until the last third of the heating zone. When the heating zone includes a plurality of heating sections, the maximum average stream temperature should ordinarily not occur until at least the last heating section. Furthermore, the heating zone should typically extend to as close to the exit of the furnace as possible. This is counter to conventional thought, which is to typically maintain the exit portion of the furnace at a low temperature to avoid having to seal the furnace outlet at a high temperature. Such cooling of the exit portion of the furnace, however, significantly promotes thermophoretic losses. Furthermore, the potential for operating problems that could result in thermophoretic losses at the back end of the furnace are reduced with the very short residence times in the furnace for the present invention, as discussed previously.

Typically, it would be desirable to instantaneously cool the aerosol upon exiting the furnace. This is not possible. It is possible, however, to make the residence time between the furnace outlet and the cooling unit as short as possible. Furthermore, it is desirable to insulate the aerosol conduit occurring between the furnace exit and the cooling unit entrance. Even more preferred is to insulate that conduit and, even more preferably, to also heat that conduit so that the wall temperature of that conduit is at least as high as the average stream temperature of the aerosol stream. Furthermore, it is desirable that the cooling unit operate in a manner such that the aerosol is quickly cooled in a manner to prevent thermophoretic losses during cooling. The quench cooler, described previously, is very effective for cooling with low losses. Furthermore, to keep the potential for thermophoretic losses very low, it is preferred that the residence time of the aerosol stream between attaining the maximum stream temperature in the furnace and a point at which the aerosol has been cooled to an average stream temperature below about 200°C is shorter than about 2 seconds, more preferably shorter than about 1 second, and even more preferably shorter than about 0.5 second and most preferably shorter than about 0.1 second. In most instances, the maximum average stream temperature attained in the furnace will

be greater than about 700°C. Furthermore, the total residence time from the beginning of the heating zone in the furnace to a point at which the average stream temperature is at a temperature below about 200°C should typically be shorter than about 5 seconds, preferably shorter than about 3 seconds, more preferably shorter than about 2 seconds, and most preferably shorter than about 1 second.

Another part of the process with significant potential for thermophoretic losses is after particle cooling until the particles are finally collected. Proper particle collection is very important to reducing losses within the system. The potential for thermophoretic losses is significant following particle cooling because the aerosol stream is still at an elevated temperature to prevent detrimental condensation of water in the aerosol stream. Therefore, cooler surfaces of particle collection equipment can result in significant thermophoretic losses.

To reduce the potential for thermophoretic losses before the particles are finally collected, it is important that the transition between the cooling unit and particle collection be as short as possible. Preferably, the output from the quench cooler is immediately sent to a particle separator, such as a filter unit or a cyclone. In that regard, the total residence time of the aerosol between attaining the maximum average stream temperature in the furnace and the final collection of the particles is preferably shorter than about 2 seconds, more preferably shorter than about 1 second, still more preferably shorter than about 0.5 second and most preferably shorter than about 0.1 second. Furthermore, the residence time between the beginning of the heating zone in the furnace and final collection of the particles is preferably shorter than about 6 seconds, more preferably shorter than about 3 seconds, even more preferably shorter than about 2 seconds, and most preferably shorter than about 1 second. Furthermore, the potential for thermophoretic losses may further be reduced by insulating the conduit section between the cooling unit and the particle collector and, even more preferably, by also insulating around the filter, when a filter is

used for particle collection. The potential for losses may be reduced even further by heating of the conduit section between the cooling unit and the particle collection equipment, so that the internal equipment surfaces are at least slightly warmer than the aerosol stream average stream temperature. Furthermore, when a filter is used for particle collection, the filter could be heated. For example, insulation could be wrapped around a filter unit, with electric heating inside of the insulating layer to maintain the walls of the filter unit at a desired elevated temperature higher than the temperature of filter elements in the filter unit, thereby reducing thermophoretic particle losses to walls of the filter unit.

Even with careful operation to reduce thermophoretic losses, some losses will still occur. For example, some particles will inevitably be lost to walls of particle collection equipment, such as the walls of a cyclone or filter housing. One way to reduce these losses, and correspondingly increase product yield, is to periodically wash the interior of the particle collection equipment to remove particles adhering to the sides. In most cases, the wash fluid will be water, unless water would have a detrimental effect on one of the components of the particles. For example, the particle collection equipment could include parallel collection paths. One path could be used for active particle collection while the other is being washed. The wash could include an automatic or manual flush without disconnecting the equipment. Alternatively, the equipment to be washed could be disconnected to permit access to the interior of the equipment for a thorough wash. As an alternative to having parallel collection paths, the process could simply be shut down occasionally to permit disconnection of the equipment for washing. The removed equipment could be replaced with a clean piece of equipment and the process could then be resumed while the disconnected equipment is being washed.

For example, a cyclone or filter unit could periodically be disconnected and particles adhering to interior walls could be removed by a water wash. The particles could then be dried in a low temperature dryer, typically at a temperature of lower than about 50°C.

Another area for potential losses in the system, and for the occurrence of potential operating problems, is between the outlet of the aerosol generator and the inlet of the furnace. Losses here are not due to thermophoresis, but rather to liquid coming out of the aerosol and impinging and collecting on conduit and equipment surfaces. Although this loss is undesirable from a material yield standpoint, the loss may be even more detrimental to other aspects of the process. For example, water collecting on surfaces may release large droplets that can lead to large particles that detrimentally contaminate the particulate product. Furthermore, if accumulated liquid reaches the furnace, the liquid can cause excessive temperature gradients within the furnace tube, which can cause furnace tube failure, especially for ceramic tubes.

One way to reduce the potential for undesirable liquid buildup in the system is to provide adequate drains, as previously described. In that regard, it is preferred that a drain be placed as close as possible to the furnace inlet to prevent liquid accumulations from reaching the furnace. The drain should be placed, however, far enough in advance of the furnace inlet such that the stream temperature is lower than about 80°C at the drain location.

Another way to reduce the potential for undesirable liquid buildup is for the conduit between the aerosol generator outlet and the furnace inlet to be of a substantially constant cross sectional area and configuration. Preferably, the conduit beginning with the aerosol generator outlet, passing through the furnace and continuing to at least the cooling unit inlet is of a substantially constant cross sectional area and geometry.

Another way to reduce the potential for undesirable buildup is to heat at least a portion, and preferably the entire length, of the conduit between the aerosol generator and the inlet to the furnace. For example, the conduit could be wrapped with a heating tape to maintain the inside walls of the conduit at a temperature higher than the temperature of the aerosol. The aerosol would then tend to concentrate toward the center of the conduit due to thermophoresis.

Fewer aerosol droplets would, therefore, be likely to impinge on conduit walls or other surfaces making the transition to the furnace.

Another way to reduce the potential for undesirable liquid buildup is to introduce a dry gas into the aerosol between the aerosol generator and the furnace. Referring now to Fig. 36, one embodiment of the process is shown for adding a dry gas 118 to the aerosol 108 before the furnace 110. Addition of the dry gas 118 causes vaporization of at least a part of the moisture in the aerosol 108, and preferably substantially all of the moisture in the aerosol 108, to form a dried aerosol 119, which is then introduced into the furnace 110.

The dry gas 118 will most often be dry air, although in some instances it may be desirable to use dry nitrogen gas or some other dry gas. If a sufficient quantity of the dry gas 118 is used, the droplets of the aerosol 108 are substantially completely dried to beneficially form dried precursor particles in aerosol form for introduction into the furnace 110, where the precursor particles are then pyrolyzed to make a desired particulate product. Also, the use of the dry gas 118 typically will reduce the potential for contact between droplets of the aerosol and the conduit wall, especially in the critical area in the vicinity of the inlet to the furnace 110. In that regard, a preferred method for introducing the dry gas 118 into the aerosol 108 is from a radial direction into the aerosol 108. For example, equipment of substantially the same design as the quench cooler, described previously with reference to Figs. 29-31, could be used, with the aerosol 108 flowing through the interior flow path of the apparatus and the dry gas 118 being introduced through perforated wall of the perforated conduit. An alternative to using the dry gas 118 to dry the aerosol 108 would be to use a low temperature thermal preheater/dryer prior to the furnace 110 to dry the aerosol 108 prior to introduction into the furnace 110. This alternative is not, however, preferred.

Still another way to reduce the potential for losses due to liquid accumulation is to operate the process with equipment configurations such that the aerosol stream flows in a vertical direction from the aerosol generator to and

through the furnace. For smaller-size particles, those smaller than about 1.5 μm , this vertical flow should, preferably, be vertically upward. For larger-size particles, such as those larger than about 1.5 μm , the vertical flow is preferably vertically downward.

5 Furthermore, with the process of the present invention, the potential for system losses is significantly reduced because the total system retention time from the outlet of the generator until collection of the particles is typically shorter than about 10 seconds, preferably shorter than about 7 seconds, more preferably shorter than about 5 seconds and most preferably shorter than about
10 3 seconds.

Many phosphors can be difficult to produce using conventional methods such that the powders have the desirable physical, chemical and luminescent characteristics. Many phosphor compounds can be difficult to produce even using a standard spray pyrolysis technique.

15 These compounds can advantageously be produced according to the present invention using a process referred to as spray-conversion. Spray-conversion is a process wherein a spray pyrolysis technique, as is described above, is used to produce an intermediate particulate product that is capable of being subsequently converted to a particulate phosphor having the desirable
20 properties. The intermediate product advantageously has many of the desirable morphological properties discussed hereinbelow, such as a small particle size and a narrow particle size distribution. Further, the intermediate particles individually include the chemical components needed to form the phosphor compound. This is distinguished from individual particles of different
25 composition that are co-mixed. The intermediate precursor particles of the present invention can advantageously be converted to the corresponding phosphor at a reduced temperature with reduced agglomeration.

As is discussed above, precursor materials including water-soluble precursors, such as nitrate salts and insoluble precursors such as colloidal silica,
30 are placed into a liquid solution, atomized and are converted at a relatively low

temperature, such as less than about 1000°C, to intermediate precursor particles that typically include low crystallinity oxide phase(s). The intermediate precursor particles have a small size and, preferably, a narrow particle size distribution, as is described in more detail below. The intermediate precursor particles are then converted by further treatment, such as by heat treating at an elevated temperature, to form a phosphor compound having high crystallinity and good luminescence characteristics. The resulting powder advantageously does not require any further milling to reduce particle size or reduce hard agglomerates since the intermediate particles have the desired size and hard agglomeration is avoided during the subsequent heat treatment. The resulting end product is a highly crystalline phosphor powder having the desirable morphological and luminescent properties. The average particle size and morphological characteristics are determined by the characteristics of the intermediate product.

Thus, the precursors can preferably be spray-converted at a temperature of at least about 700°C, such as from about 750°C to 950°C, to form a homogeneous admixture, typically having low crystallinity. The intermediate particles can then be heat treated at a temperature of, for example, 1100°C to 1600°C, to form phosphor particles having high crystallinity and good luminescent properties.

The present invention is applicable to a wide variety of phosphors, including cathodoluminescent (CL), electroluminescent (EL), photoluminescent (PL) and x-ray (XR) phosphors. Phosphors typically include a matrix compound, referred to as a host material, and the phosphor also includes one or more dopants, referred to as activator ions, to emit a specific color or to enhance the luminescence characteristics. Examples of host materials to which the present invention is applicable include yttrium oxides, yttrium oxysulfides, gadolinium oxysulfides, sulfides such as zinc sulfide, calcium sulfide and strontium sulfide, silicates such as zinc silicate and yttrium silicate, thiogallates such as strontium thiogallate and calcium thiogallate, gallates such as zinc gallate, calcium gallate and strontium gallate, aluminates such as barium aluminate or barium

magnesium aluminate (BAM) and borates such as yttrium-gadolinium borate. Specific examples of phosphor compounds according to the present invention are listed in Table I.

Table I Examples of Phosphor Compounds

HOST MATERIAL	ACTIVATOR ION (S)	EXCITATION MECHANISM
Y_2O_3	Eu, rare earths, Tb	CL, PL
Y_2O_2S	Eu, Tb	CL
ZnS	Au, Al, Cu, Ag, Cl, Mn	CL, EL
$SrGa_2S_4$	Eu, Ce	CL, EL
$Y_3Al_5O_{12}$	Tb, Cr	CL
$Y_3(Ga,Al)_5O_{12}$	Tb, Cr	CL
Zn_2SiO_4	Mn	CL, PL
Y_2SiO_5	Tb, Ce	CL
CaS	Eu, Ce	EL
SrS	Eu, Ce	EL
$CaGa_2S_4$	Eu, Ce	EL
$ZnGa_2O_4$	Mn, Cr	EL
$CaGa_2O_4$	Eu, Ce	EL
$SrGa_2O_4$	Eu, Ce	EL
Ga_2O_3	Dy, Eu	EL
$Ca_3Ga_2O_6$	Eu	EL
Zn_2GeO_4	Mn	EL
$Zn_2(Ge,Si)O_4$	Mn	EL
$(Y,Gd)BO_3$	Eu	PL
$BaMgAl_{10}O_{17}$	Mn, Eu	PL
$BaAl_xO_y$	Mn	PL
Gd_2O_2S	Tb	XR
$(Y,Gd)_2SiO_5$	Tb	XR

According to the present invention, the liquid feed preferably includes the chemical components that will form the phosphor particles. For example, the liquid feed can include a solution of metal salts such as nitrates, chlorides, sulfates, hydroxides or oxalates of the phosphor components. In addition, the liquid feed can include particulate precursors such as SiO_2 or Al_2O_3 . Particulate SiO_2 is a preferred precursor for the silicate compounds, and it may be advantageous to provide an excess of silica to obtain phosphor powders having a highly crystalline structure.

A preferred precursor are the nitrates, such as yttrium nitrate, $\text{Y}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$, for the production of yttrium oxide phosphor particles. Nitrates are typically highly soluble in water and the solutions maintain a low viscosity, even at high concentrations. A typical reaction mechanism for the formation of yttrium oxide would be:



It has been found that the precursor solution for the production of yttrium oxide should preferably include from about 4 to about 6 weight percent of the precursors.

Similarly, yttrium oxysulfides can advantageously be produced from a similar precursor system wherein the intermediate precursor particles form low crystallinity oxides which are subsequently heated in a sulfide-containing atmosphere, such as in H_2S .

Preferred precursors for the production of silicates, such as $\text{Zn}_2\text{SiO}_4\text{:Mn}$, include metals salts, particularly nitrate salts, for the Zn and the Mn. For the silica component, it is preferred to use dispersed particulate silica. It has also been found that for the production of zinc silicate that the precursor solution should include an excess of silica. It has been found that, for example, a preferred precursor solution includes an excess of about 10 atomic percent silica. Yttrium silicate can be produced in a similar fashion wherein yttrium nitrate is substituted for zinc nitrate.

Zinc sulfide can be produced from a precursor solution including thiourea and zinc nitrate. $Y_5(Ga,Al)_5O_{12}$ can advantageously be produced from a solution comprising all metal salts or can include particulate alumina.

Thiogallates such as $SrGa_2S_4$ can be produced from salts such as nitrate salts and pyrolyzed in air to produce intermediate precursor particles that are composed mainly of low crystallinity oxides such as $SrGa_2O_4$. The intermediate precursor particles are then treated in a sulfur-containing atmosphere to convert the oxides to the sulfide phase.

Oxide based phosphors can typically be produced from simple metal salts. For example, $SrGa_xO_y$ can be produced from strontium carbonate and gallium nitrate and $CaGa_xO_y$ can be produced from calcium carbonate and gallium nitrate.

For the production of europium doped barium magnesium aluminate (BAM:Eu) phosphors according to the present invention, the liquid feed can include a solution of metal salts such as nitrates, chlorides, sulfates, hydroxides or oxalates of the phosphor components. According to a preferred embodiment, BAM:Eu is formed from a precursor solution comprising barium nitrate, magnesium nitrate and fumed (particulate) alumina for the BAM host material, as well as europium nitrate to provide the Eu dopant ion or manganese nitrate to provide the Mn dopant ion. It has been found that aluminum nitrate as a precursor to aluminum oxide is not preferred, but alumina, or a similar aluminum compound such as boehmite is preferred. It has also been unexpectedly found that the precursor solution should include an amount of aluminum that is in excess of the stoichiometric amount required to form BAM. In one preferred embodiment, at least about 20 atomic percent excess alumina is added to the precursor solution. Barium aluminate can be formed in a similar manner. About 5 to 10 weight percent of the precursor in the solution is preferred.

$(Y,Gd)BO_3$ can be formed from a precursor solution comprising yttrium and gadolinium salts, such as the nitrate salts, as well as boric acid. Preferably the precursor solution includes an excess of boric acid, which advantageously

produces highly crystalline (Y,Gd)BO₃ powders. It is preferred to use a precursor concentration of from about 5 to about 10 weight percent, more preferably from about 7 to 8 weight percent, based on the equivalent weight of the phosphor compound.

5 The foregoing precursor solutions/dispersions are preferably not saturated with the precursor to avoid precipitate formation in the liquid. The solution preferably includes, for example, sufficient precursor to yield from about 1 to 50 weight percent, such as from about 1 to 15 weight percent, of the phosphor compound. That is, the solution concentrations are measured based on the
10 equivalent weight percent of the phosphor product. The final particle size of the phosphor particles is also influenced by the precursor concentration. Generally, lower precursor concentrations in the liquid feed will produce particles having a smaller average size.

 As is discussed above, the liquid feed preferably includes the dopant
15 (activator ion) precursor, typically a soluble metal salt such as a nitrate. The relative concentrations of the precursors can be adjusted to vary the concentration of the activator ion in the host material.

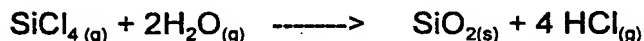
 Preferably, the solvent is aqueous-based for ease of operation, although other solvents, such as toluene, may be desirable. However, the use of organic
20 solvents can lead to undesirable carbon contamination in the phosphor particles. Further, the pH of the aqueous-based solutions can advantageously be adjusted to alter the solubility characteristics of the precursors in the solution.

 In addition to the foregoing, the liquid feed may also include other additives that contribute to the formation of the particles. For example, a fluxing
25 agent can be added to the solution to increase the crystallinity and/or density of the particles. The addition of urea to metal salt solutions, such as a metal nitrate, can increase the density of particles produced from the solution. In one embodiment, up to about 1 mole equivalent urea is added to the precursor solution, as measured against the moles of phosphor compound in the metal salt
30 solution. Small amounts, e.g. less than 1 weight percent, of boric acid added to

a precursor solution can also enhance crystallinity without substantially altering the composition of the powder. Further, if the particles are to be coated phosphor particles, as is discussed in more detail below, a soluble precursor to both the phosphor compound and the coating can be used in the precursor solution wherein the coating precursor is an involatile or volatile species.

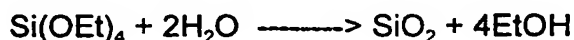
For producing the phosphor particles of the present invention, the carrier gas may comprise any gaseous medium in which droplets produced from the liquid feed may be dispersed in aerosol form. Also, the carrier gas may be inert, in that the carrier gas does not participate in formation of the phosphor particles. Alternatively, the carrier gas may have one or more active component(s) that contribute to formation of the phosphor particles. In that regard, the carrier gas may include one or more reactive components that react in the furnace to contribute to formation of the phosphor particles. In many applications for the spray-conversion of phosphor particles according to the present invention, air will be a satisfactory carrier gas for providing oxygen. In other instances, a relatively inert gas such as nitrogen may be required.

When the phosphors of the present invention are coated phosphors, such as phosphors coated with a metal oxide, precursors to metal oxide coatings can be selected from volatile metal acetates, chlorides, alkoxides or halides. Such precursors are known to react at high temperatures to form the corresponding metal oxides and eliminate supporting ligands or ions. For example, SiCl_4 can be used as a precursor to SiO_2 coatings when water vapor is present:



SiCl_4 also is highly volatile and is a liquid at room temperature, which makes transport into the reactor more controllable. Aluminum trichloride (AlCl_3) can be used in a similar manner to form an alumina coating.

Metal alkoxides can be used to produce metal oxide films by hydrolysis. The water molecules react with the alkoxide M-O bond resulting in clean elimination of the corresponding alcohol with the formation of M-O-M bonds:



Most metal alkoxides have a reasonably high vapor pressure and are therefore well suited as coating precursors.

Metal acetates are also useful as coating precursors since they readily decompose upon thermal activation by acetic anhydride elimination:



Metal acetates are advantageous as coating precursors since they are water stable and are reasonably inexpensive.

Coatings can be generated on the particle surface by a number of different mechanisms. One or more precursors can vaporize and fuse to the hot phosphor particle surface and thermally react resulting in the formation of a thin-film coating by chemical vapor deposition (CVD). Preferred coatings deposited by CVD include metal oxides and elemental metals. Further, the coating can be formed by physical vapor deposition (PVD) wherein a coating material physically deposits on the surface of the particles. Preferred coatings deposited by PVD include organic materials and elemental metals. Alternatively, the gaseous precursor can react in the gas phase forming small particles, for example less than about 5 nanometers in size, which then diffuse to the larger particle surface and sinter onto the surface, thus forming a coating. This method is referred to as gas-to-particle conversion (GPC). Whether such coating reactions occur by CVD, PVD or GPC is dependent on the reactor conditions such as precursor partial pressure, water partial pressure and the concentration of particles in the gas stream. Another possible surface coating method is surface conversion of the surface of the particle by reaction with a vapor phase reactant to convert the surface of the particles to a different material than that originally contained in the particles.

25 In addition, a volatile coating material such as PbO , MoO_3 or V_2O_5 can be introduced into the reactor such that the coating deposits on the particle by condensation. Highly volatile metals, such as silver, can also be deposited by condensation. Further, the phosphor powders can be coated using other techniques. For example, a soluble precursor to both the phosphor powder and

30

the coating can be used in the precursor solution wherein the coating precursor is involatile (e.g. $\text{Al}(\text{NO}_3)_3$) or volatile (e.g. $\text{Sn}(\text{OAc})_4$ where OAc is acetate). In another method, a colloidal precursor and a soluble phosphor precursor can be used to form a particulate colloidal coating on the phosphor.

5 The phosphor powders produced by the foregoing method may be fully converted to the crystalline phosphor compound during the pyrolyzation step. However, it is preferred that the powders are spray-converted to an intermediate form. It is then necessary to heat the spray-converted intermediate precursor particles to convert the intermediate precursor powder to a luminescent
10 phosphor compound and to increase the crystallinity (average crystallite size) of the powder. Thus, the powders can be heat-treated for an amount of time and in a preselected environment, as is discussed above. Increased crystallinity can advantageously yield an increased brightness and efficiency of the phosphor particles. If such heat treating steps are performed, the heat treating
15 temperature and time should be selected to minimize the amount of interparticle sintering. Table II illustrates examples of preferred phosphor powders according to the present invention with the preferred conversion and heat treatment conditions.

TABLE II Examples of Spray-Converted Phosphor Compounds

Host Material	Conversion Temperature (pyrolyzation)	Carrier Gas	Heat Treatment Temperature	Heat Treatment Gas
Y_2O_3	850-1000°C	Air	1350-1500°C	Air
Zn_2SiO_4	850-1000°C	Air	1100-1300°C	Air
ZnS	300-900°C	N_2	500-900°C	H_2S/N_2
$SrGa_2S_4$	700-900°C	Air	800-1100°C	H_2S/N_2
$SrGa_xO_y$	700-900°C	Air	1100-1450°C	Air
$CaGa_xO_y$	700-900°C	Air	1100-1450°C	Air
$BaMgAl_{10}O_{17}$	750-950°C	Air	1200-1700°C	H_2/N_2
$(Y,Gd)BO_3$	900-950°C	Air	1100-1400°C	Air
$BaAl_xO_y$	750-950°C	Air	1200-1700°C	Air
$(Y,Gd)_2SiO_5$	900-950°C	Air	1100-1200°C	Air

The heat treatment time is preferably not more than about 2 hours and can be as little as about 1 minute. To reduce agglomeration, the intermediate particles are preferably heat treated under sufficient agitation to minimize the agglomeration of the particles. One preferred method for agitating during heat treatment is to heat treat the powders in a rotary kiln, wherein the powders are constantly moving through a tubular furnace that is rotating on its major axis.

Further, the crystallinity of the phosphors can be increased by using a fluxing agent, either in the precursor solution or in a post-formation annealing step. A fluxing agent is a reagent, which improves the crystallinity of the material when the reagent and the material are heated together, as compared to heating the material to the same temperature and for the same amount of time in the absence of the fluxing agent. The fluxing agents typically cause a eutectic to form which leads to a liquid phase at the grain boundaries, increasing the diffusion coefficient. The fluxing agent, for example alkali metal halides such as NaCl or KCl or an organic compound such as urea ($CO(NH_2)_2$), can be added

to the precursor solution where it improves the crystallinity and/or density of the particles during their subsequent formation. Alternatively, the fluxing agent can be contacted with the phosphor powder batches after they have been collected. Upon heat treatment, the fluxing agent improves the crystallinity of the phosphor powder, and therefore improves other properties such as the brightness of the phosphor powder. Also, in the case of composite particles, the particles may be annealed for a sufficient time to permit redistribution within the particles of different material phases.

The phosphor host material can be doped with an activator ion, typically in an amount of up to about 30 atomic percent, such as from about 0.02 to about 20 atomic percent. The preferred concentration of the activator ion will vary depending on the composition and the application of the phosphor, as is discussed in more detail below. The activator ion should also be in the proper oxidation state.

One advantage of the present invention is that the activator ion is homogeneously distributed throughout the host material. Phosphor powders prepared by solid-state methods do not give uniform concentration of the activator ion, particularly in small particles, and solution routes also do not give homogenous distribution of the activator ion due to different rates of precipitation.

The powder characteristics that are preferred will depend upon the particular application of the phosphor powders. Nonetheless, it can be generally stated for most applications that the powders should have one or more of: small particle size; narrow size distribution; spherical morphology; high crystallinity; controlled porosity and homogenous dopant distribution of activator ion throughout the host material. The efficiency of the phosphor, defined as the overall conversion of excitation energy to visible photons, should be high.

According to the present invention, the phosphor powder includes particles having a small average particle size. Although the preferred average size of the phosphor particles will vary according to the application of the

phosphor powder, the average particle size of the phosphor particles is at least about 0.1 μm and more preferably is at least about 0.3 μm . Further, the average particle size is not greater than about 20 μm . For most applications, the average particle size is preferably not greater than about 10 μm , such as not greater than about 5 μm and more preferably is not greater than about 3 μm , such as from about 0.3 μm to about 3 μm . As used herein, the average particle size is the volume average particle size.

According to the present invention, the powder batch of phosphor particles can also have a narrow particle size distribution, such that the majority of particles are substantially the same size. Preferably, at least about 70 volume percent of the particles and more preferably at least about 80 volume percent of the particles are not larger than twice the average particle size. Thus, when the average particle size is about 2 μm , it is preferred that at least about 70 volume percent of the particles are not larger than 4 μm and it is more preferred that at least about 80 volume percent of the particles are not larger than 4 μm . Further, it is preferred that at least about 70 volume percent of the particles, and more preferably at least about 80 volume percent of the particles, are not larger than about 1.5 times the average particle size. Thus, when the average particle size is about 2 μm , it is preferred that at least about 70 volume percent of the particles are not larger than about 3 μm and it is more preferred that at least about 80 volume percent of the particles are not larger than about 3 μm .

Powders produced by the processes described herein, particularly those that have experienced a post treatment step, generally exit as soft agglomerates of primary spherical particles. It is well known to those in the art that micrometer-sized particles often form soft agglomerates as a result of their relatively high surface energy, as compared to larger particles. It is also known to those skilled in the art that such soft agglomerates may be dispersed easily by treatments such as exposure to ultrasound in a liquid medium or sieving. The average particle size and particle size distributions described herein are measured by mixing samples of the powders in a medium such as water with a surfactant and

a short exposure to ultrasound through either an ultrasonic bath or horn. The ultrasonic treatment supplies sufficient energy to disperse the soft agglomerates into primary spherical particles. The primary particle size and size distribution is then measured by light scattering in a Microtrac instrument. This provides a good measure of the useful dispersion characteristics of the powder because this simulates the dispersion of the particles in a liquid medium such as a paste or slurry that is used to deposit the particles in a device, such as an cathodoluminescent display devices. Thus, the references to particle size herein refer to the primary particle size, such as after lightly dispersing the soft agglomerates of particles.

Further, it is advantageous according to the present invention that the foregoing description of the average size and size distribution of the phosphor particles also applies to the intermediate precursor particles that are produced during the pyrolyzation step. That is, the size and size distribution of the particles changes very little, if at all, during the heat treatment step after pyrolyzation. The morphological properties of the final phosphor powder are substantially controlled by the properties of the intermediate precursor particles.

The phosphor particles of the present invention are comprised of a number of crystallites. According to the present invention, the phosphor particles are highly crystalline and it is preferred that the average crystallite size approaches the average particle size such that the particles are composed of only a few large crystals. The average crystallite size of the particles is preferably at least about 25 nanometers, more preferably is at least about 40 nanometers, even more preferably is at least about 60 nanometers and most preferably is at least about 80 nanometers. In one embodiment, the average crystallite size is at least about 100 nanometers. As it relates to particle size, the average crystallite size is preferably at least about 10 percent, more preferably at least about 20 percent and most preferably is at least about 30 percent of the average particle size. Such highly crystalline phosphors are believed to have

increased luminescent efficiency and brightness as compared to phosphor particles having smaller crystallites.

The phosphor particles of the present invention advantageously have a high degree of purity, that is, a low level of impurities. Impurities are those materials that are not intended in the final product and that negatively affect the properties of the phosphor. Thus, an activator ion is not considered an impurity. The level of impurities in the phosphor powders of the present invention is preferably not greater than about 1 atomic percent, more preferably is not greater than about 0.1 atomic percent and even more preferably is not greater than about 0.01 atomic percent. It will be appreciated that some of the phosphor particles according to the present invention will include a second-phase that does not hinder the properties of the phosphor. For example, the Zn_2SiO_4 phosphor particles of the present invention are advantageously produced using a slight excess of silica, and therefore the particle will include silica crystallites dispersed throughout a Zn_2SiO_4 matrix. Such a second phase is not considered an impurity.

The formation of hollow particles is common in spray pyrolysis and can occur in spray conversion. Hollow phosphor particles may be detrimental in a number of applications of phosphor powders. In the present invention, it has been found that the formation of hollow or dense particles can be controlled through a combination of the control over pyrolysis temperature, residence time, precursor selection and solution concentration. For example, with constant pyrolysis temperature and residence time, the morphology of some powders show the presence of progressively more hollow particles as the solution concentration is raised above 5 weight percent.

The phosphor particles of the present invention are also substantially spherical in shape. That is, the particles are not jagged or irregular in shape. Spherical particles are particularly advantageous because they are able to disperse and coat a device more uniformly with a reduced average thickness. Although the particles are substantially spherical, the particles may become

faceted as the crystallite size increases while maintaining a substantially spherical morphology.

In addition, the phosphor particles according to the present invention advantageously have a low surface area. The particles are substantially spherical, which reduces the total surface area for a given mass of powder. Further, the elimination of larger particles from the powder batches eliminates the porosity that is associated with open pores on the surface of such larger particles. Due to the elimination of the large particles, the powder advantageously has a lower surface area. Surface area is typically measured using a BET nitrogen adsorption method, which is indicative of the surface area of the powder, including the surface area of accessible surface pores on the surface of the powder. For a given particle size distribution, a lower value of a surface area per unit mass of powder indicates solid or non-porous particles. Decreased surface area reduces the susceptibility of the phosphor powders to adverse surface reactions, such as degradation from moisture. This characteristic can advantageously extend the useful life of the phosphor powders.

The surfaces of the phosphor particles according to the present invention are typically smooth and clean with a minimal deposition of contaminants on the particle surface. For example, the outer surfaces are not contaminated with surfactants, as is often the case with particles produced by liquid precipitation routes. Since the particles do not require milling, the particle surfaces do not include major surface defects that typically result from milling and can decrease the brightness of the powders.

In addition, the powder batches of phosphor particles according to the present invention are substantially unagglomerated, that is, they include substantially no hard agglomerates of particles. Hard agglomerates are physically coalesced lumps of two or more particles that behave as one large particle. Hard agglomerates are disadvantageous in most applications of phosphor powders. It is preferred that no more than about 1 weight percent of

the phosphor particles in the powder batch of the present invention are in the form of hard agglomerates. More preferably, no more than about 0.5 weight percent of the particles are in the form of hard agglomerates and even more preferably no more than about 0.1 weight percent of the particles are in the form of hard agglomerates. In the event that hard agglomerates of the powder do form, they can optionally be broken up, such as by jet-milling the powder.

According to one embodiment of the present invention, the phosphor particles are composite phosphor particles, wherein the individual particles include at least a first phosphor phase and at least a second phase associated with the phosphor phase. Preferably, the second phase is dispersed throughout the matrix of the first phosphor phase. The second phase can be a different phosphor compound or can be a non-phosphor compound. Such composites can advantageously permit the use of phosphor compounds in devices that would otherwise be unusable. Further, combinations of different phosphor compounds within one particle can produce emission of a selected color. The emission of the two phosphor compounds could combine to approximate white light. Other advantages can also be realized. For example, in cathodoluminescent applications, the matrix material can accelerate the impinging electrons to enhance the luminescence of the particles.

According to another embodiment of the present invention, the phosphor particles are surface modified or coated phosphor particles that include a particulate coating (Fig. 35d) or non-particulate (film) coating (Fig. 35a) that substantially encapsulates an outer surface of the particles. The coating can be a conductive metallic material, including a metal, a non-metallic compound including a semiconductor or an organic compound.

Coatings are often desirable to reduce degradation of the phosphor powder due to moisture or other influences. The thin, uniform coatings according to the present invention will advantageously permit use of the phosphor powders under corrosive conditions. Coatings also create a diffusion barrier such that activator ions cannot transfer from one particle to another,

thereby altering the luminescence characteristics. Coatings can also control the surface energy levels of the particles.

The coating can be a metal, metal oxide or other inorganic compound such as a metal sulfide, or can be an organic compound. For example, a metal oxide coating can advantageously be used, such as a metal oxide selected from the group consisting of SiO_2 , MgO , Al_2O_3 , ZnO , SnO_2 or In_2O_3 . Particularly preferred are SiO_2 and Al_2O_3 coatings. Semiconductive oxide coatings such as SnO_2 or In_2O_3 can be advantageous in some applications. In addition, phosphate coatings, such as zirconium phosphate or aluminum phosphate, can also be advantageous for use in some applications.

The coatings should be relatively thin and uniform. The coating should encapsulate the entire particle, but be sufficiently thin such that the coating does not substantially interfere with light transmission. Preferably, the coating has an average thickness of not greater than about 200 nanometers, more preferably not greater than about 100 nanometers, and even more preferably not greater than about 50 nanometers. The coating preferably completely encapsulates the phosphor particle and therefore should have an average thickness of at least about 2 nanometers, more preferably at least about 5 nanometers. In one embodiment, the coating has an average thickness of from about 2 to 50 nanometers, such as from about 2 to 10 nanometers. Further, the particles can include more than one coating substantially encapsulating the particles to achieve the desired properties.

The coating, either particulate or non-particulate, can also include a pigment or other material that alters the light characteristics of the phosphor. Red pigments can include compounds such as the iron oxides (Fe_2O_3), cadmium sulfide compounds (CdS) or mercury sulfide compounds (HgS). Green or blue pigments include cobalt oxide (CoO), cobalt aluminate (CoAl_2O_4) or zinc oxide (ZnO). Pigment coatings are capable of absorbing selected wavelengths of light exiting the phosphor, thereby acting as a filter to improve the color contrast and purity. Further, a dielectric coating, either organic or inorganic, can be used to

achieve the appropriate surface charge characteristics to carry out deposition processes such as electrostatic deposition.

In addition, the phosphor particles can be coated with an organic compound such as PMMA (polymethylmethacrylate), polystyrene or similar organic compounds, including surfactants that aid in the dispersion and/or suspension of the particles in a flowable medium. The organic coating is preferably not greater than about 100 nanometers thick and is substantially dense and continuous about particle. The organic coatings can advantageously prevent corrosion of the phosphor particles and also can improve the dispersion characteristics of the particles in a paste or other flowable medium.

The coating can also be comprised of one or more monolayer coatings, such as from about 1 to 3 monolayer coatings. A monolayer coating is formed by the reaction of an organic or an inorganic molecule with the surface of the phosphor particles to form a coating layer that is essentially one molecular layer thick. In particular, the formation of a monolayer coating by reaction of the surface of the phosphor powder with a functionalized organo silane such as halo- or amino-silanes, for example hexamethyldisilazane or trimethylsilylchloride, can be used to modify and control the hydrophobicity and hydrophilicity of the phosphor powders. Monolayer coatings of metal oxides (e.g. ZnO or SiO₂) or metal sulfides (e.g. Cu₂S) can be formed as monolayer coatings. Monolayer coatings can allow for greater control over the dispersion characteristics of the phosphor powder in a wide variety of paste compositions and other flowable mediums.

The monolayer coatings may also be applied to phosphor powders that have already been coated with an organic or inorganic coating, thus providing better control over the corrosion characteristics (through the use of a thicker coating) as well as dispersibility (through the use of a monolayer coating) of the phosphor powder.

As a direct result of the foregoing powder characteristics, the phosphor powders of the present invention have many unique and advantageous properties that are not found in phosphor powders known heretofore.

5 The phosphor powders of the present invention have a high efficiency, sometimes referred to as quantum efficiency. Efficiency is the overall conversion of excitation energy to visible photons emitted. The high efficiency of the phosphor powders according to the present invention is believed to be due to the high crystallinity and homogenous distribution of activator ion in the host material as well as a substantially defect-free particle surface.

10 The phosphor powders also have well-controlled color characteristics, sometimes referred to as emission spectrum characteristics or chromaticity. This important property is due to the ability to precisely control the composition of the host material, the homogenous distribution of the activator ion and the high purity of the powders.

15 The phosphor powders also have improved decay time, also referred to as persistence. Persistence is the amount of time for the light emission to decay to 10% of its peak brightness. In display applications, phosphors with long decay times can result in blurred images when the image moves across the display. The improved (reduced) decay time of the phosphor powders of the present invention is believed to be primarily due to the homogenous distribution of activator ion in the host material.

20 The phosphor powders also can have an improved brightness over prior art phosphor powders. That is, under a given application of energy (electrons, photons, electric field or x-rays), the phosphor powders of the present invention produce more light.

25 Thus, the phosphor powders of the present invention have a unique combination of properties that are not found in conventional phosphor powders. The powders can advantageously be used to form a number of intermediate products, for example liquid mediums such as pastes or slurries, and can be incorporated into a number of devices, wherein the devices will have significantly

improved performance resulting directly from the characteristics of the phosphor powders of the present invention.

Phosphor powders are typically deposited onto device surfaces or substrates by a number of different deposition methods which involve the direct deposition of the dry powder such as dusting, electrophotographic or electrostatic precipitation, while other deposition methods involve liquid vehicles such as ink jet printing, liquid delivery from a syringe, micro-pens, toner, slurry deposition, paste-based methods and electrophoresis. In all these deposition methods, the powders described in the present invention show a number of distinct advantages over the phosphor powders produced by other methods. For example, small, spherical, narrow size distribution phosphor particles are more easily dispersed in liquid vehicles, they remain dispersed for a longer period and allow printing of smoother and finer features compared to powder made by alternative methods.

For many applications, phosphor powders are often dispersed within a paste, which is then applied, to a surface to obtain a phosphorescent layer. The powders of the present invention offer many advantages when dispersed in such a paste. For example, the powders will disperse better than non-spherical powders of wide size distribution and can therefore produce thinner and more uniform layers with a reduced lump count. Such a thick film paste will produce a brighter display due to the increased powder density in the phosphor layer. The number of processing steps can also be advantageously reduced.

One preferred class of intermediate products according to the present invention are thick film paste compositions, also referred to as thick film inks. These pastes are particularly useful for applying the phosphor particles onto a substrate, such as for use in display devices.

In the thick film process, a viscous paste that includes a functional particulate phase, such as phosphor powder, is screen printed onto a substrate. A porous screen fabricated from stainless steel, polyester, nylon or similar inert material is stretched and attached to a rigid frame. A predetermined pattern is

formed on the screen corresponding to the pattern to be printed. For example, a UV sensitive emulsion can be applied to the screen and exposed through a positive or negative image of the design pattern. The screen is then developed to remove portions of the emulsion in the pattern regions.

5 The screen is then affixed to a printing device and the thick film paste is deposited on top of the screen. The substrate to be printed is then positioned beneath the screen and the paste is forced through the screen and onto the substrate by a squeegee that traverses the screen. Thus, a pattern of traces and/or pads of the paste material is transferred to the substrate. The substrate
10 with the paste applied in a predetermined pattern is then subjected to a drying and heating treatment to adhere the functional phase to the substrate. For increased line definition, the applied paste can be further treated, such as through a photolithographic process, to develop and remove unwanted material from the substrate.

15 Thick film pastes have a complex chemistry and generally include a functional phase, a binder phase and an organic vehicle phase. The functional phase can include the phosphor powders of the present invention which provide a luminescent layer on a substrate. The particle size, size distribution, surface chemistry and particle shape of the particles all influence the rheology of the
20 paste.

 The binder phase is typically a mixture of inorganic binders such as metal oxide or glass frit powders. For example, PbO based glasses are commonly used as binders. The function of the binder phase is to control the sintering of the film and assist the adhesion of the functional phase to the substrate and/or
25 assist in the sintering of the functional phase. Reactive compounds can also be included in the paste to promote adherence of the functional phase to the substrate.

 Thick film pastes also include an organic vehicle phase that is a mixture of solvents, polymers, resins or other organics whose primary function is to
30 provide the appropriate rheology (flow properties) to the paste. The liquid

solvent assists in mixing of the components into a homogenous paste and substantially evaporates upon application of the paste to the substrate. Usually the solvent is a volatile liquid such as methanol, ethanol, terpineol, butyl carbitol, butyl carbitol acetate, aliphatic alcohols, esters, acetone and the like. The other organic vehicle components can include thickeners (sometimes referred to as organic binders), stabilizing agents, surfactants, wetting agents and the like. Thickeners provide sufficient viscosity to the paste and also acts as a binding agent in the unfired state. Examples of thickeners include ethyl cellulose, polyvinyl acetate, resins such as acrylic resin, cellulose resin, polyester, polyamide and the like. The stabilizing agents reduce oxidation and degradation, stabilize the viscosity or buffer the pH of the paste. For example, triethanolamine is a common stabilizer. Wetting agents and surfactants are well known in the thick film paste art and can include triethanolamine and phosphate esters.

The different components of the thick film paste are mixed in the desired proportions in order to produce a substantially homogenous blend wherein the functional phase is well dispersed throughout the paste. The powder is often dispersed in the paste and then repeatedly passed through a roll-mill to mix the paste. The roll mill can advantageously break -up soft agglomerates of powders in the paste. Typically, the thick film paste will include from about 5 to about 95 weight percent, such as from about 60 to 80 weight percent, of the functional phase, including the phosphor powders of the present invention.

Phosphor paste compositions are disclosed in: U.S. Patent No. 4,724,161; U.S. Patent No. 4,806,389; and U.S. Patent No. 4,902,567, each of which are incorporated herein by reference in their entirety. Generally, phosphors are deaggregated and are combined with organic additives to form the paste.

Some applications of thick film pastes require higher tolerances than can be achieved using standard thick-film technology, as is described above. As a result, some thick film pastes have photo-imaging capability to enable the

formation of lines and traces with decreased width and pitch. In this type of process, a photoactive thick film paste is applied to a substrate substantially as is described above. The paste can include, for example, a liquid vehicle such as polyvinyl alcohol that is not cross-linked. The paste is then dried and exposed to ultraviolet light through a photomask to polymerize the exposed portions of paste and the paste is developed to remove unwanted portions of the paste. This technology permits higher density lines and pixels to be formed. The combination of the foregoing technology with the phosphor powders of the present invention permits the fabrication of devices with resolution and tolerances as compared to conventional technologies using conventional phosphor powders.

In addition, a laser can be used instead of ultraviolet light through a mask. The laser can be scanned over the surface in a pattern thereby replacing the need for a mask. The laser light is of sufficiently low intensity that it does not heat the glass or polymer above its softening point. The unirradiated regions of the paste can be removed leaving a pattern.

Likewise, conventional paste technology utilizes heating of a substrate to remove the vehicle from a paste and to fuse particles together or modify them in some other way. A laser can be used to locally heat the paste layer and can be scanned over the paste layer thereby forming a pattern. The laser heating is confined to the paste layer and drives out the paste vehicle and heats the powder in the paste without appreciably heating the substrate. This allows heating of particles, delivered using pastes, without damaging a glass or even polymeric substrate.

Other deposition methods for the phosphor powders can also be used. For example, a slurry method can be used to deposit the powder. The powder is typically dispersed in an aqueous slurry including reagents such as potassium silicate and polyvinyl alcohol, which aids in the adhesion of the powder to the surface. For example, the slurry can be poured onto the substrate and left to settle to the surface. After the phosphor powder has sedimented onto the

substrate, the supernatant liquid is decanted off and the phosphor powder layer is left to dry.

Phosphor particles can also be deposited electrophoretically or electrostatically. The particles are charged and are brought into contact with the substrate surface having localized portions of opposite charge. The layer is typically lacquered to adhere the particles to the substrate. Shadow masks can be used to produce the desired pattern on the substrate surface.

Ink jet printing is another method for depositing the phosphor powders in a predetermined pattern. The phosphor powder is dispersed in a liquid medium and dispensed onto a substrate using an ink jet printing head that is computer controlled to produce a pattern. The phosphor powders of the present invention having a small size, narrow size distribution and spherical morphology can be printed into a pattern having a high density and high resolution. Other deposition methods utilizing a phosphor powder dispersed in a liquid medium include micro-pen or syringe deposition, wherein the powders are dispersed and applied to a substrate using a pen or syringe and are then allowed to dry.

Patterns of phosphors can also be formed by using an ink jet or micropen (small syringe) to dispense sticky material onto a surface in a pattern. Powder is then transferred to the sticky regions. This transfer can be done in several ways. A sheet covered with powder can be applied to the surface with the sticky pattern. The powder sticks to the sticky pattern and does not stick to the rest of the surface. A nozzle can also be used to transfer powder directly to the sticky regions.

Many methods for directly depositing materials onto surfaces require heating of the particles once deposited to sinter them together and densify the layer. The densification can be assisted by including a molecular precursor to a material in the liquid containing the particles. The particle/molecular precursor mixture can be directly written onto the surface using ink jet, micropen, and other liquid dispensing methods. This can be followed by heating in a furnace or heating using a localized energy source such as a laser. The heating converts

the molecular precursor into the functional material contained in the particles thereby filling in the space between the particles with functional material.

Thus, the phosphor powders produced according to the present invention result in smoother phosphor powder layers when deposited by such liquid or dry powder based deposition methods. Smoother phosphor powder layers are the result of the smaller average particle size, spherical particle morphology and narrower particle size distribution compared to phosphor powders produced by other methods. Smoother phosphor powder layers are valuable in various applications, especially those where the phosphor powders are used in a display device where a high resolution is critical. For example, a smoother phosphor powder layer in a display application where the phosphor layer produces light that is photographed results in improved definition and distinction of the photographed image.

A variety of deposition techniques often degrade the properties of the powders, especially the powder brightness. An example is the three-roll mill used to form pastes that are photoprinted, screen printed, directly written with a microsyringe and others. A method for increasing the brightness of the phosphor particles once deposited on the surface is to irradiate them with a laser (Argon ion, krypton ion, YAG, excimer, etc...). The laser light increases the temperature of the particles thereby annealing them and increasing the brightness. The laser heating of the particles can be carried out for particles on glass or even polymeric substrates since the laser causes local heating of the particles without heating the glass above its softening point. This approach is useful for the phosphor powders of the present invention.

The phosphor particle layer deposited onto a surface often needs to be coated to protect the layer from plasmas, moisture, electrons, photons, etc. Coatings can be formed by sputtering, but this method requires a mask to avoid deposition onto undesired areas of the substrate. Laser-induced chemical vapor deposition (LCVD) of metal oxides and other materials onto particles can allow localized deposition of material to coat phosphor particles without coating other

areas. The laser heating of the particles that drives the CVD can be carried out for particles on glass or even polymeric substrates because the laser causes local heating of the particles without heating the glass or polymer above its softening point.

5 Thus, the phosphor powders of the present invention have a unique combination of properties that are not found in conventional phosphor powders. The powders can advantageously be incorporated into a number of devices, wherein the devices will have significantly improved performance resulting directly from the characteristics of the phosphor powders of the present invention. The devices can include light-emitting lamps and display devices for
10 visually conveying information and graphics. Such display devices include traditional CRT-based display devices, such as televisions, and also include flat panel displays. Flat panel displays are relatively thin devices that present graphics and images without the use of a traditional picture tube and operate
15 with modest power requirements. Generally, flat panel displays include a phosphor powder selectively dispersed on a viewing panel, wherein the excitation source lies behind and in close proximity to the panel.

 For display devices, it is important for the phosphor layer to be as thin and uniform as possible with a minimal number of voids. Fig. 37 schematically
20 illustrates a lay down of large agglomerated particles in a pixel utilizing conventional phosphor powders. The device 3700 includes a transparent viewing screen 3702 and, in the case of an FED, a transparent electrode layer 3704. The phosphor particles 3706 are dispersed in pixels 3708. The phosphor particles are large and agglomerated and result in a number of voids and
25 unevenness in the surface. This results in decreased brightness and decreased image quality.

 Fig. 38 illustrates the same device fabricated utilizing powders according to the present invention. The device 3810 includes transparent viewing screen 3812 and a transparent electrode 3814. The phosphor powders 3816 are
30 dispersed in pixels in 3818. The pixels are thinner and more uniform than the

conventional pixel. In a preferred embodiment, the phosphor layer constituting the pixel has an average thickness of not greater than about 3 times the average particle size of the powder, preferably not greater than about 2 times the average particle size and even more preferably not greater than about 1.5 times the average particle size. This unique characteristic is possible due to the unique combination of small particle size, narrow size distribution and spherical morphology of the phosphor particles. The device will therefore produce an image having much higher resolution due to the ability to form smaller, more uniform pixels and much higher brightness since light scattering is significantly reduced and the amount of light lost due to non-luminescent particles is reduced.

Specifically, the brightness of the phosphor powder layer of the present invention is enhanced due to the thin, dense nature of the layer. The powder layer is comprised of spherical particles of the appropriate particle size distribution, preferably a bimodal distribution, resulting in a thinner, brighter layer.

CRT devices, utilizing a cathode ray tube, include traditional display devices such as televisions and computer monitors. CRTs operate by selectively firing electrons from one or more cathode ray tubes at cathodoluminescent phosphor particles, which are located in predetermined regions (pixels) of a display screen. The cathode ray tube is located at a distance from the display screen, which increases as screen size increases. By selectively directing the electron beam at certain pixels, a full color display with high resolution can be achieved.

A CRT display device is illustrated schematically in Fig. 39. The device 3902 includes 3 cathode ray tubes 3904, 3906 and 3908 located in the rear portion of the device. The cathode ray tubes generate electrons, such as electron 3910. An applied voltage of 20 to 30 kV accelerates the electrons toward the display screen 3912. In a color CRT, the display screen is patterned with red (R), green (G) and blue (B) phosphors, as is illustrated in Fig. 40. Three colored phosphor pixels are grouped in close proximity, such as group 3914, to produce multicolor images. Graphic output is created by selectively directing the

electrons at the pixels on the display screen 3912 using, for example, electromagnets 3916. The electron beams are rastered in a left to right, top to bottom fashion to create a moving image. The electrons can also be filtered through an apertured metal mask to block electrons that are directed at the wrong phosphor.

The phosphor powder is typically applied to the CRT display screen using a slurry. The slurry is formed by suspending the phosphor particles in an aqueous solution which can also include additives such as PVA (polyvinyl alcohol) and other organic compounds to aid in the dispersion of the particles in the solution as well as other compounds such as metal chromates. The display screen is placed in a coating machine, such as a spin coater, and the slurry is deposited onto the inner surface of the display screen and spread over the entire surface. The display screen is spun to thoroughly coat the surface and spin away any excess slurry. The slurry on the screen is then dried and exposed through a shadow mask having a predetermined dot-like or stripe-like pattern. The exposed film is developed and excess phosphor particles are washed away to form a phosphor screen having a predetermined pixel pattern. The process can be performed in sequence for different color phosphors to enable a full color display to be produced.

It is generally desired that the pixels are formed with a highly uniform phosphor powder layer thickness. The phosphors should not peel from the display screen and no cross contamination of the colored phosphors should occur. These characteristics are significantly influenced by the morphology, size and surface condition of the phosphor particles.

CRT devices typically employ cathodoluminescent phosphor particles rather than thin-film phosphors due to the high luminescence requirements. The resolution of images on powdered phosphor screens can be improved if the screen is made with particles having a small size and uniform size distribution, such as the phosphor particles according to the present invention. Image quality on the CRT device is also influenced by the packing voids of the particles and

the number of layers of phosphor particles, which are not involved in the generation of cathodoluminescence. That is, particles, which are not excited by the electron beam, will only inhibit the transmission of luminescence through the device. Large particles and aggregated particles both form voids and further contribute to loss of light transmission. Significant amounts of light can be scattered by reflection in voids. Further, for a high quality image, the phosphor layer should have a thin and highly uniform thickness. Ideally, the average thickness of the phosphor layer should be about 1.5 times the average particle size of the phosphor particles.

CRTs typically operate at high voltages such as from about 20 kV to 30 kV. Cathodoluminescent phosphors used for CRTs should have high brightness and good chromaticity. Cathodoluminescent phosphors, which are particularly useful in CRT devices include ZnS:Cu or Al for green, ZnS:Ag, Au or Cl for blue and $Y_2O_2S:Eu$ for red. The phosphor particles can advantageously be coated in accordance with the present invention to prevent degradation of the host material or diffusion of activator ions. Silica or silicate coatings can also improve the rheological properties of the phosphor slurry. The particles can also include a pigment coating, such as particulate Fe_2O_3 , to modify and enhance the properties of the emitted light.

Other CRT-based devices operating on a similar principle are heads-up and heads-down displays. A heads-up display is a small, high resolution display that is placed in close proximity to the eyes of a user, for example a pilot, so that the display can provide information to the user without requiring the user to be distracted. Such displays should have high brightness and good resolution. Similarly, heads-down displays are utilized, for example, in airplane cockpits to provide data to the pilots. Such phosphors should also be bright and have a long lifetime. The small, spherical phosphor powders of the present invention are ideally suited for such applications.

Another device utilizing cathodoluminescent phosphors is a vacuum fluorescent display (VFD). VFD's operate at an excitation voltage of less than

about 500 volts, such as about 40 volts. The most common VFD phosphor is presently ZnO:Zn (blue). ZnS, CdS and Zn, Cd:S are also useful for green and red. As is discussed above, the phosphor powders of the present invention are particularly advantageous at such low applied voltages.

5 The introduction of high-definition televisions (HDTV) has increased the interest in projection television (PTV). In this concept, the light produced by three independent cathode ray tubes is projected onto a faceplate on the tube that includes particulate phosphors, to form 3 colored projection images. The three images are projected onto a display screen by reflection to produce a full
10 color image. Because of the large magnification used in imaging, the phosphors on the faceplate of the cathode ray tube must be excited with an intense and small electron spot. Maximum excitation density may be two orders of magnitude larger than with conventional cathode ray tubes. Typically, the efficiency of the phosphor decreases with increasing excitation density. For the
15 foregoing reasons, the cathodoluminescent phosphor powders of the present invention are particularly useful in HDTV applications.

 One of the problems with CRT-based devices is that they are large and bulky and have significant depth as compared to the screen size. Therefore, there is significant interest in developing flat panel displays to replace CRT-
20 based devices in many applications.

 Flat panel displays (FPDs) offer many advantages over CRTs including lighter weight, portability and decreased power requirements. Flat panel displays can be either monochrome or color displays. It is believed that flat panel displays will eventually replace the bulky CRT devices, such as televisions,
25 with a thin product that can be hung on a wall, like a picture. Currently, flat panel displays can be made thinner, lighter and with lower power consumption than CRT devices, but not with the visual quality and cost performance of a CRT device.

 The high electron voltages and small currents traditionally required to
30 activate cathodoluminescent phosphors efficiently in a CRT device have

hindered the development of flat panel displays. Phosphors for flat panel displays such as field emission displays must typically operate at a lower voltage, higher current density and higher efficiency than phosphors used in existing CRT devices. The low voltages used in such displays, such as less than about 12 kV, result in an electron penetration depth in the range of several micrometers down to tens of nanometers, depending on the applied voltage. Thus, the control of the size and crystallinity of the phosphor particles is critical to device performance. If large or agglomerated powders are used, only a small fraction of the electrons will interact with the phosphor. Use of phosphor powders having a wide size distribution can also lead to non-uniform pixels and sub-pixels, which will produce a blurred image.

Further, the phosphor powders of the present invention are particularly applicable to excitation voltages of not greater than 12 kV, such as not greater than 8 kV and preferably not greater than 5 kV. This is due to the clean surface of the particles with a reduced number of surface defects, compared to a powder that has been milled.

One type of flat panel display is a field emission display (FED). These devices advantageously eliminate the size, weight and power consumption problems of CRTs while maintaining comparable image quality, and therefore are particularly useful for portable electronics, such as for laptop computers. FEDs generate electrons from millions of cold microtip emitters with low power emission that are arranged in a matrix addressed array with several thousand tip emitters allocated to each pixel in the display. The microtip emitters are typically located approximately 0.2 millimeter from a cathodoluminescent phosphor screen, which generates the display image. This allows for a thin, light-weight display.

Fig. 41 illustrates a high-magnification, schematic cross-section of an FED device according to an embodiment of the present invention. The FED device 4180 includes a plurality of microtip emitters 4182 mounted on a cathode 4184, which is attached to a backing plate 4186. The cathode is separated from a gate

or emitter grid 4188 by an insulating spacer 4190. Opposed to the cathode 4184 and separated by a vacuum is a faceplate assembly 4191 including phosphor pixel 4192 and a transparent anode 4194. The phosphor pixel layers can be deposited using a paste or electrophoretically. The FED can also include a transparent glass substrate 4196 onto which the anode 4194 is printed. During operation, a positive voltage is applied to the emitter grid 4188 creating a strong electric field at the emitter tip 4182. The electrons 4198 migrate to the faceplate 4191, which is maintained at a higher positive voltage. The faceplate collector bias is typically about 1000 volts. Several thousand microtip emitters 4182 can be utilized for each pixel in the display.

Cathodoluminescent phosphors, which are particularly useful for FED devices, include thiogallates such as $\text{SrGa}_2\text{S}_4:\text{Eu}$ for green, $\text{SrGa}_2\text{S}_4:\text{Ce}$ for blue and $\text{ZnS}:\text{Ag}$ or Cl for blue. $\text{Y}_2\text{O}_3:\text{Eu}$ can be used for red. $\text{ZnS}:\text{Ag}$ or Cu can also be used for higher voltage FED devices. $\text{Y}_2\text{SiO}_5:\text{Tb}$ or Eu can also be useful. For use in FED devices, these phosphors are preferably coated, such as with a very thin metal oxide coating, since the high electron beam current densities can cause breakdown and dissociation of the sulfur-containing phosphor host material. Dielectric coatings such as SiO_2 and Al_2O_3 can be used. Further, semiconducting coatings such as SnO or In_2O_3 can be particularly advantageous to absorb secondary electrons.

Coatings for the sulfur-containing FED phosphors preferably have an average thickness of from about 1 to 10 nanometers, more preferably from about 1 to 5 nanometers. Coatings having a thickness in excess of about 10 nanometers will decrease the brightness of the device since the electron penetration depth of 1-2 kV electrons is only about 10 nanometers. Such thin coatings can advantageously be monolayer coatings, as is discussed above.

The primary obstacle to further development of FEDs is the lack of adequate phosphor powders. FEDs require low-voltage phosphor materials, that is, phosphors which emit sufficient light under low applied voltages, such as less than about 500 volts, and high current densities. The cathodoluminescent

phosphor powders of the present invention advantageously have improved brightness under such low applied voltages and the coated phosphor particles of the present invention resist degradation under high current densities. The improved brightness can be attributed to the high crystallinity and high purity of the particles. Phosphor particles with low crystallinity and high impurities due to processes such as milling do not have the desirable high brightness. The phosphor particles of the present invention also have the ability to maintain the brightness and chromaticity over long periods of time, such as in excess of 10,000 hours. Further, the spherical morphology of the phosphor powder decreases light scattering and therefore improves the visual properties of the display. The small average size of the particles is advantageous since the electron penetration depth is only several nanometers, due to the low applied voltage.

The present invention is also applicable to other displays, including electroluminescent displays. EL displays are very thin structures, which can have very small screen sizes, such as few inches diagonally, while producing a very high resolution image. These displays, due to the very small size, are utilized in many military applications where size is a strict requirement such as in aircraft cockpits, small hand-held displays and heads-up displays. These displays function by applying a high electric potential between two addressing electrodes. EL displays are most commonly driven by an A.C. electrical signal. The electrodes are in contact with a semiconducting phosphor thin-film and the large potential difference creates hot electrons, which move through the phosphor, allowing for excitation followed by light emission.

An EL display is schematically illustrated in Figs. 42 and 43. The EL display device 4220 includes a phosphor layer 4222 sandwiched between two dielectric insulating layers 4224 and 4226. On the back side of the insulating layers is a backplate 4228, which includes row electrodes 4230. On the front of the device is a glass faceplate 4232, which includes transparent column electrodes 4234, such as electrodes made from transparent indium tin oxide.

While current electroluminescent display configurations utilize a thin film phosphor layer 4222 and do not typically utilize phosphor powders, the use of very small monodispersed phosphor particles according to the present invention is advantageous for use in such devices. For example, small monodispersed particles could be deposited on a glass substrate using a thick film paste and sintered to produce a well connected film and therefore could replace the expensive and material-limited CVD technology currently used to deposit such films. Such a well-connected film could not be formed from large, agglomerated phosphor particles. Similarly, composite phosphor particles are a viable alternative to the relatively expensive multilayer stack currently employed in electroluminescent displays. Thus, a composite phosphor particle comprising the phosphor and a dielectric material could be used.

Particularly preferred phosphors for use in electroluminescent display applications include the metal sulfides such as ZnS:Cu , CaS:Ce , SrS:RE (RE = rare earth), and ZnS:Mn . Further, mixed metal sulfides such as $\text{Sr}_x\text{Ca}_y\text{Ba}_{1-x-y}\text{S}_N\text{:Ce}$ can be used. Further, the thiogallate phosphors according to the present invention can also have advantages for use in electroluminescent displays. Oxide-based phosphors can also be useful, such as $\text{SrGa}_x\text{O}_y\text{:Eu}$, preferably including 5 to 7 atomic percent Eu, and $\text{CaGa}_x\text{O}_y\text{:Ce}$, preferably including from about 0.5 to 3 atomic percent Ce.

Another use for electroluminescent phosphor powders according to the present invention is in the area of electroluminescent lamps. Electroluminescent lamps are formed on a rigid or flexible substrate, such as a polymer substrate, and are commonly used as back lights for membrane switches, cellular phones, watches, personal digital assistants and the like. A simple electroluminescent lamp is schematically illustrated in Fig. 44. The device 4440 includes a phosphor powder/polymer composite 4442 is sandwiched between two electrodes 4444 and 4446, the front electrode 4444 being transparent. The composite layer 4442 includes phosphor particles 4448 dispersed in a polymer matrix 4450.

Electroluminescent lamps can also be formed on rigid substrates, such as stainless steel, for use in highway signage and similar devices. The rigid device includes a phosphor particle layer, a ceramic dielectric layer and a transparent conducting electrode layer. Such devices are sometimes referred to as solid state ceramic electroluminescent lamps (SSCEL). To form such rigid devices, a phosphor powder is typically sprayed onto a rigid substrate.

Electroluminescent lamp manufacturers currently have only simple metal sulfides such as ZnS phosphor powder host material at their disposal. ZnS:Cu produces a blue color, while ZnS:Mn, Cu produces an orange color. These materials have poor reliability and brightness, especially when filtered to generate other colors. Additional colors, higher reliability and higher brightness powders are critical needs for the electroluminescent lamp industry to supply designers with the ability to penetrate new market segments. The phosphor layers should also be thinner and denser, without sacrificing brightness, to minimize water intrusion and eliminate light scattering. Higher brightness electroluminescent lamps require thinner phosphor layers, which requires smaller particle size phosphor powders that cannot be produced by conventional methods. Such thinner layers will also use less phosphor powder. Presently available EL lamps utilize powders having an average size of about 5 μm or higher. The phosphor powders of the present invention having a small particle size and narrow size distribution, will enable the production of brighter and more reliable EL lamps that have an increased life-expectancy. Further, the phosphor powders of the present invention will enable the production of EL lamps wherein the phosphor layer has a significantly reduced thickness, without sacrificing brightness or other desirable properties. Conventional EL lamps have phosphor layers on the order of 100 μm thick. The powders of the present invention advantageously enable the production of an EL lamp having a phosphor layer that is not greater than about 15 μm thick, such as not greater than about 10 μm thick. The phosphor layer is preferably not thicker than about 3 times the weight

average particle size, more preferably not greater than about 2 times the weight average particle size.

As discussed above, preferred electroluminescent phosphors for use in electroluminescent lamps include ZnS:Cu for blue or blue-green and ZnS:Mn, Cu for orange. Other materials that are desirable for EL lamp applications include BaS:RE, Cu or Mn, CaS:RE or Mn, SrS:RE or Mn, and $Sr_xCa_yBa_{1-x-y}S:RE$ (where RE is a rare earth element) for other colors. CaS:Ga or Cu and SrS:Ga or Cu are also useful. The thiogallate phosphors of the present invention, such as $SrGa_2S_4$ and $CaGa_2S_4$, can be particularly advantageous for use in electroluminescent lamps. As is discussed above, many of these phosphors cannot be produced using conventional techniques and therefore have not been utilized in EL lamp applications. When used in an EL lamp, these phosphors should be coated to prevent degradation of the phosphor due to hydrolysis or other adverse reactions. Preferably, such a coating has an average thickness of from about 2 to 50 nanometers.

Electroluminescent powders used in AC powder EL (ACEL) are generally required to have a moderately conductive phase either on their surface or contained within them. As an example, ZnS:Cu is well known to exist as a ZnS host with copper sulfide crystallites contained within each particle. The copper sulfide crystallites are the source of the electric field that excites the dopant in the ZnS lattice the causes light emission. In the case of other electroluminescent powders, such as the metal oxide-based doped metal gallates ($M_xGa_yO_z$), these powders should also contain either a conductive additional phase in each particle or a conductive coating on the particle surface. For example, according to this embodiment of the invention, $ZnGa_2O_4$ exhibits better performance when it is either coated or contains an additional conductive phase such as ZnO.

As stated above, electroluminescent lamps are becoming increasingly important for back lighting alphanumeric displays in small electronic devices such as cellular phones, pagers, personal digital assistants, wrist watches,

calculators and the like. They are also useful in applications such as instrument panels, portable advertising displays, safety lighting, emergency lighting for rescue and safety devices, photographic backlighting, membrane switches and other similar applications. One of the problems associated with electroluminescent devices is that they generally require the application of alternating current (AC) voltage to produce light. A significant obstacle to the development of the useful direct current (DC) electroluminescent (DCEL) devices is a need for a phosphor powder that will function adequately under a DC electric field. The phosphor powder for functioning under a DC electric field should meet at least three requirements: 1) the particles should have a small average particle size; 2) the particles should have a uniform size, that is, the particles should have a narrow size distribution with no large particles or agglomerates; and 3) the particles should have good luminescence properties, particularly a high brightness. The phosphor powders of the present invention advantageously meet these requirements. Therefore, the electroluminescent phosphor powders of the present invention will advantageously permit the use of electroluminescent devices without requiring an inverter to convert a DC voltage to an AC voltage. Such devices are not believed to be commercially available at this time. When utilized in a device applying DC voltage, it is preferred to coat the phosphor particles with a thin layer of a conductive compound, such as a metal, for example copper metal, or a conductive compound such as copper sulfide.

The present invention is also applicable to photoluminescent phosphors. Photoluminescent phosphors are useful in plasma display devices. Plasma displays have an image quality that is comparable to current CRT devices and can be easily scaled to large sizes such as 20 to 60 diagonal inches. The displays are bright and lightweight and have a thickness of from about 1.5 to 3 inches. A plasma display functions in a similar manner as fluorescent lighting. In a plasma display, plasma source, typically a gas mixture, is placed between an opposed array of addressable electrodes and a high energy electric field is generated between the electrodes. Upon reaching a critical voltage, a plasma

is formed from the gas and UV photons are emitted by the plasma. Color plasma displays contain three-color photoluminescent phosphor particles deposited on the inside of the glass faceplate. The phosphors selectively emit light when illuminated by the photons. Plasma displays operate at relatively low currents and can be driven either by an AC or DC signal. AC plasma systems use a dielectric layer over the electrode, which forms a capacitor. This impedance limits current and provides a necessary charge in the gas mixture.

A cross-section of a plasma display device is illustrated in Fig. 45. The plasma display 4540 comprises two opposed panels 4542 and 4544 in parallel opposed relation. A working gas, typically including xenon, is disposed and sealed between the two opposing panels 4542 and 4544. The rear panel 4544 includes a backing plate 4546 on which are printed a plurality of electrodes 4548 (cathodes) which are in parallel spaced relation. An insulator 4550 covers the electrodes and spacers 4552 are utilized to separate the rear panel 4544 from the front panel 4542.

The front panel 4542 includes a glass face plate 4554, which is transparent when observed by the viewer (V). Printed onto the rear surface of the glass face plate 4554 are a plurality of electrodes 4556 (anodes) in parallel spaced relation. An insulator 4558 separates the electrode from the pixels of photoluminescent phosphor powder 4560. The phosphor powder 4560 is typically applied using a thick film paste. When the display 4540 is assembled, the electrodes 4548 and 4556 are perpendicular to each other, forming an XY grid. Thus, each pixel of phosphor powder can be activated by the addressing an XY coordinate defined by the intersecting electrodes 4548 and 4556.

One of the problems currently encountered in plasma display devices is the long decay time of the phosphor particles, which creates a "tail" on a moving image. Through control of the phosphor chemistry, such decay-related problems can be reduced. Further, the spherical, non-agglomerated nature of the phosphor particles improves the resolution of the plasma display panel.

Plasma display panels typically operate using a xenon gas composition. The $Y_2O_3:Eu$ phosphors of the present invention, doped with from 4 to about 6 atomic percent Eu, are useful for plasma displays for providing red color. $(Y,Gd)BO_3$ phosphors including about 14 to 20 atomic percent Eu are also useful for red color. Further, the $Zn_2SiO_4:Mn$ phosphors of the present invention, preferably with 0.05 to 2 atomic percent Mn, are useful for producing a green color. BAM phosphors of the present invention are useful for producing a blue color, particularly when doped with from about 8 to 12 atomic percent Eu. The phosphors can advantageously be coated, such as with MgO, to reduce degradation from the plasma.

The photoluminescent phosphors of the present invention are also useful as taggents for security purposes. In this application the phosphors, which are undetectable under normal lighting, become visible upon illumination by a particular energy.

For security purposes, the phosphor particles are dispersed into a liquid vehicle which can be applied onto a surface by standard ink deposition methods, such as by using an ink jet or a syringe, or by screen printing. The phosphor particles of the present invention, having a small size and narrow size distribution, advantageously permit better control over the printed feature size and complexity. The methodology of the present invention also permits unique combinations of phosphor compounds that are not available using conventional methods. Such phosphors can be applied to currency, confidential documentation, explosives and munitions, or any other item that may require positive identification. The phosphor powders can advantageously be dispersed in an ink, which is then used to form indicia on a document or other item, such as a postal envelope.

Useful phosphor compounds for security applications include $Y_2O_3:Eu$, preferably including 6 to 9 atomic percent Eu. $(Y,Gd)BO_3$ including 14 to 20 atomic percent Eu can also be useful. Such phosphors emit visible light upon excitation by an infrared source. The phosphor powders of the present invention

provide many advantages in such applications. For example, the small, monodispersed nature of the particles makes the particles easy to supply in smaller quantities.

In addition to the foregoing, the phosphors of the present invention can also be used as target materials for the deposition of phosphor thin-films by electron beam deposition, sputtering and the like. The particles can be consolidated to form the target for the process. The homogenous concentration of activator ions in the particles will lead to more uniform and brighter film.

Further, the photoluminescent phosphor particles according to the present invention can be used for fluorescent lighting elements, including common overhead lighting tubes as well as lamps used to backlight LCD displays, which are commonly used on laptop computers. Typically, the excitation source for such displays includes mercury vapor.

A fluorescent tubular lighting element is schematically illustrated in Fig. 46. The lighting element 4600 includes a glass tube 4602 that is sealed to contain a gas composition, typically including mercury. The ends of the glass tube 4602 include electrodes 4604 and 4606 that ionize the gas composition thereby stimulating photoluminescent phosphors 4608 that are disposed on the inner surface of the glass tube 4602. The mixture of different color phosphors produces a white light, which is typically desired in fluorescent lighting applications.

Fig. 47 schematically illustrates a close-up of the phosphor powders 4608 of the present invention on a glass tube 4602. The small and spherical phosphors form a uniform thin layer having a reduced average thickness, such as less than about 3 times the average particle size of the phosphor particles 4608. This uniform and thin surface can advantageously produce bright, uniform light while consuming less power than conventional phosphor powders. It will be appreciated that the lighting element can have a variety of shapes and forms, and often are in the form of very thin tubing for use as an LCD backlight. Such LCD backlights are particularly applicable to the present invention since they

require phosphor powders that are small, spherical and have a narrow size distribution so that the powders will readily disperse in the lighting element.

Preferred photoluminescent phosphor powders for such fluorescent lighting applications including $Y_2O_3:Eu$, particularly those including from about 6 to about 9 atomic percent Eu. $(Y,Gd)BO_3:Eu$, including from about 14 to about 20 atomic percent Eu, are also useful. Other useful phosphors include $Zn_2SiO_4:Mn$, particularly from about 0.05 to about 2 atomic percent Mn as well as $BAM:Eu$, including from about 6 to about 12 atomic percent Eu. Combinations of the foregoing phosphors can be used to produce a white light element. To form these lighting elements, the phosphor particles are typically dispersed in a liquid vehicle such as a slurry which are then applied to the interior of a glass tube and dried to form the phosphor layer.

The present invention is also applicable to LEDs. LEDs (light emitting diodes) are solid-state devices that emit light of a particular wavelength when powered by an external electric circuit. The advantages of LEDs are their high brightness and low power consumption. LED efficiencies are typically around 45 lumens/watt, compared to about 12 lumens/watt for fluorescent lamps. LEDs are planned to be used for household and commercial lighting -- replacing incandescent light bulbs and possibly replacing fluorescent lamps -- traffic lights, large area visual displays and security devices. For use as a security device, the phosphor would be placed in an article such as a document, currency, passport, stamp, bankcard, or the like and an LED would be used to verify the authenticity of the article by making the phosphor luminesce.

LEDs are generally based on nitrides such as GaN or $(In,Ga)N$ or other 3/5 or 2/6 compositions such as GaP, GaAs or ZnSe. Polymer LEDs are also known. The nature of the materials and their band-gap determines the wavelength of the light that is emitted and the emitted light has a wavelength that varies between the UV and red.

Currently, the most common LEDs emit blue light having a wavelength of around 450 nm. It is desirable to have an LED-based white light source, but this

is not possible with an LED alone. However, the color of the light emitted by an LED can be modified by placing a layer of luminescent material, such as a photoluminescent phosphor powder, on top of the LED such that the light emitted by the LED excites the phosphor powder which re-emits the light at a different wavelength. By selecting the proper phosphor powder, white LEDs can be produced. An example of a blue-to-white LED phosphor is YAG:Ce, where YAG (yttrium aluminum garnet) is $\text{Y}_3\text{Al}_5\text{O}_{12}$. There are many variations on this composition which result in subtle variations of color temperature, such as $(\text{Y,Gd})_3\text{Al}_5\text{O}_{12}:\text{Ce}$ and $(\text{Y,Gd})_3(\text{Ga,Al})_5\text{O}_{12}:\text{Ce}$.

Other materials that are useful for LED applications are the existing plasma TV phosphors, such as BAM:Eu. Future generations of LEDs will be optimized to emit light in the UV range and UV-excited phosphors such as those currently used in the lighting and plasma TV industry will be useful to achieve RGB colors for displays. There are also LEDs that emit in the IR range, and upconverter phosphors would be useful to convert the IR to visible light.

The powders of the present invention will be particularly useful in the foregoing LED applications. In particular, the phosphor powders will advantageously have higher efficiencies since milling of the powder is not required, leading to clean, defect-free surfaces. The spherical particle morphology and bimodal size distribution will enable higher brightness with less material through better phosphor layer characteristics. Further, the versatility of spray processing will enable the formation of many existing and new compositions, including complex compositions.

The present invention is also applicable to x-ray phosphors. One preferred device utilizing the x-ray phosphors of the present invention is an x-ray image intensifier. As is illustrated in Fig. 48, a subject 4802 is placed between an x-ray source 4804 and an image intensifying screen 4806. Behind the image intensifying screen is photographic film 4808 that captures the image. As x-rays are passed through the subject 4802, some are absorbed or deflected and the resulting pattern of x-ray energy impinges on the intensifying screen 4806. The

screen 4806 includes x-ray phosphors, which convert the x-ray energy to visible light, and the light strikes the photographic film 4808, thus producing an image.

A cross-section of an x-ray image intensifying screen is illustrated in Fig. 49. The screen 4900 utilizes two intensifying screens 4902 and 4904. The screens each include a base 4906 and 4908 and a layer of x-ray phosphors 4910 and 4912. The x-rays impinge on the x-ray phosphors, which then emit visible light and produces an image on the photographic film 4914.

Phosphors for x-ray imaging should have a high x-ray absorption capability, a high density and a blue or green luminescence to match the sensitivity of the film. Particular x-ray phosphors which meet these requirements include: $\text{Gd}_2\text{O}_2\text{S:Tb}$, preferably including from about 5 to about 20 atomic percent Tb; Gd_2SiO_5 or $(\text{Y,Gd})_2\text{SiO}_5\text{:Tb}$, particularly including from about 5 to about 20 mole percent Tb; and $(\text{Y,Gd})\text{BO}_3\text{:Tb}$, preferably including from about 10 to about 20 atomic percent Tb. Other x-ray phosphors include those based on Lu_2SiO_5 , GdTao_4 , PbHfO_3 , HfO_2 and $\text{Gd}_3\text{Ga}_5\text{O}_{12}$. Tungstates such as CaWO_4 , SrWO_4 and PbWO_4 are also useful.

The x-ray phosphor particles of the present invention are also advantageous for the manufacture of intensifying screens since the phosphors can form thin, uniform layers having high brightness. The high relative density of the phosphor layer can improve the efficiency and resolution of the image intensifier.

EXAMPLES

A yttrium oxide phosphor powder batch was produced according to the present invention. An aqueous precursor solution was formed comprising yttrium nitrate and europium nitrate in a ratio to yield a phosphor comprising Y_2O_3 and 8.6 atomic percent Eu. The total precursor concentration was 7.5 weight percent based on the final product.

The liquid solution was atomized using ultrasonic transducers at a frequency of 1.6 MHz. Air was used as a carrier gas and the aerosol was carried

through a tubular furnace having a temperature of 800°C. The total residence time in the furnace was about 1-2 seconds. The pyrolyzation at 800°C resulted in intermediate precursor particles of a low crystallinity yttrium compound.

The intermediate precursor particles were then heated in batch mode at a temperature of 1400°C for 60 minutes in air. The heating ramp rate was 10EC/minute.

The resulting powder is illustrated in the SEM photomicrograph of Fig. 50. The particle size distribution is illustrated in Fig. 51. The average particle size was 2.476 μm and 90 percent of the particles had a size of less than 4.150 μm . The x-ray diffraction pattern illustrated in Fig. 52 shows that the particles are substantially phase pure Y_2O_3 .

A zinc silicate powder batch was produced according to the present invention. A precursor solution was formed comprising zinc nitrate and manganese nitrate along with colloidal silica (Cabot L-90, Cabot Corporation, Massachusetts). An excess of 50 molar percent silica was used in the precursor liquid and the concentration of manganese was 5 atomic percent. The total precursor concentration was about 7.5 weight percent based on the final product. The liquid solution was atomized using ultrasonic transducers at a frequency of 1.6 MHz. Air was used as a carrier gas and the aerosol was carried through a tubular furnace having a temperature of 900°C. The total residence time in the furnace was about 1-2 seconds. The pyrolyzation at 900°C resulted in intermediate precursor particles having a low crystallinity.

The intermediate precursor particles were then heated in batch mode at a temperature of 1175°C for 60 minutes in air. The heating ramp rate was 10EC/minute.

The resulting powder is illustrated in the SEM photomicrograph of Fig. 53. The particle size distribution is illustrated in Fig. 54. The average particle size was 2.533 μm and 90 percent of the particles had a size of less than 4.467 μm . The x-ray diffraction pattern illustrated in Fig. 55 shows that the particles are substantially phase pure Zn_2SiO_4 .

For the production of a thiogallate, an aqueous solution was formed including 2 mole equivalents gallium nitrate ($\text{Ga}(\text{NO}_3)_3$) and 1 mole equivalent strontium nitrate ($\text{Sr}(\text{NO}_3)_2$). About 0.05 mole equivalents of europium nitrate ($\text{Eu}(\text{NO}_3)_3$) was also added.

5 The aqueous solution was formed into an aerosol using ultrasonic transducers of a frequency of about 1.6 MHz. The aerosol was carried in air through a furnace heated to a temperature of about 800°C to spray-convert the solution. The intermediate product was a SrGa_2O_4 oxide having a small average particle size and low impurities.

10 The oxide intermediate precursor product was then heated at 900°C under a flowing gas that included H_2S and nitrogen in a 1:1 ratio, for about 20 minutes. The resulting powder was substantially phase pure $\text{SrGa}_2\text{S}_4:\text{Eu}$ (3 atomic percent Eu) having good crystallinity and good luminescent characteristics.

15 In a further example, $\text{SrGa}_2\text{O}_4:\text{Eu}$ was prepared by mixing the appropriate ratios of strontium carbonate, gallium nitrate and europium nitrate. The total precursor concentration was 5 weight percent.

20 The liquid solution was atomized using ultrasonic transducers at a frequency of 1.6 MHz. Air was used as the carrier gas and the aerosol was carried through a furnace heated to 800°C. This process formed an $\text{SrGa}_2\text{O}_4:\text{Eu}$ phosphor having about 4 atomic percent Eu. This powder is illustrated in Fig. 56.

25 The following examples demonstrate the advantages of the BAM phosphor powders according to the present invention and the method for making the powders.

1. BAM Precursor Selection

30 Using aluminum nitrate hydrate ($\text{Al}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$), a 5 weight percent solution was made based on $\text{Ba}_{0.95}\text{Eu}_{0.05}\text{MgAl}_{10}\text{O}_{17}$ (i.e. 5 atomic percent europium as a dopant). Stirring the solution resulted in a crystalline deposition of barium nitrate. Four equivalents of ethylene glycol were added as a chelating

agent to increase the solubility of the barium nitrate, however the deposit still remained.

Similar solutions containing aluminum nitrate and having a total precursor concentration of 5, 7.5 and 10 weight percent were prepared. Polyether was added as a second chelating agent to increase the solubility of the barium nitrate. The results are listed in Table III.

Table III

Example	Weight Percent Solution	Result
BAM-1	5.0	Mostly in solution
BAM-2	7.5	Slight suspension of solid
BAM-3	10.0	Difficult to put into solution

The solutions from Examples BAM-1 & BAM-2 were nebulized using an ultrasonic generator at a frequency of about 1.6 MHz and carried through a furnace at 750°C, substantially as described herein. Both atomized poorly (low production rate), although some powder was collected from Example BAM-1. The collected powder had a brown/black appearance. Heating the powder to 1350°C under flowing forming gas (7% H₂/93%N₂) produced a grey powder that x-ray diffraction indicated was phase pure. Further heat treatment in air at 1350°C for 2 hours produced a white, phase pure powder.

Nonetheless, it was concluded that aluminum nitrate as a precursor was not a viable alternative due to solubility problems associated with barium nitrate.

2. BAM Process Temperature

Based on the foregoing, fumed particulate alumina was selected as a precursor component. Europium nitrate, barium nitrate and magnesium nitrate were used as the remaining precursors, for a total precursor concentration of 7.5 weight percent based on the final BAM composition. The solution was nebulized using ultrasonic transducers at a frequency of about 1.6 MHz to form an aerosol.

The aerosol was carried through a tubular furnace at varying temperatures, as indicated in Table IV.

Table IV

Example	Conversion Temperature	Crystalline Phases	Crystalline Phases after Heat Treatment
BAM-4	600°C	Ba(NO ₃) ₂	BaAl ₂ O ₄ , BAM
BAM-5	700°C	Ba(NO ₃) ₂	BaAl ₂ O ₄ , BAM
BAM-6	800°C	Amorphous	BaAl ₂ O ₄ , BAM
BAM-7	900°C	Mainly amorphous Some BaO	BaAl ₂ O ₄ , BAM

Below 700°C, the barium nitrate did not fully decompose. It was concluded that a preferred range for producing a spray-converted intermediate product was at least about 700°C and preferably 800°C to 900°C.

A sample of each of the spray-converted intermediate powders was heat treated at 1500°C for 2 hours under a forming gas of 10% H₂/ 90% N₂ and a sample of each powder was heat treated at 1400°C for 1 hour in air. Each post treatment resulted in a powder that was not phase pure (BaAl₂O₄ impurity) and there was virtually no difference in the x-ray diffraction patterns for the powders treated in forming gas versus the powders treated in air.

3. BAM Precursor Concentration

In an effort to eliminate the aluminate impurity identified above, the metal stoichiometry in the precursor solution was varied. The solutions and results of processing are summarized in Table V. All solutions were formed into an aerosol using ultrasonic atomizers at a frequency of about 1.6 MHz and were spray converted at a temperature of 750°C in a tubular furnace.

Table V

Example	Precursor Stoichiometry	Heat Treatment Atmosphere	Weight Loss during Heat Treatment	Crystalline Phases of Final Powder	Relative PL Intensity
BAM-8	$\text{Ba}_{0.95}\text{Eu}_{0.05}\text{Mg}_{1.5}\text{Al}_{10}\text{O}_x$	H_2/N_2	10.4%	BaAl_2O_4	13
BAM-9	$\text{Ba}_{0.95}\text{Eu}_{0.05}\text{Mg}_{1.75}\text{Al}_{10}\text{O}_x$	H_2/N_2	8.5%	BaAl_2O_4	18
BAM-10	$\text{Ba}_{0.95}\text{Eu}_{0.05}\text{Mg}_{2.0}\text{Al}_{10}\text{O}_x$	H_2/N_2	8.2%	BaAl_2O_4	7
BAM-11	$\text{Ba}_{0.95}\text{Eu}_{0.05}\text{MgAl}_{12}\text{O}_x$	H_2/N_2	7.8%	BaAl_2O_4 + BAM	42
BAM-12	$\text{Ba}_{0.95}\text{Eu}_{0.05}\text{MgAl}_{14}\text{O}_x$	H_2/N_2	7.2%	BAM	37
BAM-13	$\text{Ba}_{0.95}\text{Eu}_{0.05}\text{MgAl}_{16}\text{O}_x$	H_2/N_2	5.9%	BAM	29
BAM-14	$\text{Ba}_{0.95}\text{Eu}_{0.05}\text{MgAl}_{20}\text{O}_x$	H_2/N_2	4.5%	BAM	14
BAM-15	$\text{Ba}_{0.95}\text{Eu}_{0.05}\text{Mg}_{0.89}\text{Al}_{13}\text{O}_x$	Air	12.0%	BaAl_2O_4 + BAM	—
BAM-16	$\text{Ba}_{0.95}\text{Eu}_{0.05}\text{Mg}_{1.11}\text{Al}_{13}\text{O}_x$	Air	13.6%	BaAl_2O_4 + BAM	—
BAM-17	$\text{Ba}_{0.95}\text{Eu}_{0.05}\text{MgAl}_{13}\text{O}_x$	Air	7.9%	BaAl_2O_4 + BAM	—
BAM-18	$\text{Ba}_{0.95}\text{Eu}_{0.05}\text{MgAl}_{12}\text{O}_x$	Air	7.0%	BaAl_2O_4 + BAM	—
BAM-19	$\text{Ba}_{0.95}\text{Eu}_{0.05}\text{MgAl}_{14}\text{O}_x$	Air	7.0%	BAM	—

Increasing the magnesium content in Examples BAM-8, 9 and 10 did not lead to a decrease of the barium aluminate phase. However, when the aluminum content was increased keeping the barium and magnesium ratio fixed, phase pure BAM was produced. Therefore, it was concluded that an excess of aluminum in the precursor, such as 40% or more excess, is desirable for producing phase pure BAM.

The photoluminescence (PL) data indicates that the maximum intensity occurs for powders treated in a reducing atmosphere, likely due to the reduction of europium to the Eu^{2+} state.

Based on the foregoing examples, BAM powder was produced as follows. Barium nitrate, europium nitrate, magnesium nitrate and colloidal alumina were formed into a precursor solution having a concentration of about 8 weight

percent based on the equivalent of BAM. Europium was incorporated at 5 atomic percent. The solution was atomized using ultrasonic transducers at a frequency of about 1.6 MHz. The aerosol was pyrolyzed in a tubular furnace at a reaction temperature of 950°C to form a spray-converted intermediate precursor powder. The powder was then heat-treated at 1450°C for 2 hours under an atmosphere of H_2/N_2 . This powder is illustrated the photomicrograph of Fig. 57. The particle size distribution of the powder is illustrated in Fig. 58. The average particle size was 1.88 μm and 90% of the particles by weight were less than 3.51 μm in size. The powder exhibited strong photoluminescence characteristics.

A yttrium gadolinium borate powder batch was also produced according to the present invention. An aqueous precursor solution was formed comprising yttrium nitrate, gadolinium nitrate, europium nitrate and boric acid in a ratio to yield a (Y,Gd)BO₃ phosphor having a Y:Gd ratio of 3 and an Eu concentration of 16 atomic percent. The total precursor concentration was 8.0 weight percent based on the final product.

The liquid solution was atomized using ultrasonic transducers at a frequency of 1.6 MHz. Air was used as a carrier gas and the aerosol was carried through a tubular furnace having a temperature of 950°C. The total residence time in the furnace was about 1-2 seconds. The pyrolyzation at 950°C resulted in intermediate precursor particles having low crystallinity.

The intermediate precursor particles were then heated in batch mode at a temperature of 1150°C for 60 minutes in air. The heating ramp rate was 10EC/minute.

The resulting powder is illustrated in the SEM photomicrograph of Fig. 59. The particle size distribution is illustrated in Fig. 60. The average particle size was 2.139 μm and 90 percent of the particles had a size of less than 3.608 μm . The x-ray diffraction pattern illustrated in Fig. 61 shows that the particles are substantially phase pure yttrium borate and gadolinium borate having a high crystallinity.

An x-ray phosphor compound, $Y_{1.88}Gd_{0.12}SiO_5:Ce$, that is useful as an x-ray phosphor was produced according to the present invention. A liquid precursor solution was formed including yttrium nitrate, gadolinium nitrate, cerium nitrate and colloidal silica (Cabot HS-5, Cabot Corporation, Boyertown, PA) to yield the foregoing compound including 0.5 atomic percent Ce. The total precursor concentration was 5 weight percent based on the equivalent weight of the final compound. The liquid was atomized into an aerosol using ultrasonic transducers operating at a frequency of 1.6 MHz and the aerosol was carried in air through furnace having a temperature of about 650°C to produce intermediate precursor particles.

The intermediate precursor particles were then batch annealed in air at a temperature of 1350°C for 1 hour. The resulting powder had a small particle size and a narrow particle size distribution.

While various embodiments of the present invention have been described in detail, it is apparent that modifications and adaptations of those embodiments will occur to those skilled in the art. However, it is to be expressly understood that such modifications and adaptations are within the spirit and scope of the present invention.

What is Claimed is:

1. A powder batch comprising cathodoluminescent phosphor particles, wherein said particles have a volume average particle size of from about 0.1 μm to about 10 μm and have a substantially spherical morphology, wherein at least about 70 volume percent of said particles are not larger than about two times said average particle size.

2. A powder batch as recited in Claim 1, wherein said particles have a volume average particle size of from about 0.3 μm to about 10 μm .

3. A powder batch as recited in Claim 1, wherein said particles have a volume average particle size of from about 0.3 μm to about 5 μm .

4. A powder batch as recited in Claim 1, wherein said particles comprise yttrium oxide.

5. A powder batch as recited in Claim 4, wherein said particles further comprise Eu.

6. A powder batch as recited in Claim 1, wherein said particles comprise yttrium oxysulfide.

7. A powder batch as recited in Claim 6, wherein said particles further comprise a dopant selected from the group consisting of Eu, Tb and combinations thereof.

8. A powder batch as recited in Claim 1, wherein said particles comprise zinc sulfide.

9. A powder batch as recited in Claim 8, wherein said particles further comprise a dopant selected from the group consisting of Au, Al, Cu and combinations thereof.

10. A powder batch as recited in Claim 8, wherein said particles further comprise a dopant selected from the group consisting of Ag, Cl and combinations thereof.

11. A powder batch as recited in Claim 1, wherein said particles comprise strontium thiogallate.

12. A powder batch as recited in Claim 11, wherein said particles further comprise a dopant selected from the group consisting of Eu, Ce and combinations thereof.

13. A powder batch as recited in Claim 1, wherein said particles
5 comprise yttrium aluminate.

14. A powder batch as recited in Claim 13, wherein said particles further comprise a dopant selected from the group consisting of Tb and Cr.

15. A powder batch as recited in Claim 1, wherein said particles
comprise zinc silicate.

10 16. A powder batch as recited in Claim 15, wherein said particles further comprise Mn.

17. A powder batch as recited in Claim 1, wherein said particles
comprise yttrium silicate.

18. A powder batch as recited in Claim 17, wherein said particles
15 further comprise a dopant selected from the group consisting of Tb and Ce.

19. A powder batch as recited in Claim 1, wherein at least about 90
volume percent of said particles are not larger than about two times said average
particle size.

20. A powder batch as recited in Claim 1, wherein said phosphor
20 particles comprises crystallites having an average crystallite size of at least
about 25 nanometers.

21. A powder batch comprising electroluminescent phosphor particles,
wherein said particles have a volume average particle size of from about 0.1 μm
to about 10 μm and have a substantially spherical morphology, wherein at least
25 about 70 volume percent of said particles are not larger than about two times
said average particle size.

22. A powder batch as recited in Claim 21, wherein said particles have
a volume average particle size of from about 0.3 μm to about 5 μm .

23. A powder batch as recited in Claim 21, wherein said particles
30 comprise zinc sulfide.

24. A powder batch as recited in Claim 23, wherein said particles comprise a dopant selected from the group consisting of Au, Al, Cu and combinations thereof.

25. A powder batch as recited in Claim 23, wherein said particles comprise a dopant selected from the group consisting of Ag, Cl and combinations thereof.

26. A powder batch as recited in Claim 23, wherein said particles comprise Mn.

27. A powder batch as recited in Claim 21, wherein said particles comprise M^1S wherein M^1 is selected from the group consisting of Ca, Sr and combinations thereof.

28. A powder batch as recited in Claim 27, wherein said particles comprise a dopant selected from the group consisting of Eu, Ce and combinations thereof.

29. A powder batch as recited in Claim 21, wherein said particles comprise $M^2Ga_2S_4$ wherein M^2 is selected from the group consisting of Ca, Sr and combinations thereof.

30. A powder batch as recited in Claim 29, wherein said particles comprise a dopant selected from the group consisting of Eu, Ce and combinations thereof.

31. A powder batch as recited in Claim 21, wherein said particles comprise zinc gallate and a dopant selected from the group consisting of Mn and Cr.

32. A powder batch as recited in Claim 21, wherein said particles comprise $M^3Ga_2O_4$ wherein M^3 is selected from the group consisting of Ca and Sr and a dopant selected from the group consisting of Ce and Eu.

33. A powder batch as recited in Claim 21, wherein said particles comprise Ga_2O_3 and a dopant selected from the group consisting of Dy and Eu.

34. A powder batch as recited in Claim 21, wherein said particles comprise $Ca_3Ga_2O_6:Eu$.

35. A powder batch as recited in Claim 21, wherein said particles comprise $\text{Zn}_2\text{GeO}_4\text{:Mn}$.

36. A powder batch as recited in Claim 21, wherein said particles comprise $\text{Zn}_2(\text{Ge,Si})\text{O}_4\text{:Mn}$.

5 37. A powder batch as recited in Claim 21, wherein at least about 90 volume percent of said particles are not larger than about two times said average particle size.

38. A powder batch as recited in Claim 21, wherein said particles are comprised of crystallites having an average crystallite size of at least about 25
10 nanometers.

39. A powder batch as recited in Claim 21, wherein said particles comprise an activator ion.

40. A powder batch comprising photoluminescent phosphor particles, wherein said particles have a volume average particle size of from about $0.1\ \mu\text{m}$ to about $10\ \mu\text{m}$ and have a substantially spherical morphology, wherein at least
15 about 70 volume percent of said particles are not larger than two times said average particle size.

41. A powder batch as recited in Claim 40, wherein said particles have a volume average particle size of from about $0.3\ \mu\text{m}$ to about $5\ \mu\text{m}$.

20 42. A powder batch as recited in Claim 40, wherein said particles comprise yttrium oxide.

43. A powder batch as recited in Claim 42, wherein said particles further comprise a dopant selected from the group consisting of Eu, Tb and combinations thereof.

25 44. A powder batch as recited in Claim 40, wherein said particles comprise yttrium-gadolinium borate.

45. A powder batch as recited in Claim 44, wherein said particles further comprise Eu.

30 46. A powder batch as recited in Claim 40, wherein said particles comprise zinc silicate.

47. A powder batch as recited in Claim 46, wherein said particles further comprise Mn.

48. A powder batch as recited in Claim 40, wherein said particles comprise barium magnesium aluminate.

5 49. A powder batch as recited in Claim 48, wherein said particles further comprise a dopant selected from the group consisting of Mn and Eu.

50. A powder batch as recited in Claim 40, wherein said particles comprise $\text{BaAl}_x\text{O}_y\text{:Mn}$.

10 51. A powder batch as recited in Claim 40, wherein at least about 90 volume percent of said particles are not larger than about two times said average particle size.

52. A powder batch as recited in Claim 40, wherein said particles comprise crystallites having an average crystallite size of at least about 25 nanometers.

15 53. A powder batch comprising x-ray phosphor particles, wherein said particles have a volume average particle size of from about $0.1\ \mu\text{m}$ to about $10\ \mu\text{m}$ and a substantially spherical morphology, wherein at least about 70 volume percent of said particles are not larger than about two times said average particle size.

20 54. A powder batch as recited in Claim 53, wherein said phosphor particles have a volume average particle size of from about $0.3\ \mu\text{m}$ to about $5\ \mu\text{m}$.

55. A powder batch as recited in Claim 53, wherein said phosphor particles comprise a gadolinium-containing phosphor compound.

25 56. A powder batch as recited in Claim 53, wherein said phosphor particles comprise gadolinium oxysulfide.

57. A powder batch as recited in Claim 56, wherein said phosphor particles further comprise Tb.

30 58. A powder batch as recited in Claim 53, wherein said phosphor particles comprise $(\text{Y,Gd})_2\text{SiO}_5$.

59. A powder batch as recited in Claim 58, wherein said phosphor particles further comprise Tb.

60. A powder batch as recited in Claim 53, wherein at least about 90 volume percent of said particles are not larger than about two times said average particle size.

61. A powder batch as recited in Claim 53, wherein said phosphor particles comprise crystallites having an average crystallite size of at least about 25 nanometers.

62. A flowable medium suitable for applying phosphor particles onto a substrate, comprising:

a) a liquid vehicle phase; and
b) a functional phase dispersed throughout said liquid vehicle phase, said functional phase comprising phosphor particles having a substantially spherical morphology and a volume average particle size of not greater than about 5 μm .

63. A flowable medium as recited in Claim 62, wherein said phosphor particles have a particle size distribution wherein at least about 70 volume percent of said phosphor particles are not larger than twice said average particle size.

64. A flowable medium as recited in Claim 62, wherein said phosphor particles have a particle size distribution wherein at least about 80 volume percent of said phosphor particles are not larger than twice said average particle size.

65. A flowable medium as recited in Claim 62, wherein said vehicle phase is an aqueous-based solution.

66. A flowable medium as recited in Claim 62, wherein said vehicle phase is an aqueous-based solution comprising a dispersing agent.

67. A flowable medium as recited in Claim 62, wherein said flowable medium comprises from about 5 to about 95 weight percent of said phosphor particles.

68. A flowable medium as recited in Claim 62, wherein said flowable medium comprises from about 60 to about 85 weight percent of said phosphor particles.

5 69. A flowable medium as recited in Claim 62, wherein said phosphor particles comprise yttrium oxide.

70. A flowable medium as recited in Claim 62, wherein said phosphor particles comprise yttrium oxysulfide.

71. A flowable medium as recited in Claim 62, wherein said phosphor particles comprise zinc sulfide.

10 72. A flowable medium as recited in Claim 62, wherein said phosphor particles comprise $M^2Ga_2S_4$, wherein M^2 is selected from the group consisting of Sr, Ca and combinations thereof.

73. A flowable medium as recited in Claim 62, wherein said phosphor particles comprise yttrium aluminate.

15 74. A flowable medium as recited in Claim 62, wherein said phosphor particles comprise $Y_5(Ga,Al)_5O_{12}$.

75. A flowable medium as recited in Claim 62, wherein said phosphor particles comprise zinc silicate.

20 76. A flowable medium as recited in Claim 62, wherein said phosphor particles comprise yttrium silicate.

77. A flowable medium as recited in Claim 62, wherein said phosphor particles comprise M^1S , wherein M^1 is selected from the group consisting of Sr, Ca and combinations thereof.

25 78. A flowable medium as recited in Claim 62, wherein said phosphor particles comprise barium magnesium aluminate.

79. A flowable medium as recited in Claim 62, wherein said phosphor particles comprise yttrium-gadolinium borate.

80. A flowable medium as recited in Claim 62, wherein said phosphor particles comprise barium aluminate.

81. A flowable medium as recited in Claim 62, wherein said phosphor particles comprise $(Y,Gd)_2O_2S$.

82. A flowable medium as recited in Claim 62, wherein said phosphor particles comprise Gd_2O_2S .

5 83. A paste composition suitable for applying phosphor particles onto a substrate, comprising:

a) a liquid vehicle phase; and

b) a functional phase dispersed throughout said vehicle phase, said functional phase comprising substantially spherical cathodoluminescent phosphor particles having a volume average particle size of not greater than about 5 μm and a particle size distribution wherein at least about 70 volume percent of said particles are not larger than twice said average particle size.

10 84. A paste composition as recited in Claim 83, wherein at least about 90 weight percent of said particles are not larger than twice said average particle size.

15 85. A cathodoluminescent device, comprising:

a) an excitation source; and

b) at least a first layer of cathodoluminescent phosphor particles adapted to be stimulated by said excitation source, wherein said phosphor particles have a volume average particle size of from about 0.1 μm to about 10 μm , a substantially spherical morphology and wherein at least about 70 volume percent of said particles are not larger than about two times said average particle size.

20 86. A cathodoluminescent device as recited in Claim 85, wherein said phosphor particles have a volume average particle size of from about 0.3 μm to about 5 μm .

87. A cathodoluminescent device as recited in Claim 85, wherein said excitation source has an excitation potential of not greater than about 12 kV.

88. A cathodoluminescent device as recited in Claim 85, wherein said excitation source has an excitation potential of at least about 12 kV.

89. A cathodoluminescent device as recited in Claim 85, wherein said excitation source has an excitation potential of at least about 20 kV.

90. A cathodoluminescent device as recited in Claim 85, wherein said phosphor particles comprise $Y_2O_3:Eu$.

5 91. A cathodoluminescent device as recited in Claim 85, wherein said phosphor particles comprise $Y_2O_2S:Eu$.

92. A cathodoluminescent device as recited in Claim 85, wherein said phosphor particles comprise ZnS and a dopant selected from the group consisting of Au, Al, Cu, Ag, Cl and combinations thereof.

10 93. A cathodoluminescent device as recited in Claim 85, wherein said particles comprise $Zn_2SiO_4:Mn$.

94. A cathodoluminescent device as recited in Claim 85, wherein said particles comprise $Y_5(Ga,Al)_5O_{12}:Tb$.

15 95. A cathodoluminescent device as recited in Claim 85, wherein said particles comprise $SrGa_2S_4$ and a dopant selected from the group consisting of Eu and Ce.

96. A cathodoluminescent device as recited in Claim 85, wherein said particles comprise Y_2SiO_5 and a dopant selected from the group consisting of Tb and Ce.

20 97. A cathodoluminescent device as recited in Claim 85, wherein said device is a field emission display.

98. A cathodoluminescent device as recited in Claim 85, wherein said device is a CRT.

25 99. A cathodoluminescent device as recited in Claim 85, wherein said device is a projection CRT.

100. A cathodoluminescent device as recited in Claim 85, wherein said device is a heads-up display.

101. A cathodoluminescent device as recited in Claim 85, wherein said device is a heads-down display.

102. A cathodoluminescent display device, comprising:

a) an excitation source having an excitation potential of not greater than about 12 kV; and

b) at least a first layer of cathodoluminescent phosphor particles adapted to be stimulated by said excitation source, wherein said phosphor particles have a volume average particle size of from about 0.1 μm to about 10 μm , a substantially spherical morphology and wherein at least about 70 volume percent of said particles are not larger than about two times said average particle size.

103. A cathodoluminescent display device as recited in Claim 102, wherein said phosphor particles have a volume average particle size of from about 0.3 μm to about 5 μm .

104. A cathodoluminescent display device as recited in Claim 102, wherein said particles comprise Y_2O_3 and Eu.

105. A cathodoluminescent display device as recited in Claim 102, wherein said particles comprise Zn_2SiO_4 and Mn.

106. A cathodoluminescent device as recited in Claim 102, wherein said particles comprise Y_2SiO_5 and a dopant selected from the group consisting of Tb and Ce.

107. A cathodoluminescent device as recited in Claim 102, wherein said particles comprise SrGa_2S_4 and a dopant selected from the group consisting of Eu and Ce.

108. A cathodoluminescent display device as recited in Claim 102, wherein said device is a field emission display.

109. A cathodoluminescent display device, comprising:

a) an excitation source having an excitation potential of at least about 20 kV; and

b) at least a first layer of cathodoluminescent phosphor particles adapted to be stimulated by said excitation source, wherein said phosphor particles have a volume average particle size of from about 0.1 μm to

about 10 μm , a substantially spherical morphology and wherein at least about 70 volume percent of said particles are not larger than about two times said average particle size.

5 110. A cathodoluminescent display device as recited in Claim 109, wherein said phosphor particles have a volume average particle size of from about 0.3 μm to about 5 μm .

111. A cathodoluminescent display device as recited in Claim 109, wherein at least about 80 volume percent of said particles are not larger than about two times said average particle size.

10 112. A cathodoluminescent display device as recited in Claim 109, wherein said particles comprise Y_2O_3 and Eu.

113. A cathodoluminescent display device as recited in Claim 109, wherein said particles comprise Zn_2SiO_4 and Mn.

15 114. A cathodoluminescent display device as recited in Claim 109, wherein said device is a CRT.

115. A cathodoluminescent display device as recited in Claim 109, wherein said device is a projection CRT.

116. A cathodoluminescent display device as recited in Claim 109, wherein said device is a heads-down display.

20 117. A cathodoluminescent display device as recited in Claim 109, wherein said device is a heads-up display.

118. A field emission display, comprising:

(a) a back plate portion comprising a plurality of electron tip emitters;

25 (b) a transparent front plate portion comprising a layer of phosphor powder comprising substantially spherical cathodoluminescent phosphor particles, wherein said phosphor particles have a volume average particle size of not greater than about 5 μm and a particle size distribution wherein at least about 70 volume percent of said particles are not larger than
30 twice said average particle size.

119. A field emission display as recited in Claim 118, wherein said phosphor particles comprise Y_2O_3 .

120. A field emission display as recited in Claim 118, wherein said phosphor particles comprise Y_2O_3 and from about 4 to about 6 atomic percent Eu.

121. A field emission display as recited in Claim 118, wherein said phosphor particles comprise Y_2SiO_5 .

122. A field emission display as recited in Claim 118, wherein said phosphor particles comprise Y_2SiO_5 and Tb.

123. A field emission display as recited in Claim 118, wherein said phosphor particles comprise Y_2SiO_5 and Ce.

124. A field emission display as recited in Claim 118, wherein said phosphor particles comprise $SrGa_2S_4$ and a dopant selected from the group consisting of Eu and Ce.

125. A field emission display as recited in Claim 118, wherein at least about 90 weight percent of said particles are not larger than twice said average particle size.

126. A field emission display as recited in Claim 118, wherein said phosphor particles have a substantially spherical morphology.

127. A field emission display as recited in Claim 118, wherein said phosphor particles form a pixel layer having an average thickness of not greater than about three times said average particle size.

128. A field emission display as recited in Claim 118, wherein said phosphor particles are coated phosphor particles comprising a coating substantially encapsulating said particles.

129. A CRT display device, comprising:

(a) an excitation source comprising an electron emitter;

(b) a transparent front plate portion comprising a layer of phosphor powder comprising substantially spherical cathodoluminescent phosphor particles, wherein said phosphor particles have a volume average

particle size of not greater than about 5 μm and a particle size distribution wherein at least about 70 volume percent of said particles are not larger than twice said average particle size.

5 130. A CRT display device as recited in Claim 129, wherein said phosphor particles comprise yttrium oxysulfide.

131. A CRT display device as recited in Claim 129, wherein said phosphor particles comprise $\text{Y}_2\text{O}_2\text{S}$ and a dopant selected from the group consisting of Eu, Tb and combinations thereof.

10 132. A CRT display device as recited in Claim 129, wherein said phosphor particles comprise zinc sulfide.

133. A CRT display device as recited in Claim 129, wherein said phosphor particles comprise ZnS and a dopant selected from the group consisting of Au, Al, Ag, Cl, Cu and combinations thereof.

15 134. A CRT display device as recited in Claim 129, wherein said phosphor particles comprise yttrium oxide.

135. A CRT display device as recited in Claim 129, wherein said phosphor particles comprise $\text{Y}_2\text{O}_3:\text{Eu}$.

136. A CRT display device as recited in Claim 129, wherein said phosphor particles comprise strontium thiogallate.

20 137. A CRT display device as recited in Claim 129, wherein said phosphor particles comprise $\text{SrGa}_2\text{S}_4:\text{Eu}$.

138. A CRT display device as recited in Claim 129, wherein said particles comprise Y_2SiO_5 and a dopant selected from the group consisting of Tb and Ce.

25 139. A CRT display device as recited in Claim 129, wherein said particles comprise $\text{Y}_5(\text{Ga,Al})_5\text{O}_{12}:\text{Tb}$.

140. A CRT display device as recited in Claim 129, wherein said volume average particle size is from about 0.3 μm to about 5 μm .

141. A CRT display device as recited in Claim 129, wherein at least about 90 volume percent of said particles are not larger than twice said average particle size.

142. A CRT display device as recited in Claim 129, wherein said phosphor particles form a pixel layer having an average thickness of not greater than about three times said average particle size.

143. A projection CRT display device, comprising:

a) a cathodoluminescent excitation source;

b) a display screen; and

c) a phosphor layer disposed between said excitation source and said display screen, wherein said phosphor layer comprises substantially spherical cathodoluminescent phosphor particles, wherein said phosphor particles have a volume average particle size of not greater than about 5 μm and a particle size distribution wherein at least about 70 volume percent of said particles are not larger than twice said average particle size.

144. A projection CRT as recited in Claim 143, wherein said phosphor particles comprise $\text{Y}_2\text{O}_3:\text{Eu}$.

145. A projection CRT as recited in Claim 143, wherein said phosphor particles comprise $\text{Y}_2\text{O}_2\text{S}:\text{Eu}$.

146. A projection CRT as recited in Claim 143, wherein said phosphor particles comprise Zn_2SiO_4 .

147. A projection CRT as recited in Claim 143, wherein said phosphor particles comprise crystallites having an average crystallite size of at least about 25 nanometers.

148. An electroluminescent device, comprising an electrically insulative substrate, a first electrode supported on said substrate, a phosphor layer disposed over said first electrode, a second electrode disposed over said phosphor layer, and means for applying an electric field between said first and second electrodes, wherein said phosphor layer comprises an electroluminescent phosphor powder comprising electroluminescent phosphor

particles, wherein said phosphor particles are substantially spherical and have a volume average particle size of not greater than about 5 μm .

149. An electroluminescent device as recited in Claim 148, wherein at least about 70 volume percent of said phosphor particles are not larger than twice said average particle size.

150. An electroluminescent device as recited in Claim 148, wherein said volume average particle size is from about 0.3 to about 3 μm .

151. An electroluminescent device as recited in Claim 148, wherein said phosphor particles comprise zinc sulfide.

152. An electroluminescent device as recited in Claim 148, wherein said phosphor particles comprise M^1S , wherein M^1 is selected from the group consisting of Sr, Ca and mixtures thereof.

153. An electroluminescent device as recited in Claim 148, wherein said phosphor particles comprise $\text{M}^2\text{Ga}_2\text{S}_4$, wherein M^2 is selected from the group consisting of Sr, Ca and mixtures thereof.

154. An electroluminescent device as recited in Claim 148, wherein said phosphor particles are coated phosphor particles.

155. An electroluminescent device as recited in Claim 148, wherein said phosphor layer has a thickness that is not greater than about 3 times said average particle size.

156. An electroluminescent device as recited in Claim 148, wherein said device is an electroluminescent lamp.

157. An electroluminescent device as recited in Claim 148, wherein said device is an electroluminescent lamp and wherein said phosphor powder is dispersed in a flexible polymer.

158. An electroluminescent device as recited in Claim 148, wherein said device is an electroluminescent lamp and wherein said phosphor powder is dispersed on a rigid substrate.

159. An electroluminescent device as recited in Claim 148, wherein said device is an electroluminescent display device.

160. An electroluminescent device as recited in Claim 148, wherein said device is an ACEL lamp and wherein said phosphor particles comprise a gallate.

161. A photoluminescent device comprising:

a) an excitation source; and

b) at least a first layer of photoluminescent phosphor particles adapted to be stimulated by said excitation source, wherein said phosphor particles have a volume average particle size of from about 0.1 μm to about 10 μm , a substantially spherical morphology and wherein at least about 70 volume percent of said particles are not larger than two times said average particle size.

162. A photoluminescent device as recited in Claim 161, wherein said phosphor particles have a volume average particle size of from about 0.3 μm to about 5 μm .

163. A photoluminescent device as recited in Claim 161, wherein said excitation source comprises a gas and wherein said gas comprises xenon.

164. A photoluminescent device as recited in Claim 161, wherein said excitation source comprises a gas and wherein said gas comprises mercury.

165. A photoluminescent device as recited in Claim 161, wherein said particles comprise $\text{Y}_2\text{O}_3:\text{Eu}$.

166. A photoluminescent device as recited in Claim 161, wherein said particles comprise $(\text{Y},\text{Gd})\text{BO}_3:\text{Eu}$.

167. A photoluminescent device as recited in Claim 161, wherein said particles comprise $\text{Zn}_2\text{SiO}_4:\text{Mn}$.

168. A photoluminescent device as recited in Claim 161, wherein said particles comprise barium magnesium aluminate and Eu as a dopant.

169. A photoluminescent device as recited in Claim 161, wherein said particles comprise barium aluminate.

170. A photoluminescent device as recited in Claim 161, wherein said particles comprise barium magnesium aluminate and Mn as a dopant.

171. A photoluminescent device as recited in Claim 161, wherein said layer is a substantially uniform layer of photoluminescent phosphor particles, said layer having an average thickness of not greater than about three times said average particle size.

5 172. A photoluminescent device as recited in Claim 161, wherein said device is a plasma display panel.

173. A photoluminescent device as recited in Claim 161, wherein said device is a fluorescent lamp.

10 174. A photoluminescent device as recited in Claim 161, wherein said device is an LCD backlight.

175. A plasma display panel, comprising:

 a) a rear panel comprising a plurality of row electrodes;
 b) a front panel comprising a plurality of column electrodes,
wherein said row electrodes and said column electrodes are in perpendicular
15 relation to form a plurality of addressable x-y coordinates;

 c) a photoluminescent phosphor powder dispersed on a substrate disposed between said electrodes, wherein said phosphor powder comprises particles having a volume average particle size of not greater than about 5 μm and a particle size distribution wherein at least about 70 volume
20 percent of said particles are not larger than two times said average particle size.

176. A plasma display panel as recited in Claim 175, wherein said particles have a substantially spherical morphology.

177. A plasma display panel as recited in Claim 175, wherein said average particle size is from about 0.3 μm to about 5 μm .

25 178. A plasma display panel as recited in Claim 175, wherein said particles have a particle size distribution wherein at least about 80 volume percent of said particles are not larger than about two times said volume average particle size.

179. A plasma display panel as recited in Claim 175, wherein said phosphor particles comprise crystallites having an average crystallite size of at least about 25 nanometers.

180. A plasma display panel as recited in Claim 175, wherein said phosphor powder is dispersed in a substantially uniform layer having an average thickness of not greater than about three times said average particle size.

181. A plasma display panel as recited in Claim 175, wherein said phosphor powder comprises barium magnesium aluminate and Eu as a dopant.

182. A plasma display panel as recited in Claim 175, wherein said phosphor powder comprises $\text{BaMgAl}_{10}\text{O}_{17}$ and from about 8 to about 12 atomic percent Eu and wherein said excitation source comprises xenon gas.

183. A plasma display panel as recited in Claim 175, wherein said phosphor powder comprises barium aluminate and Mn as a dopant.

184. A plasma display panel as recited in Claim 175, wherein said phosphor powder comprises barium aluminate and from about 8 to about 12 atomic percent Mn and wherein said excitation source comprises xenon gas.

185. A plasma display panel as recited in Claim 175, wherein said phosphor powder comprises $\text{Zn}_2\text{SiO}_4\text{:Mn}$.

186. A plasma display panel as recited in Claim 175, wherein said phosphor powder comprises Zn_2SiO_4 and from about 0.05 to about 2 atomic percent Mn and wherein said excitation source comprises xenon gas.

187. A plasma display panel as recited in Claim 175, wherein said phosphor powder comprises $\text{Y}_2\text{O}_3\text{:Eu}$.

188. A plasma display panel as recited in Claim 175, wherein said phosphor powder comprises Y_2O_3 and Eu and wherein said excitation source comprises xenon gas.

189. A plasma display panel as recited in Claim 175, wherein said phosphor powder comprises $(\text{Y,Gd})\text{BO}_3\text{:Eu}$.

190. A plasma display panel as recited in Claim 175, wherein said phosphor powder comprises (Y,Gd)BO₃ and Eu and wherein said excitation source comprises xenon gas.

191. A plasma display panel as recited in Claim 175, wherein said phosphor powder comprises:

- a) first phosphor particles of BaMgAl_xO_y:Eu;
- b) second phosphor particles selected from the group consisting of Zn₂SiO₄:Mn, BaAl_xO_y:Mn and mixtures thereof; and
- c) third phosphor particles selected from the group consisting of Y₂O₃:Eu, (Y,Gd)BO₃:Eu and mixtures thereof.

192. A fluorescent lighting element, comprising:

- a) an excitation source; and
- b) a phosphor powder layer comprising photoluminescent phosphor particles adapted to be stimulated by said excitation source, wherein said phosphor particles have a substantially spherical morphology, a volume average particle size of from about 0.1 μm to about 10 μm and wherein at least about 70 volume percent of said particles are not larger than two times said average particle size.

193. A fluorescent lighting element as recited in Claim 192, wherein said phosphor powder layer comprises a preselected combination of phosphor particles to produce white light.

194. A fluorescent lighting element as recited in Claim 192, wherein said fluorescent lighting element is an LCD backlight.

195. A fluorescent lighting element as recited in Claim 192, wherein said excitation source comprises mercury vapor.

196. A fluorescent lighting element as recited in Claim 192, wherein said phosphor particles comprise Y₂O₃ and Eu.

197. A fluorescent lighting element as recited in Claim 192, wherein said phosphor particles comprise (Y,Gd)BO₃ and Eu.

198. A fluorescent lighting element as recited in Claim 192, wherein said phosphor particles comprise Zn_2SiO_4 and Mn.

199. A fluorescent lighting element as recited in Claim 192, wherein said phosphor particles comprise barium magnesium aluminate and Eu as a dopant.

200. A fluorescent lighting element as recited in Claim 192, wherein said excitation source comprises xenon gas.

201. An article of manufacture comprising indicia for identifying said article, wherein said indicia comprises a layer of photoluminescent phosphor particles in a predetermined pattern, wherein said phosphor particles have a substantially spherical morphology, have a volume average particle size of from about $0.3\ \mu\text{m}$ to about $5\ \mu\text{m}$ and wherein at least about 80 volume percent of said particles are not larger than about two times said average particle size.

202. An article of manufacture as recited in Claim 201, wherein said phosphor particles comprise $\text{Y}_2\text{O}_3:\text{Eu}$.

203. An article of manufacture as recited in Claim 201, wherein said phosphor particles comprise Y_2O_3 and Eu.

204. An article of manufacture as recited in Claim 201, wherein said phosphor particles comprise $(\text{Y},\text{Gd})\text{BO}_3:\text{Eu}$.

205. An article of manufacture as recited in Claim 201, wherein said phosphor particles comprise $(\text{Y},\text{Gd})\text{BO}_3$ and Eu.

206. An article of manufacture as recited in Claim 201, wherein said article is currency.

207. An article of manufacture as recited in Claim 201, wherein said article is a postage stamp.

208. A method for securing a document, comprising the steps of providing a document and applying photoluminescent phosphor particles on said document, wherein said phosphor particles have a volume average particle size of from about $0.1\ \mu\text{m}$ to about $5\ \mu\text{m}$ and a substantially spherical morphology.

209. A method as recited in Claim 208, wherein at least about 70 volume percent of said particles have a particle size that is not greater than about two times said average particle size.

210. A method as recited in Claim 208, wherein said applying step
5 comprises dispersing said phosphor particles in a liquid medium and applying said phosphor particles to said document by ink-jet printing.

211. An method as recited in Claim 208, wherein said phosphor particles comprise $Y_2O_3:Eu$.

212. An method as recited in Claim 208, wherein said phosphor
10 particles comprise Y_2O_3 and from about 6 to about 9 atomic percent Eu.

213. An method as recited in Claim 208, wherein said phosphor particles comprise $(Y,Gd)BO_3:Eu$.

214. A method as recited in Claim 208, wherein said phosphor particles
comprise $(Y,Gd)BO_3$ and from about 14 to about 20 atomic percent Eu.

215. An method as recited in Claim 208, wherein said article is a
15 confidential document.

216. An x-ray imaging device, comprising:

a) an excitation source; and
b) at least a first layer of x-ray phosphor particles adapted to
20 be stimulated by said excitation source, wherein said phosphor particles have a volume average particle size of from about $0.1\ \mu m$ to about $10\ \mu m$, a substantially spherical morphology and wherein at least about 70 volume percent of said particles are not larger than about two times said average particle size.

217. An x-ray device as recited in Claim 216, wherein said phosphor
25 particles have a volume average particle size of from about $0.3\ \mu m$ to about $5\ \mu m$.

218. An x-ray device as recited in Claim 216, wherein said phosphor particles comprise Gd.

219. An x-ray device as recited in Claim 216, wherein said particles
30 comprise Gd_2O_2S .

220. An x-ray device as recited in Claim 216, wherein said particles comprise $\text{Gd}_2\text{O}_2\text{S:Tb}$.

221. An x-ray device as recited in Claim 216, wherein said particles comprise $(\text{Y,Gd})_2\text{SiO}_5$.

5 222. An x-ray device as recited in Claim 216, wherein said particles comprise $(\text{Y,Gd})_2\text{SiO}_5\text{:Tb}$.

223. An x-ray device as recited in Claim 216, wherein said first layer of phosphor particles has an average thickness of not greater than about three times said average particle size.

10 224. An x-ray device as recited in Claim 216, wherein said phosphor particles comprise crystallites having an average crystallite size of at least about 25 nanometers.

225. A method for the production of an x-ray imaging device comprising the steps of:

15 a) providing a support substrate;
b) depositing x-ray phosphor particles on said substrate, wherein said phosphor particles have a substantially spherical morphology and wherein said particles comprise crystallites having an average crystallite size of at least about 25 nanometers.

20 226. A method as recited in Claim 225, wherein said particles have an average particle size of not greater than about $5\text{ }\mu\text{m}$.

227. A method as recited in Claim 225, wherein said depositing step comprises the step of sedimentation of said particles in a liquid vehicle.

25 228. A method as recited in Claim 227, wherein said liquid vehicle comprises isopropanol.

229. A method as recited in Claim 225, wherein said depositing step comprises depositing said phosphor particles using a doctor blade to apply a paste comprising said phosphor particles.

230. A method for the production of a phosphor powder, comprising the steps of:

- a) forming a liquid comprising precursors to a phosphor compound;
- 5 b) generating an aerosol of droplets from said liquid;
- c) pyrolyzing said droplets to remove liquid therefrom and form intermediate precursor particles; and
- d) heating said intermediate precursor particles to form a powder batch of phosphor particles.

10 231. A method as recited in Claim 230, wherein said liquid comprises a particulate precursor.

232. A method as recited in Claim 230, wherein said step of generating an aerosol comprises the step of ultrasonically atomizing said liquid.

15 233. A method as recited in Claim 230, wherein said pyrolyzing step comprises pyrolyzing said droplets at a temperature of at least about 700°C.

234. A method as recited in Claim 230, wherein said heating step comprises heating to a temperature of from about 1100°C to about 1600°C.

20 235. A method as recited in Claim 230, wherein said heating step comprises the step of heating said intermediate precursor particles with agitation.

236. A method as recited in Claim 230, wherein said heating step comprises the step of heating said intermediate precursor particles with sufficient agitation to substantially prevent the formation of hard agglomerates in the phosphor powder.

25 237. A method as recited in Claim 230, wherein said heating step comprises heating said intermediate precursor particles in a rotary kiln.

238. A method as recited in Claim 230, wherein said intermediate precursor particles have an average particle size that is not greater than about 5 μm .

239. A method as recited in Claim 230, wherein no more than about 0.1 weight percent of said phosphor particles are in the form of hard agglomerates.

240. A method as recited in Claim 230, wherein said phosphor particles have an average particle size of not greater than about 5 μm and wherein said particles have not been milled.

241. A method as recited in Claim 230, further comprising the step of adding water to said liquid during processing to maintain the precursor concentration below a predetermined value.

242. A method for the production of a yttrium oxide phosphor powder, comprising the steps of:

a) forming a liquid comprising precursors to a yttrium oxide phosphor compound;

b) generating an aerosol of droplets from said liquid;

c) pyrolyzing said droplets to remove liquid therefrom and form intermediate precursor particles; and

d) heating said intermediate precursor particles to form a powder batch of phosphor particles.

243. A method as recited in Claim 242, wherein said precursors comprise yttrium nitrate.

244. A method as recited in Claim 242, wherein said phosphor particles further comprise Eu.

245. A method as recited in Claim 242, wherein said precursor comprises europium nitrate.

246. A method as recited in Claim 242, wherein said liquid comprises from about 4 to about 6 weight percent precursors.

247. A method as recited in Claim 242, wherein said step of generating an aerosol comprises the step of ultrasonically atomizing said liquid.

248. A method as recited in Claim 242, wherein said pyrolyzing step comprises pyrolyzing said droplets at a temperature of from about 850°C to about 1000°C.

249. A method as recited in Claim 242, wherein said heating step comprises heating to a temperature of from about 1350°C to about 1500°C.

250. A method as recited in Claim 242, wherein said heating step comprises the step of heating said intermediate precursor particles with agitation.

251. A method as recited in Claim 242, wherein said heating step comprises the step of heating said intermediate precursor particles with sufficient agitation to substantially prevent the formation of hard agglomerates in the phosphor powder.

252. A method as recited in Claim 242, wherein said heating step comprises heating said intermediate precursor particles in a rotary kiln.

253. A method as recited in Claim 242, wherein said intermediate precursor particles have an average particle size that is not greater than about 5 μm .

254. A method as recited in Claim 242, wherein no more than about 0.1 weight percent of said phosphor particles are in the form of hard agglomerates.

255. A method as recited in Claim 242, wherein said phosphor particles have an average particle size of not greater than about 5 μm and wherein said particles have not been milled.

256. A method as recited in Claim 242, further comprising the step of adding water during said step of generating an aerosol to maintain the precursor concentration below a predetermined value.

257. A method for the production of a zinc silicate phosphor powder, comprising the steps of:

- a) forming a liquid comprising precursors to a zinc silicate phosphor compound;
- b) generating an aerosol of droplets from said liquid;
- c) pyrolyzing said droplets to remove liquid therefrom and form intermediate precursor particles; and

d) heating said intermediate precursor particles to form a powder batch of phosphor particles.

258. A method as recited in Claim 257, wherein said precursors comprise particulate silica.

5 259. A method as recited in Claim 257, wherein said precursors comprise zinc nitrate.

260. A method as recited in Claim 257, wherein said precursors comprise an excess of silica.

10 261. A method as recited in Claim 257, wherein said precursors comprise at least about 10 atomic percent excess silica.

262. A method as recited in Claim 257, wherein said phosphor particles further comprise Mn.

263. A method as recited in Claim 257, wherein said precursor comprises manganese nitrate.

15 264. A method as recited in Claim 257, wherein said step of generating an aerosol comprises the step of ultrasonically atomizing said liquid.

265. A method as recited in Claim 257, wherein said pyrolyzing step comprises pyrolyzing said droplets at a temperature of from about 850°C to about 1000°C.

20 266. A method as recited in Claim 257, wherein said heating step comprises heating to a temperature of from about 1100°C to about 1200°C.

267. A method as recited in Claim 257, wherein said heating step comprises the step of heating said intermediate precursor particles with agitation.

25 268. A method as recited in Claim 257, wherein said heating step comprises the step of heating said intermediate precursor particles with sufficient agitation to substantially prevent the formation of hard agglomerates in the phosphor powder.

30 269. A method as recited in Claim 257, wherein said heating step comprises heating said intermediate precursor particles in a rotary kiln.

270. A method as recited in Claim 257, wherein said intermediate precursor particles have an average particle size that is not greater than about 5 μm .

5 271. A method as recited in Claim 257, wherein no more than about 0.1 weight percent of said phosphor particles are in the form of hard agglomerates.

272. A method as recited in Claim 257, wherein said phosphor particles have an average particle size of not greater than about 5 μm and wherein said particles have not been milled.

10 273. A method as recited in Claim 257, further comprising the step of adding water during said step of generating an aerosol to maintain the precursor concentration below a predetermined value.

274. A method for the production of a barium magnesium aluminate phosphor powder batch, comprising the steps of:

- 15 a) forming a liquid comprising precursors to a barium magnesium aluminate phosphor compound;
- b) generating an aerosol of droplets from said liquid;
- c) pyrolyzing said droplets to remove liquid therefrom and at least partially react said precursors to form intermediate precursor particles; and
- 20 d) heating said intermediate precursor particles to form a powder batch of phosphor particles.

275. A method as recited in Claim 274, wherein said step of generating an aerosol comprises the step of ultrasonically atomizing said liquid.

276. A method as recited in Claim 274, wherein said liquid comprises particulate alumina.

25 277. A method as recited in Claim 274, wherein said liquid comprises excess alumina.

278. A method as recited in Claim 274, wherein said liquid comprises at least about 20 atomic percent excess alumina.

30 279. A method as recited in Claim 274, wherein said liquid comprises barium nitrate.

280. A method as recited in Claim 274, wherein said liquid comprises magnesium nitrate.

281. A method as recited in Claim 274, wherein said phosphor particles further comprise a dopant selected from the group consisting of Eu and Mn.

5 282. A method as recited in Claim 281, wherein said liquid comprises a precursor selected from the group consisting of europium nitrate and manganese nitrate.

283. A method as recited in Claim 274, wherein said liquid comprises from about 5 to 10 weight percent precursors.

10 284. A method as recited in Claim 274, wherein said intermediate precursor particles have an average size of not greater than about 5 μm .

285. A method as recited in Claim 274, wherein said pyrolyzing step comprises pyrolyzing said droplets at a reaction temperature of from about 750°C to about 950°C.

15 286. A method as recited in Claim 274, wherein said heating step comprise heating said intermediate precursor particles to a temperature of from about 1200°C to about 1650°C.

20 287. A method as recited in Claim 274, wherein said heating step comprises heating said intermediate precursor particles in a reducing gas composition.

25 288. A method as recited in Claim 274, wherein said heating step comprises a first heating step wherein said intermediate phosphor particles are heated in air at a temperature of from about 1200°C to about 1600°C to crystallize the $\text{BaMgAl}_x\text{O}_y$ phase and a second heating step comprising heating in a hydrogen-containing atmosphere at a temperature of from about 1400°C to about 1650°C to reduce Eu^{3+} to Eu^{2+} .

30 289. A method as recited in Claim 274, wherein said heating step comprises the step of heating said intermediate precursor particles while applying sufficient agitation to substantially prevent the formation of hard agglomerates.

290. A method as recited in Claim 274, wherein said heating step comprises the step of heating in a rotary kiln.

291. A method as recited in Claim 274, wherein said phosphor particles have a weight average particle size of not greater than about 5 μm .

5 292. A method for the production of a barium aluminate phosphor powder batch, comprising the steps of:

a) forming a liquid comprising precursors to a barium aluminate phosphor compound;

b) generating an aerosol of droplets from said liquid;

10 c) pyrolyzing said droplets to remove liquid therefrom and at least partially react said precursors to form intermediate precursor particles; and

d) heating said intermediate precursor particles to form a powder batch of phosphor particles.

15 293. A method as recited in Claim 292, wherein said step of generating an aerosol comprises the step of ultrasonically atomizing said liquid.

294. A method as recited in Claim 292, wherein said liquid comprises particulate alumina.

295. A method as recited in Claim 292, wherein said liquid comprises excess alumina.

20 296. A method as recited in Claim 292, wherein said liquid comprises at least about 20 atomic percent excess alumina.

297. A method as recited in Claim 292, wherein said liquid comprises barium nitrate.

25 298. A method as recited in Claim 292, wherein said phosphor particles further comprise Mn as a dopant.

299. A method as recited in Claim 292, wherein said liquid comprises a manganese nitrate precursor.

300. A method as recited in Claim 292, wherein said liquid comprises from about 6 to 10 weight percent precursors.

301. A method as recited in Claim 292, wherein said intermediate precursor particles have an average size of not greater than about 5 μm .

302. A method as recited in Claim 292, wherein said pyrolyzing step comprises pyrolyzing said droplets at a reaction temperature of from about 750°C to about 950°C.

303. A method as recited in Claim 292, wherein said heating step comprise heating said intermediate precursor particles to a temperature of from about 1300°C to about 1600°C.

304. A method as recited in Claim 292, wherein said heating step comprises the step of heating said intermediate compound while applying sufficient agitation to substantially prevent the formation of hard agglomerates.

305. A method as recited in Claim 292, wherein said heating step comprises the step of heating in a rotary kiln.

306. A method as recited in Claim 292, wherein said phosphor particles have a weight average particle size of not greater than about 5 μm .

307. A method for the production of a yttrium-gadolinium borate phosphor powder batch, comprising the steps of:

a) forming a liquid comprising precursors to a yttrium-gadolinium borate phosphor compound;

b) generating an aerosol of droplets from said liquid;
c) pyrolyzing said droplets to remove liquid therefrom and at least partially react said precursors to form intermediate precursor particles; and

d) heating said intermediate precursor particles to form a powder batch of phosphor particles.

308. A method as recited in Claim 307, wherein said step of generating an aerosol comprises the step of ultrasonically atomizing said liquid.

309. A method as recited in Claim 307, wherein said liquid comprises boric acid.

310. A method as recited in Claim 307, wherein said liquid comprises an excess of boric acid.

311. A method as recited in Claim 307, wherein said phosphor particles comprise Eu.

312. A method as recited in Claim 311, wherein said liquid comprises europium nitrate.

5 313. A method as recited in Claim 307, wherein said pyrolyzing step comprises pyrolyzing said droplets at a reaction temperature of from about 900°C to about 950°C.

314. A method as recited in Claim 307, wherein said heating step comprises heating said intermediate precursor particles to a temperature of from about 1300°C to about 1400°C.

315. A method as recited in Claim 307, wherein said heating step comprises heating said intermediate precursor particles in air.

316. A method as recited in Claim 307, wherein said heating step comprises the step of heating said intermediate precursor particles while applying sufficient agitation to substantially prevent the formation of hard agglomerates.

317. A method as recited in Claim 307, wherein said heating step comprises the step of heating said intermediate precursor particles in a rotary kiln.

20 318. A method as recited in Claim 307, wherein said phosphor particles have an average particle size of not greater than about 5 μm .

319. A method for the production of a x-ray phosphor powder, comprising the steps of:

- 25 a) forming a liquid comprising precursors to a x-ray phosphor compound;
- b) generating an aerosol of droplets from said liquid;
- c) pyrolyzing said droplets to remove liquid therefrom and at least partially react said precursors to form intermediate precursor particles; and
- 30 d) heating said intermediate precursor particles to form a powder batch of phosphor particles.

320. A method as recited in Claim 319, wherein said phosphor particles comprise Gd.

321. A method as recited in Claim 319, wherein said phosphor particles comprise $\text{Gd}_2\text{O}_2\text{S}$.

5 322. A method as recited in Claim 319, wherein said phosphor particles comprise $\text{Gd}_2\text{O}_2\text{S:Tb}$.

323. A method as recited in Claim 319, wherein said phosphor particles comprise $(\text{Y,Gd})_2\text{O}_2\text{S}$.

10 324. A method as recited in Claim 319, wherein said phosphor particles comprise $(\text{Y,Gd})_2\text{O}_2\text{S:Tb}$.

325. A method as recited in Claim 319, wherein said phosphor particles comprise $(\text{Y,Gd})_2\text{O}_2\text{S:Tb, Ce}$.

326. A method as recited in Claim 319, wherein said step of generating an aerosol comprises ultrasonically atomizing said liquid.

15 327. A method as recited in Claim 319, wherein said pyrolyzing step comprises pyrolyzing said droplets at a reaction temperature of from about 700°C to about 950°C .

20 328. A method as recited in Claim 319, wherein said heating step comprises the step of heating said intermediate precursor particles to a temperature of from about 1200°C to about 1500°C .

329. A method as recited in Claim 319, wherein said heating step comprises the step of heating said intermediate precursor particles with agitation.

25 330. A method as recited in Claim 319, wherein said heating step comprises the step of heating said intermediate precursor particles with sufficient agitation to substantially prevent the formation of hard agglomerates in the phosphor powder.

331. A method as recited in Claim 319, wherein said heating step comprises heating said intermediate precursor particles in a rotary kiln.

332. A method as recited in Claim 319, wherein said intermediate precursor particles have an average particle size that is not greater than about 5 μm .

333. A method as recited in Claim 319, wherein no greater than about 0.1 weight percent of said phosphor particles are in the form of hard agglomerates.

334. A method as recited in Claim 319, wherein said phosphor particles have an average size of not greater than about 5 μm and wherein said particles have not been milled.

335. A light emitting diode device, comprising:
a light emitting diode adapted to emit light of a first wavelength; and
a phosphor powder layer disposed above said light emitting diode and adapted to emit light of a second wavelength when stimulated by light of said first wavelength, wherein said phosphor powder layer comprises phosphor particles having a volume average particle size of not greater than about 10 μm and having a substantially spherical morphology.

A light emitting diode device as recited in Claim 335, wherein said phosphor powder layer has an average thickness of not greater than about 3 times said average particle size.

A light emitting diode device as recited in Claim 335, wherein said phosphor particles comprise yttrium aluminate.

A light emitting diode device as recited in Claim 335, where said phosphor particles comprise $\text{Y}_3\text{Al}_5\text{O}_{12}$ and Ce as a dopant.

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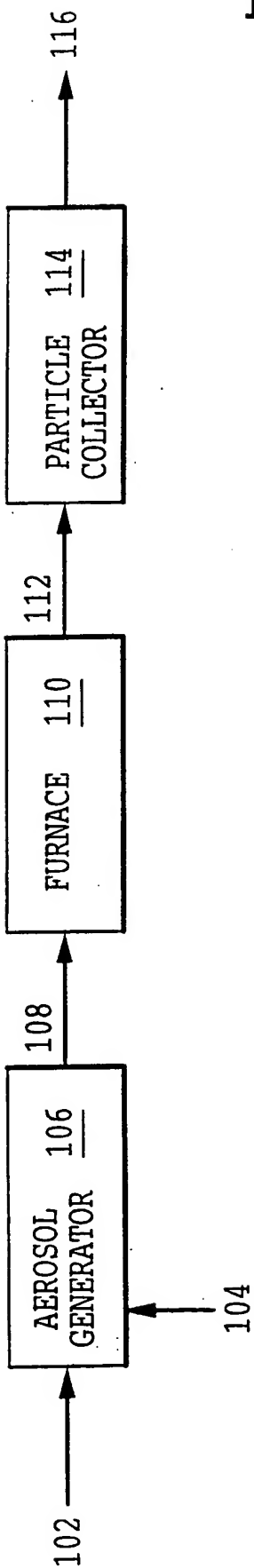


FIG.1

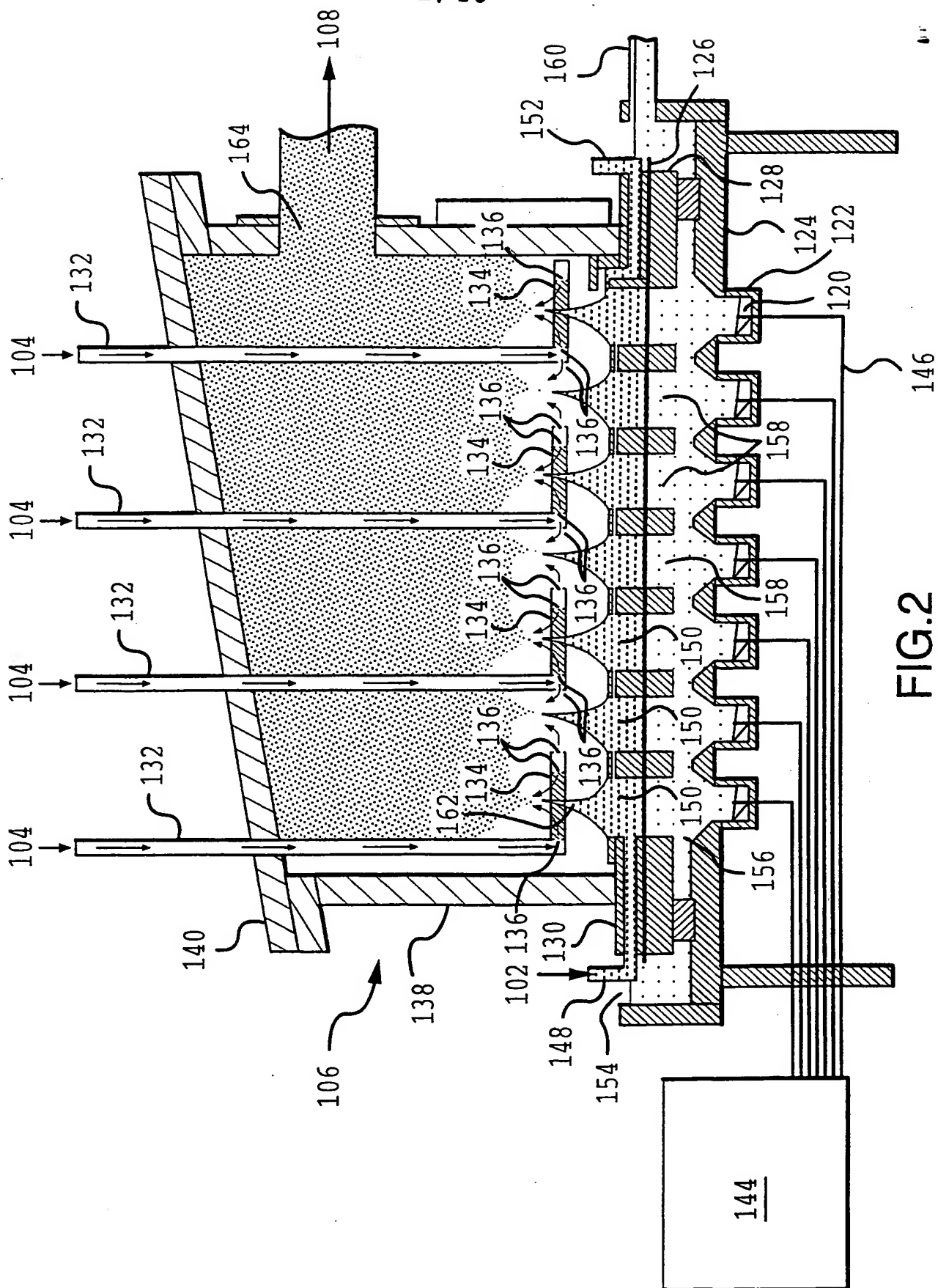


FIG. 2

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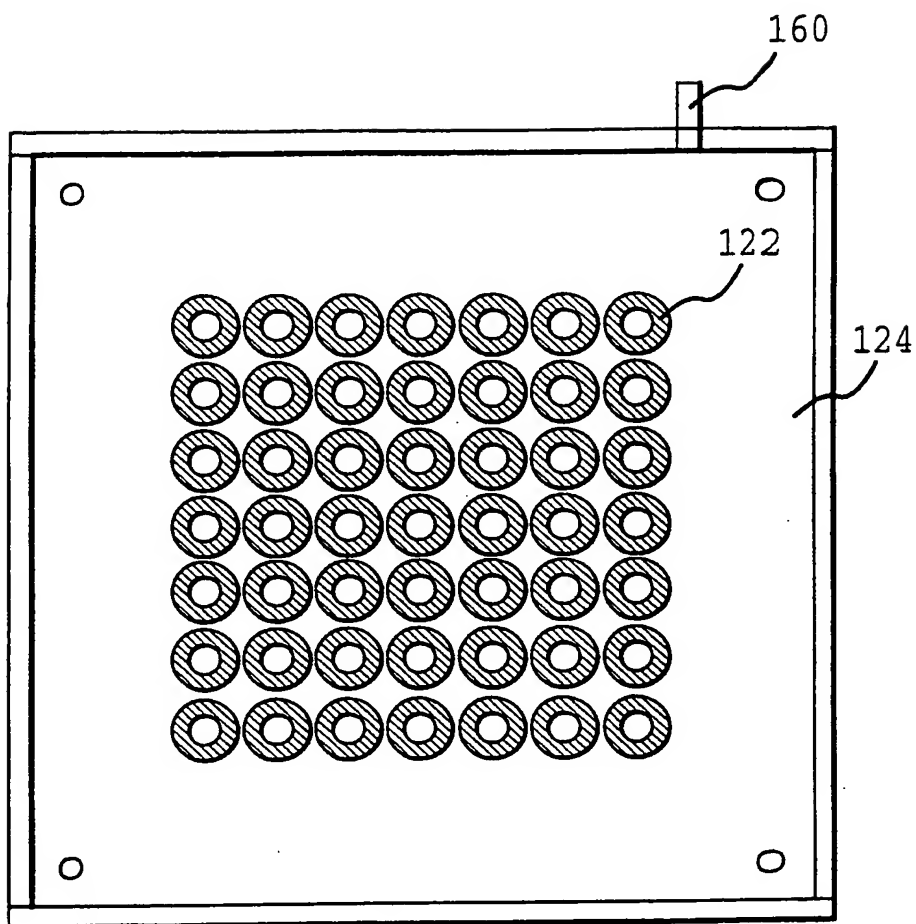


FIG.3

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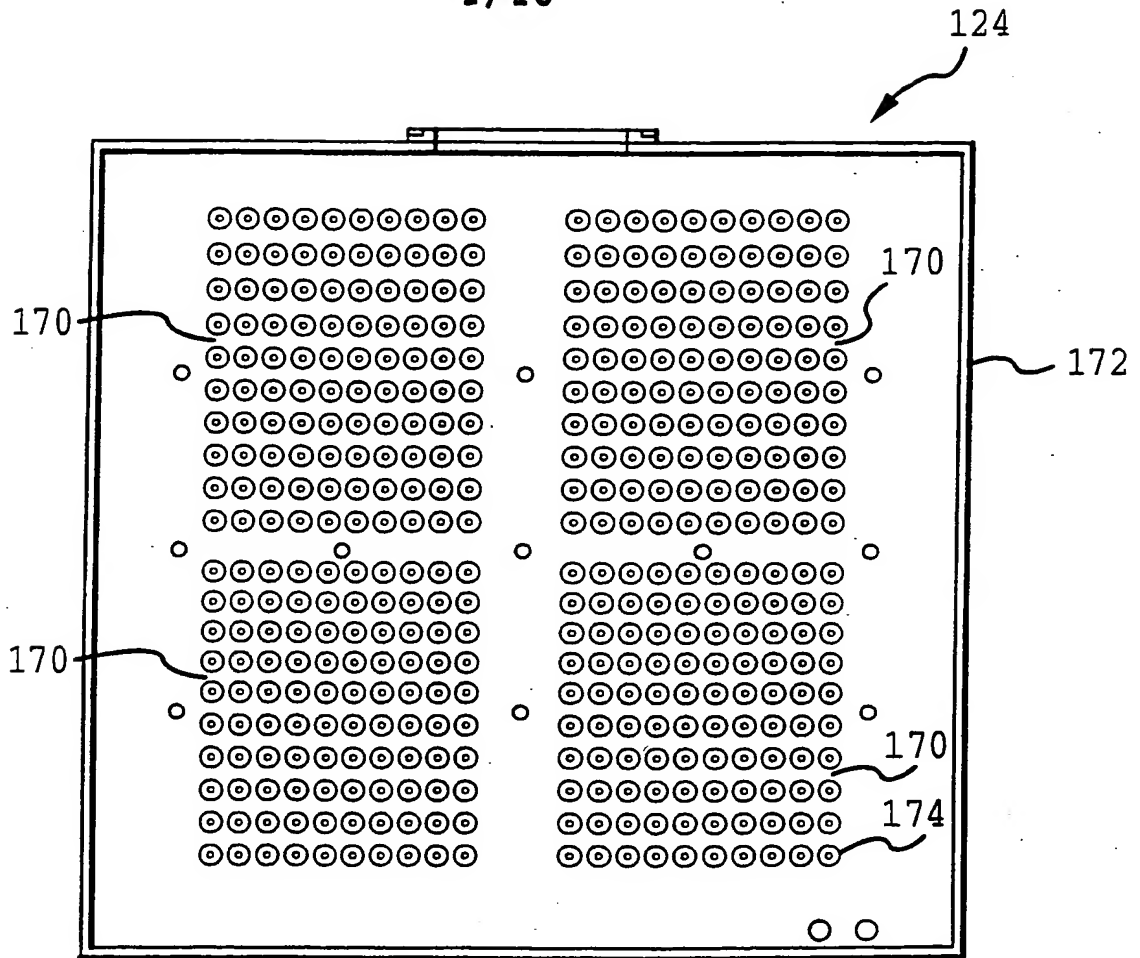


FIG. 4

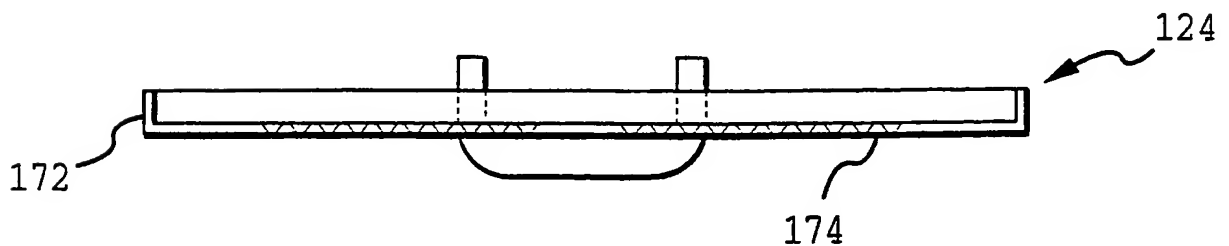


FIG. 5

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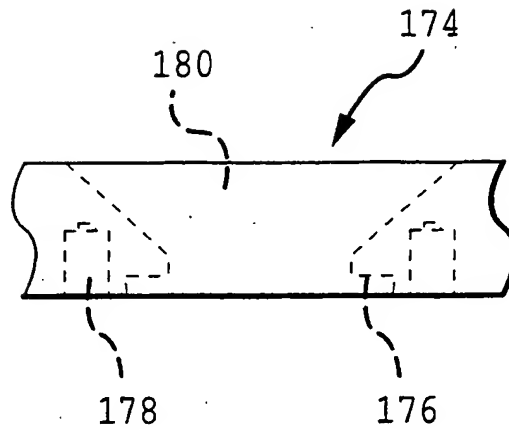


FIG. 6

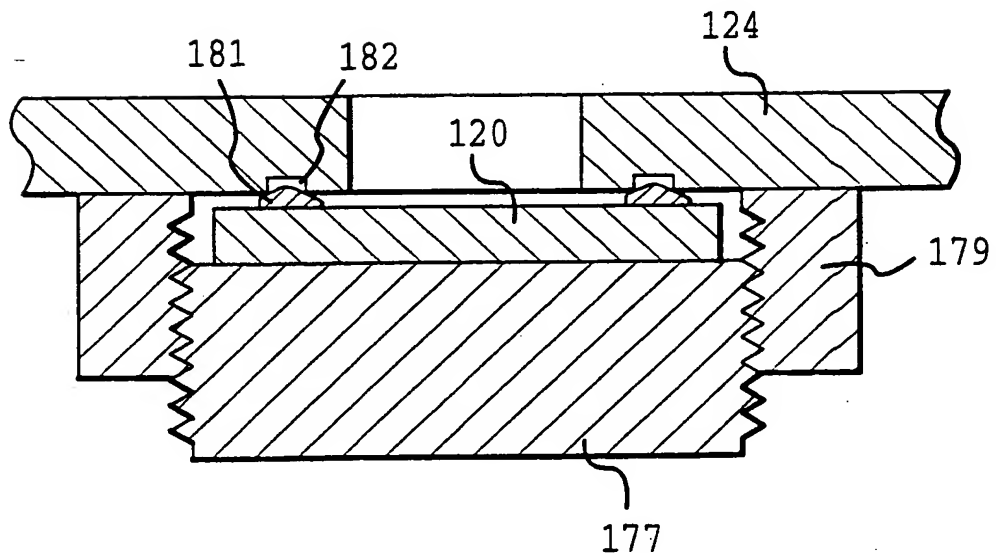


FIG. 7

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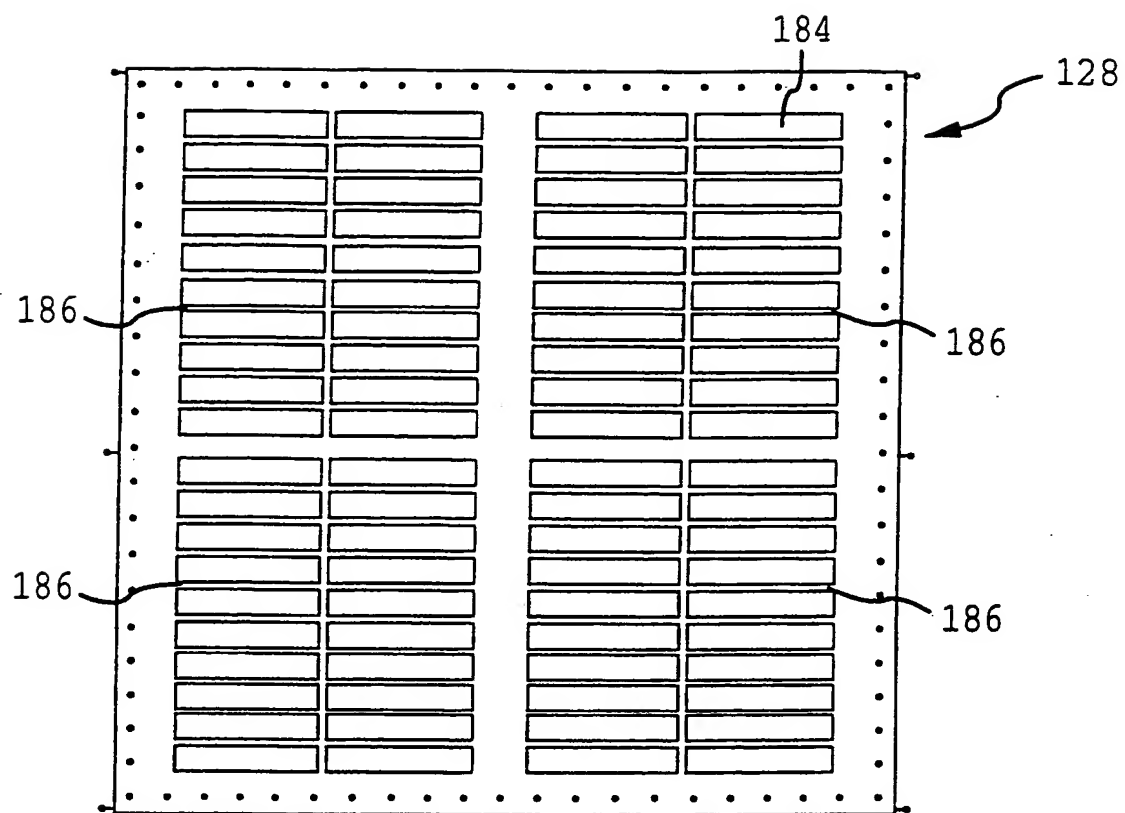


FIG. 8

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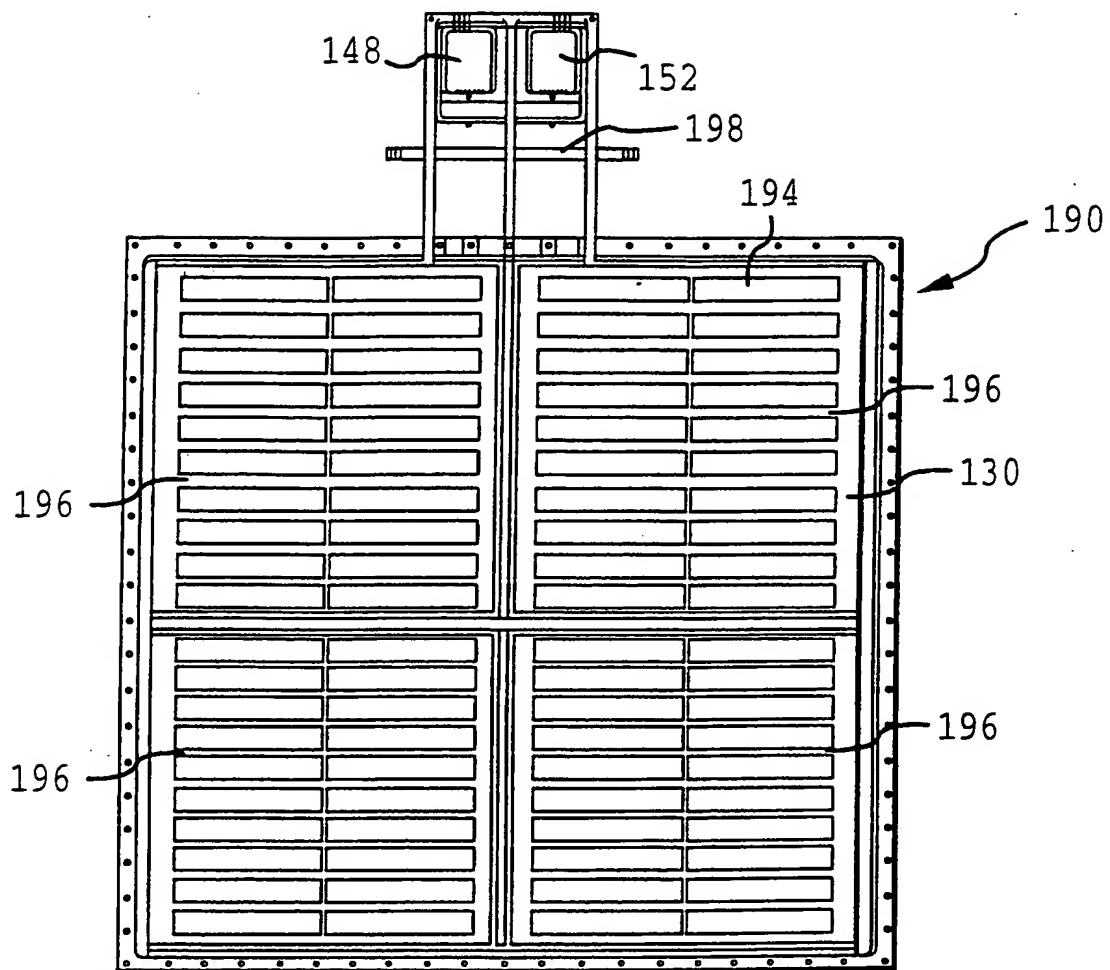


FIG. 9

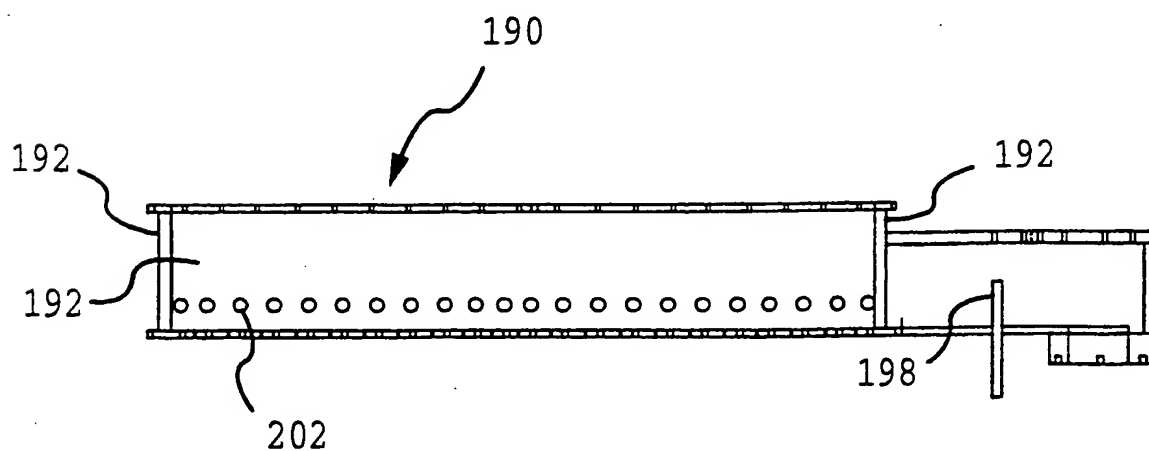


FIG. 10

SUBSTITUTE SHEET (RULE 26)

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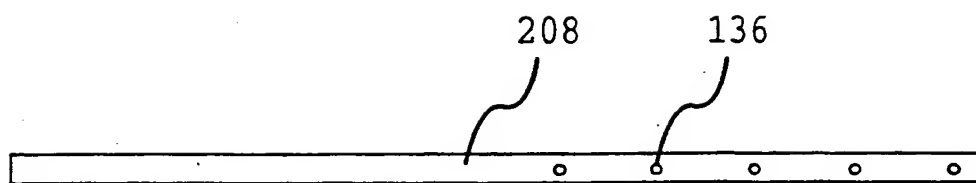


FIG.11

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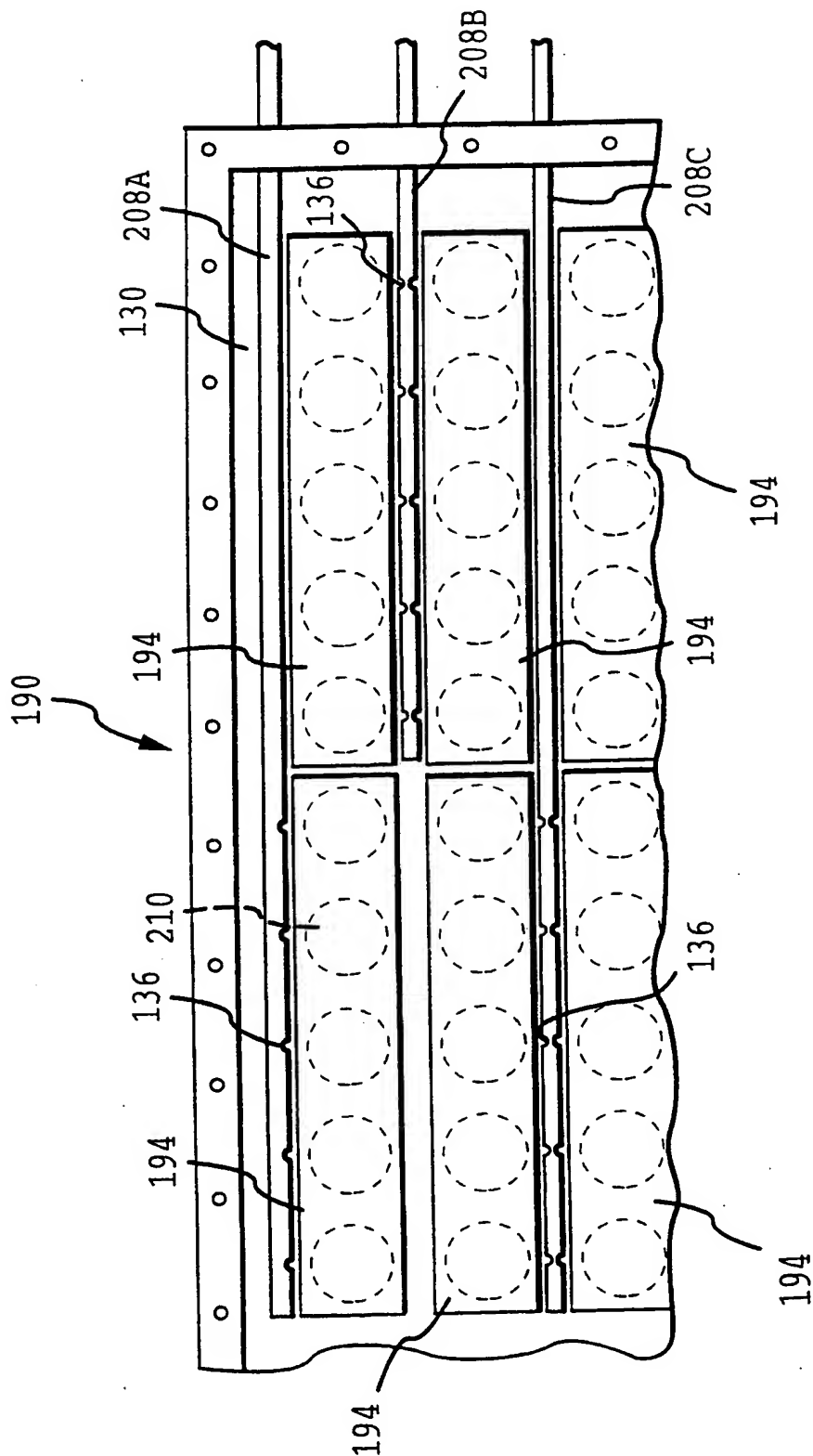


FIG.12

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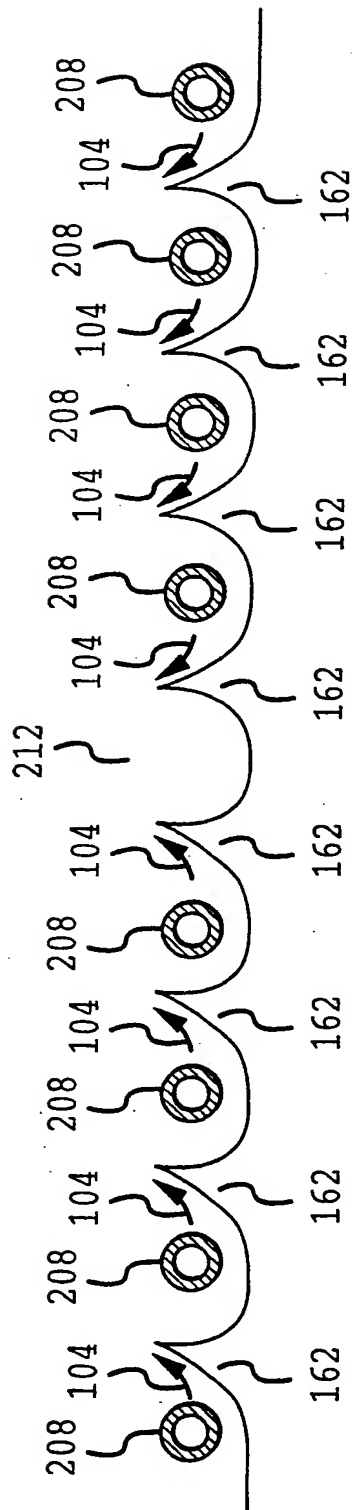


FIG. 13

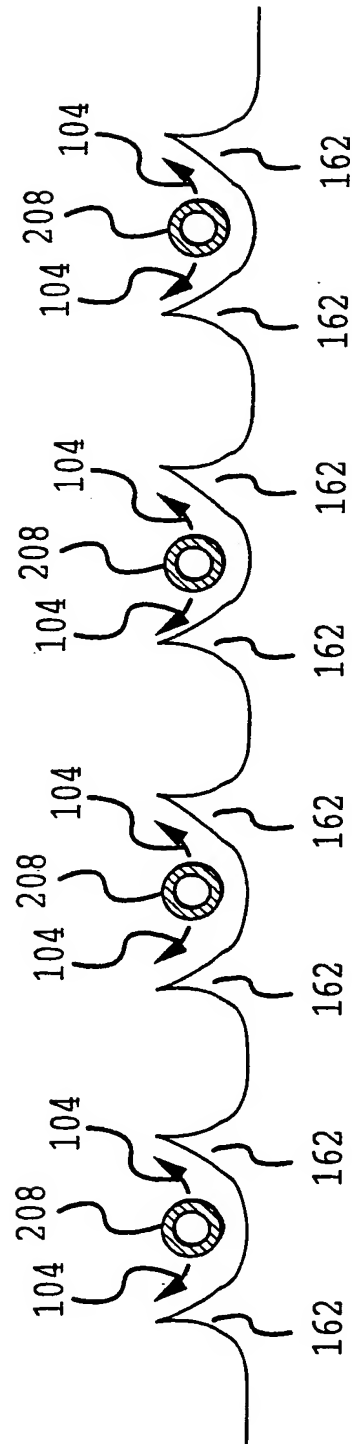


FIG. 14

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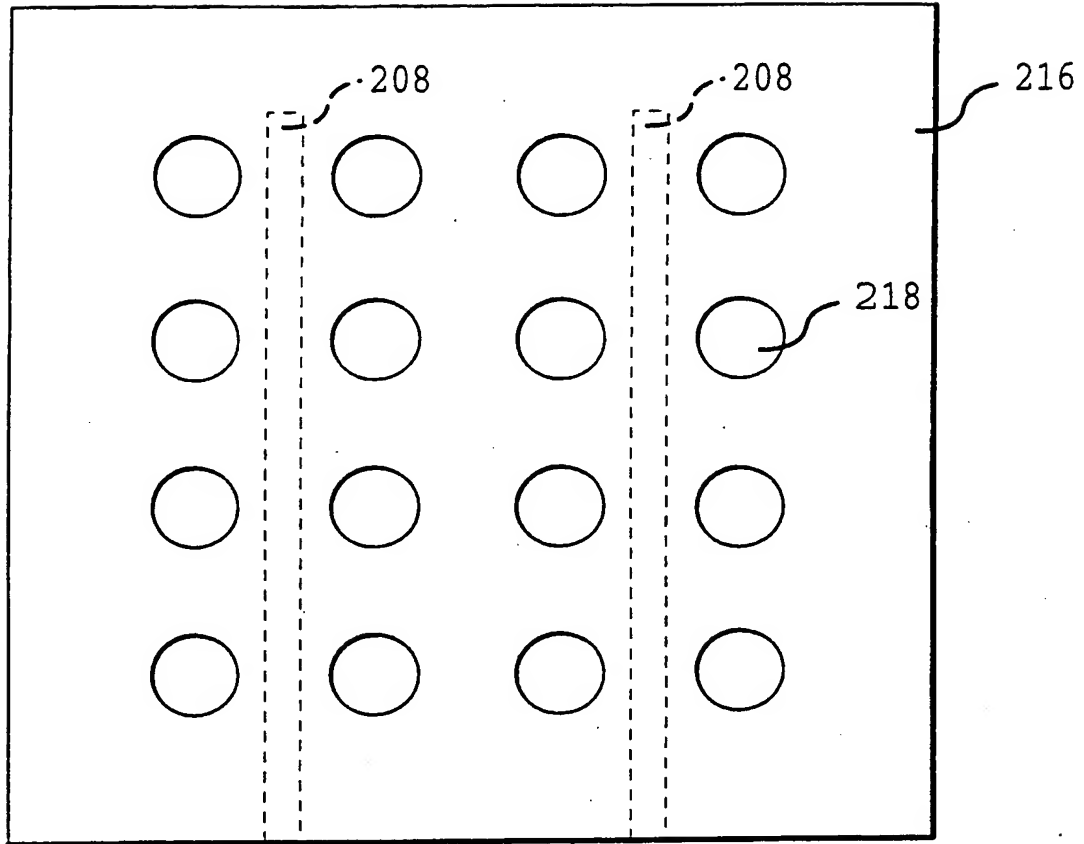


FIG. 15

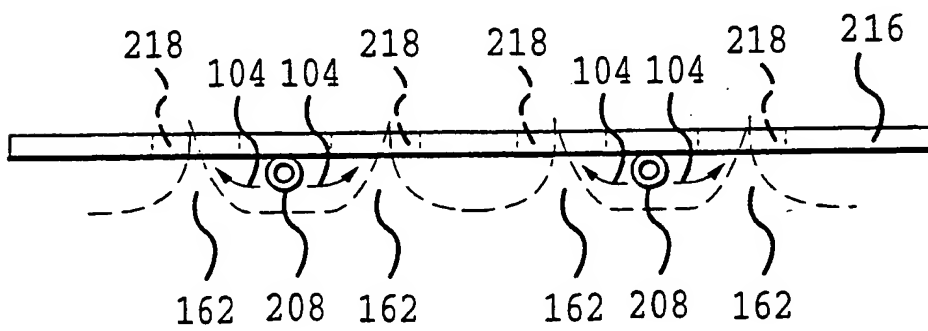


FIG. 16

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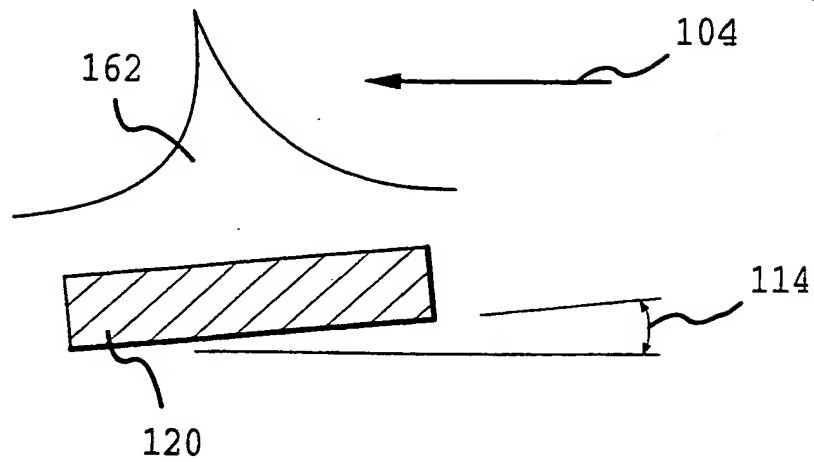


FIG.17

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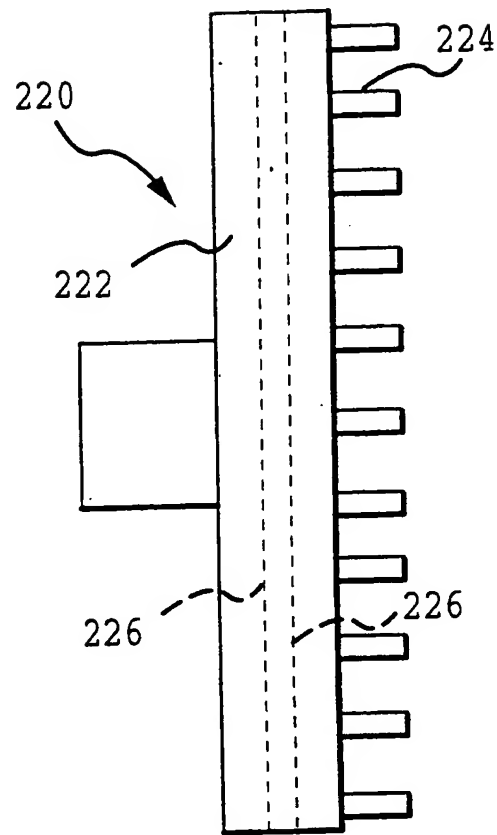


FIG.18

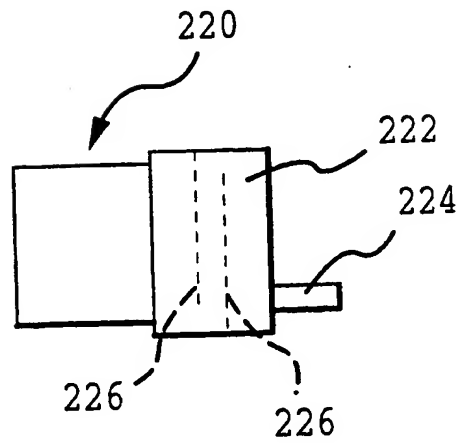


FIG.19

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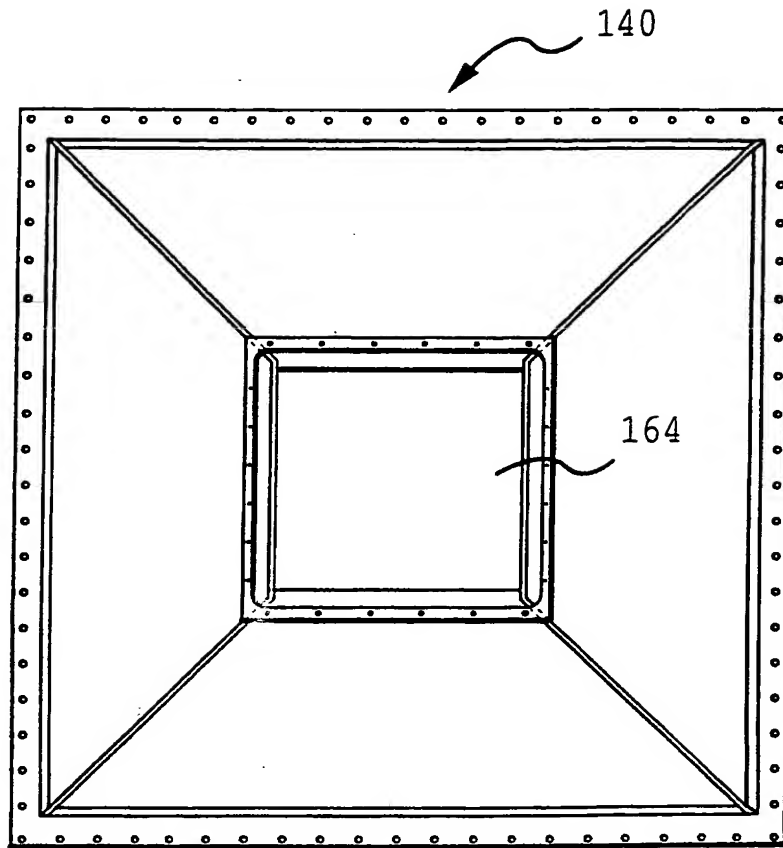


FIG.20

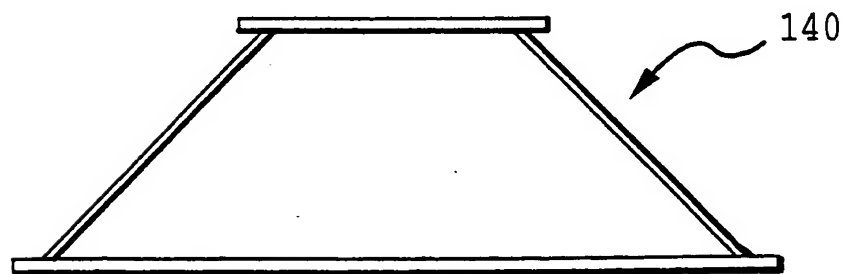


FIG.21

SUBSTITUTE SHEET (RULE 26)

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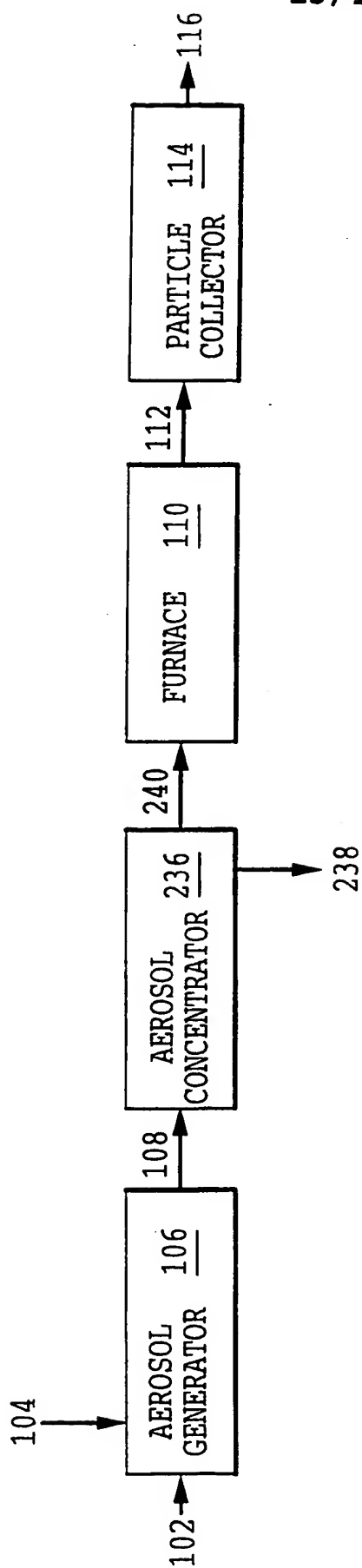


FIG.22

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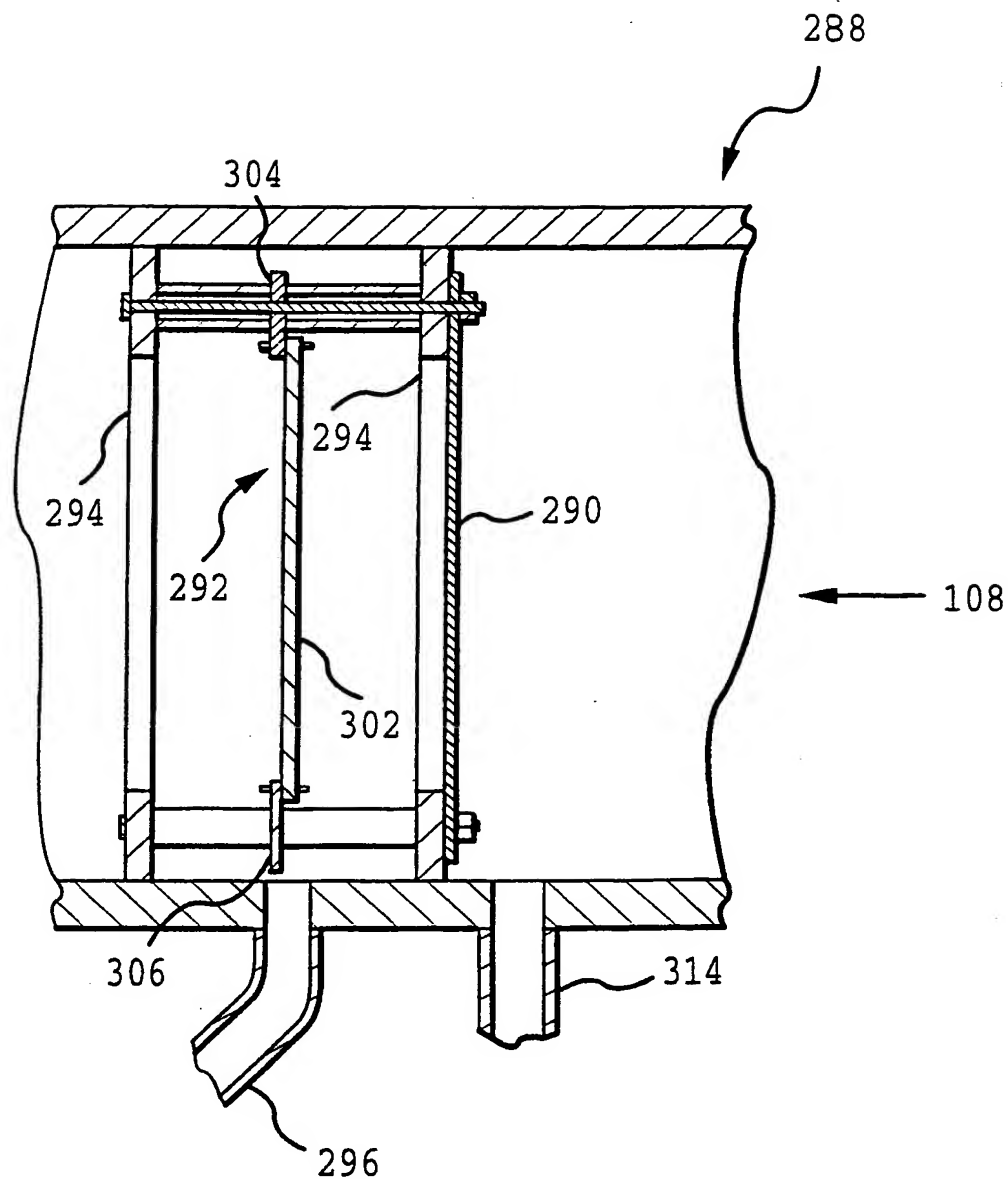


FIG.23

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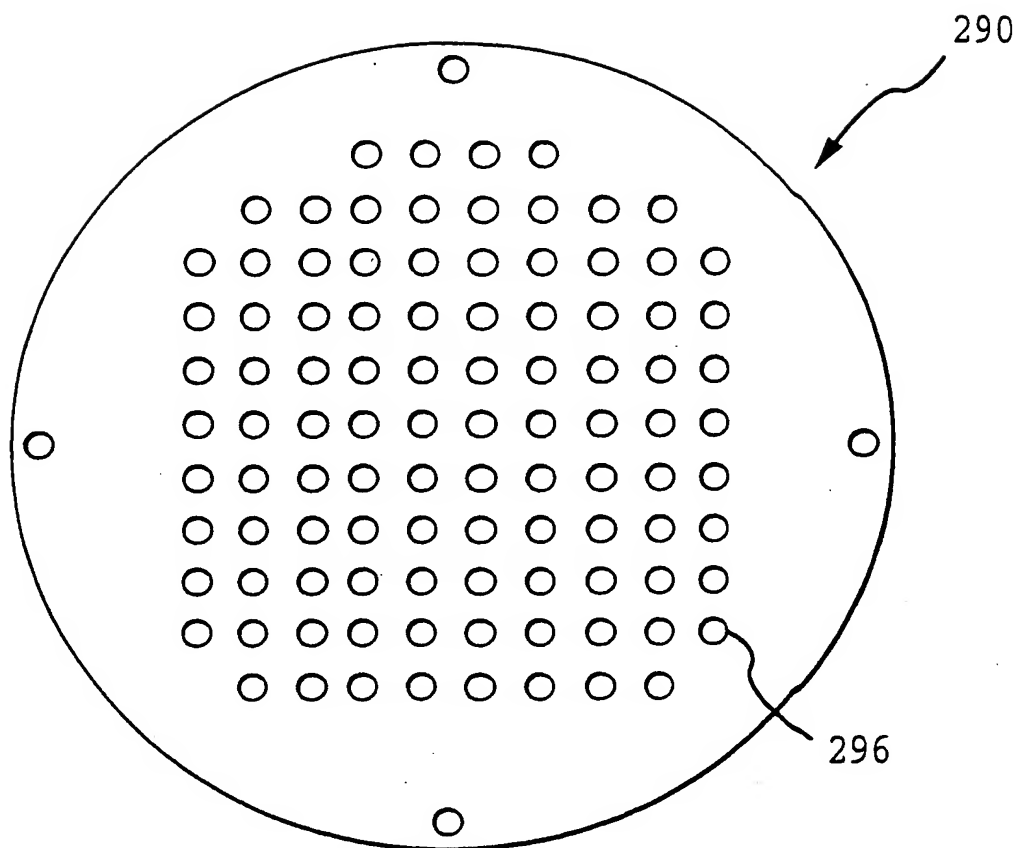


FIG.24

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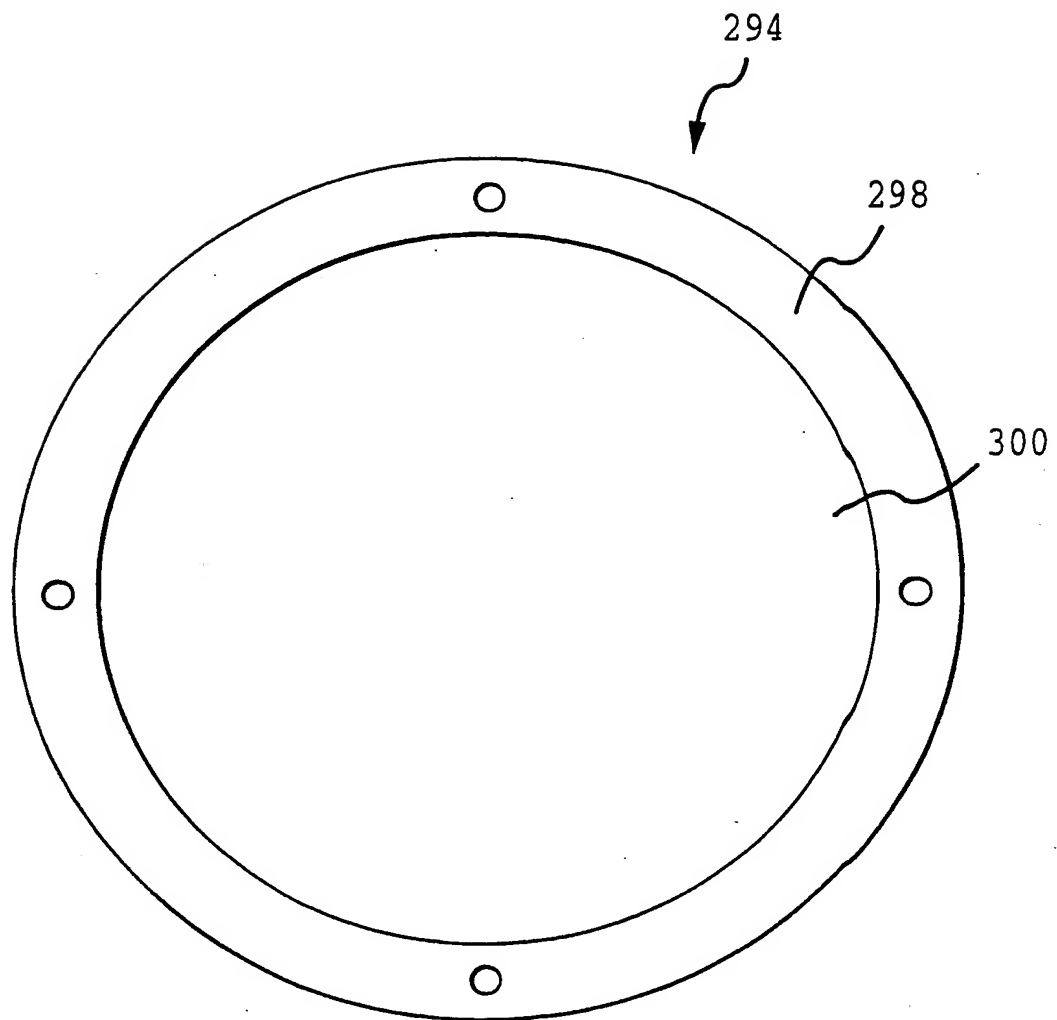


FIG.25

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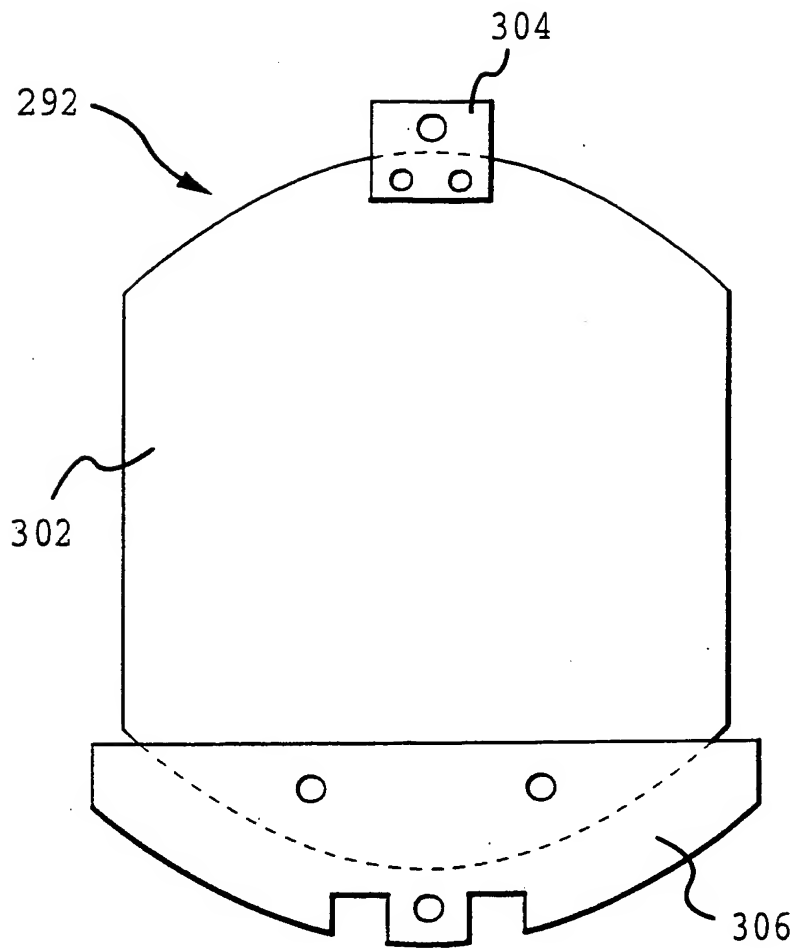


FIG. 26

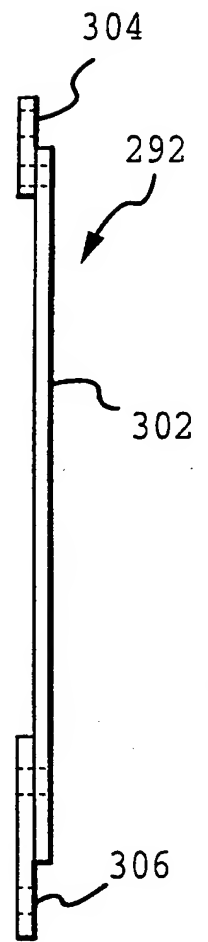


FIG. 27

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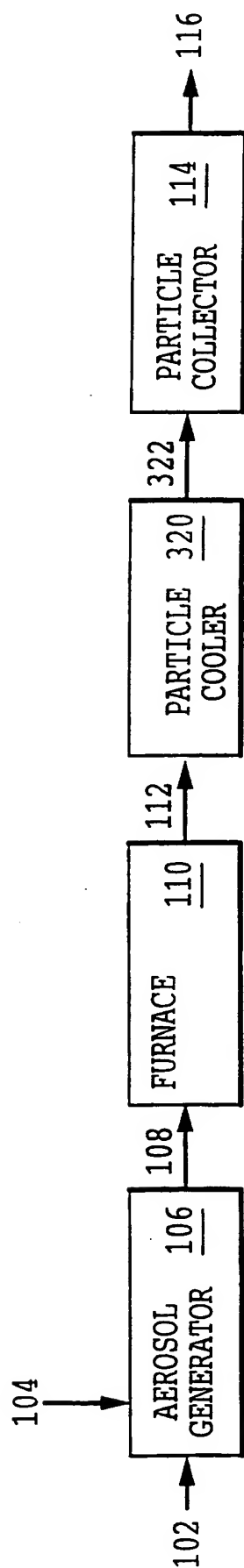


FIG.28

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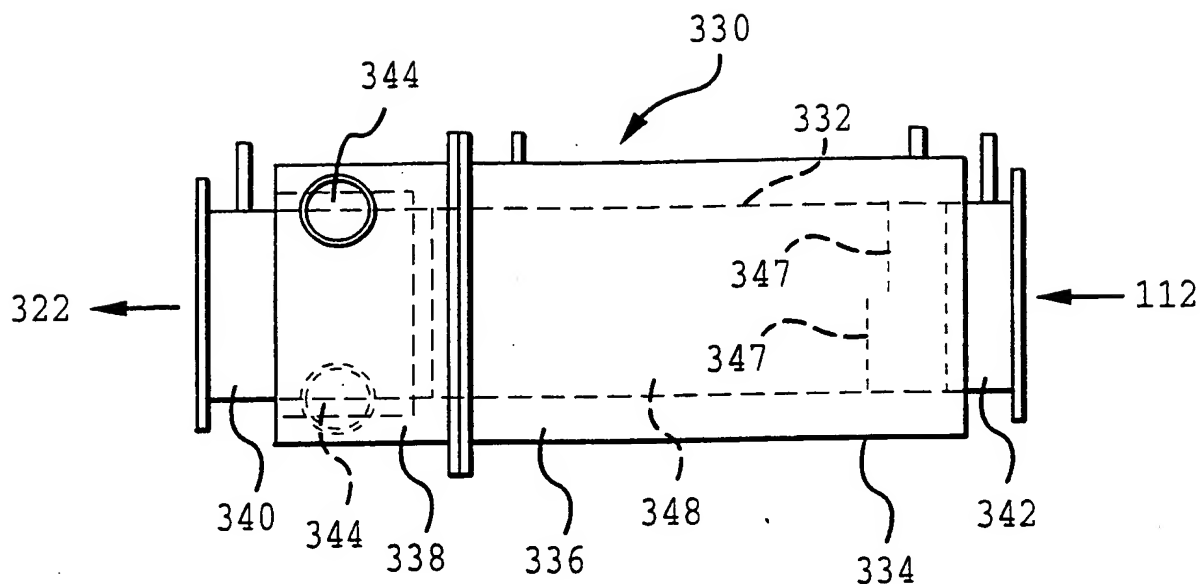


FIG. 29

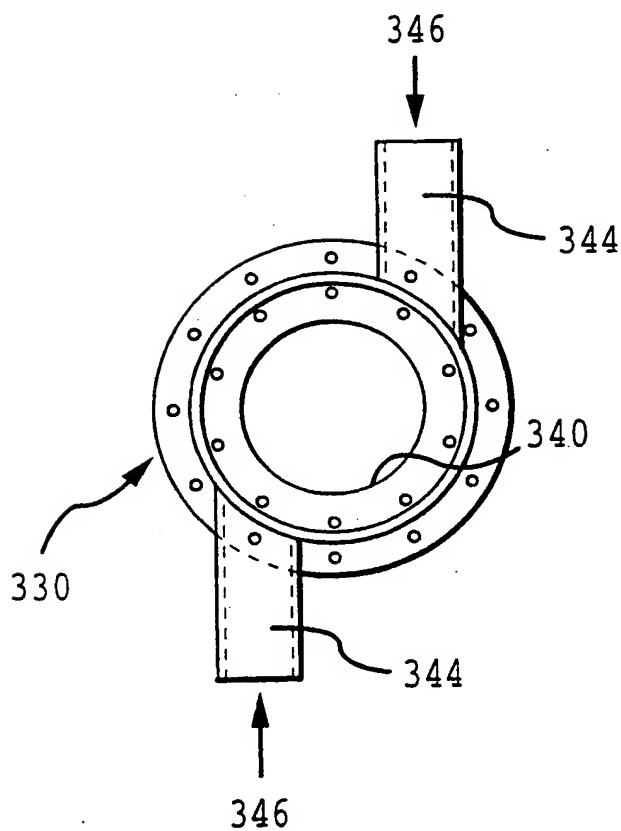


FIG. 30

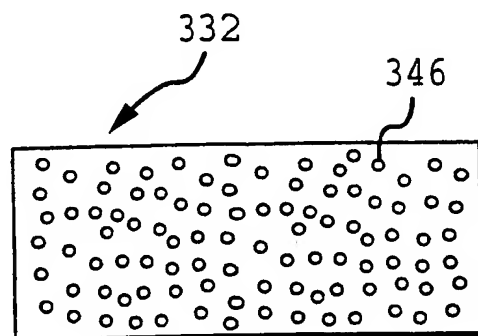


FIG. 31

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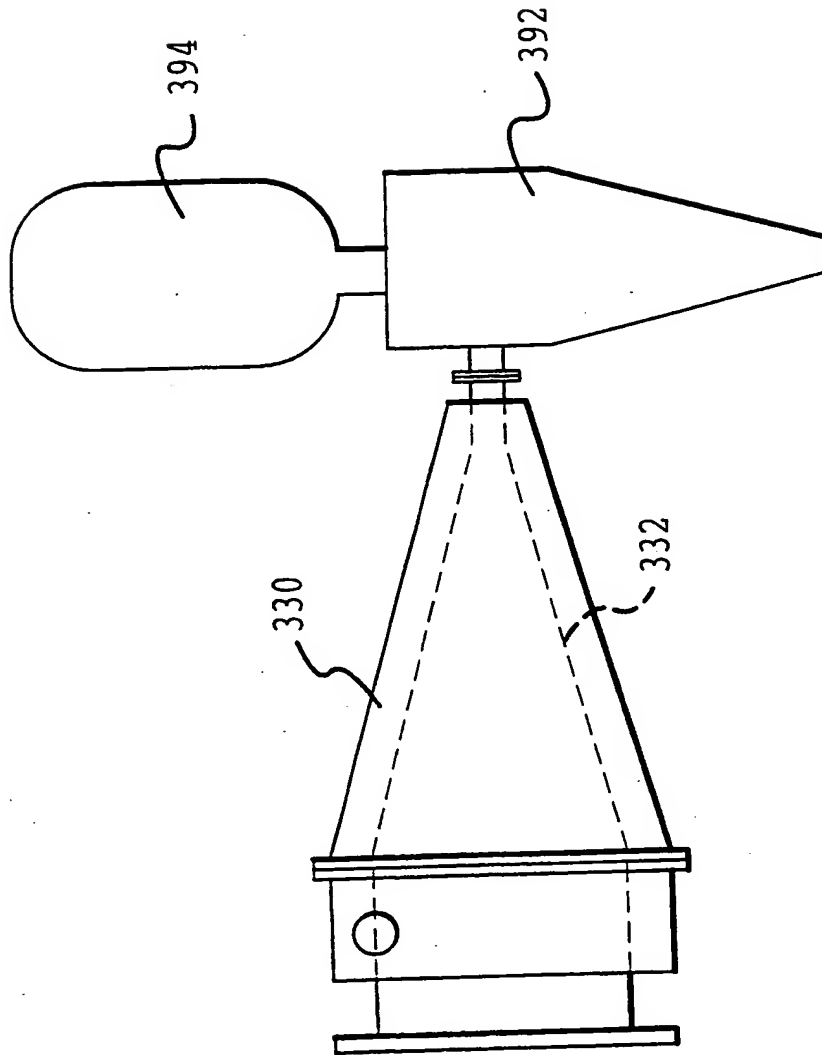


FIG. 32

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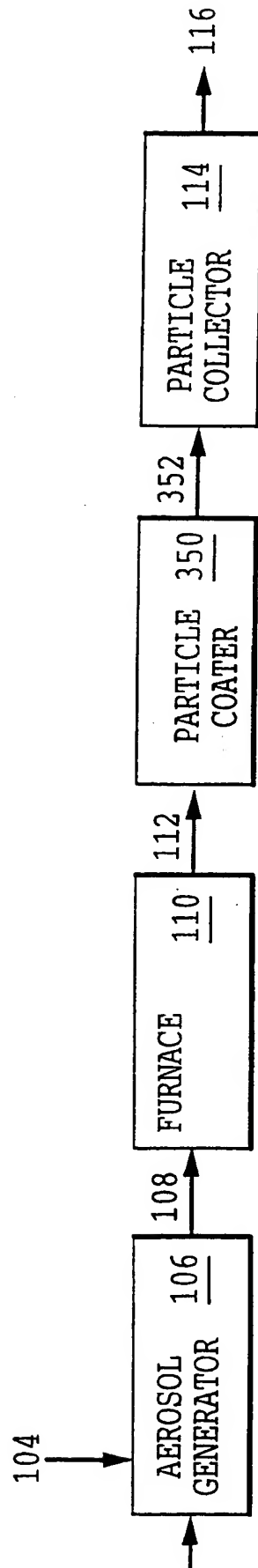


FIG.33

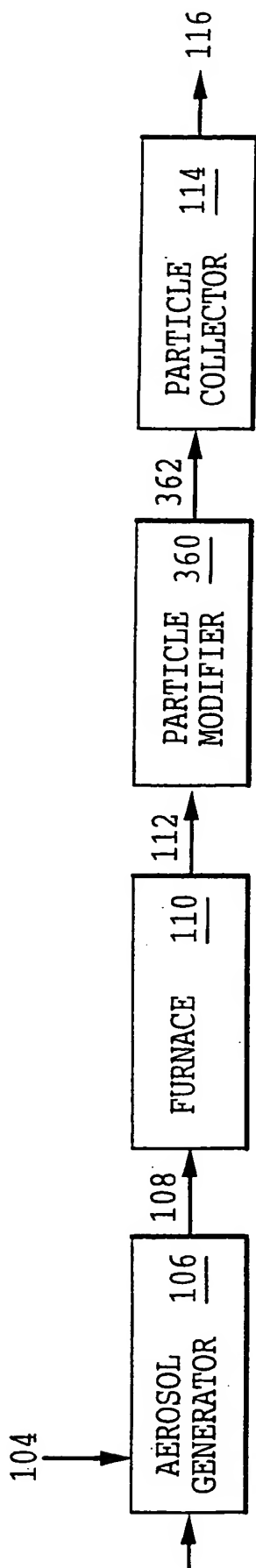


FIG.34

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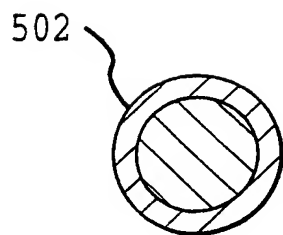


FIG. 35A

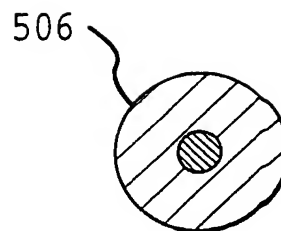


FIG. 35B

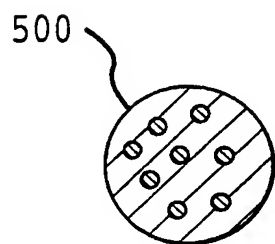


FIG. 35C

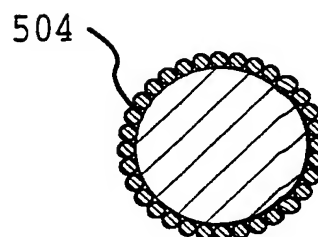


FIG. 35D

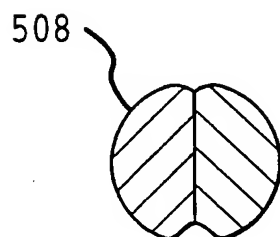


FIG. 35E

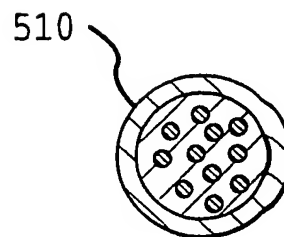


FIG. 35F

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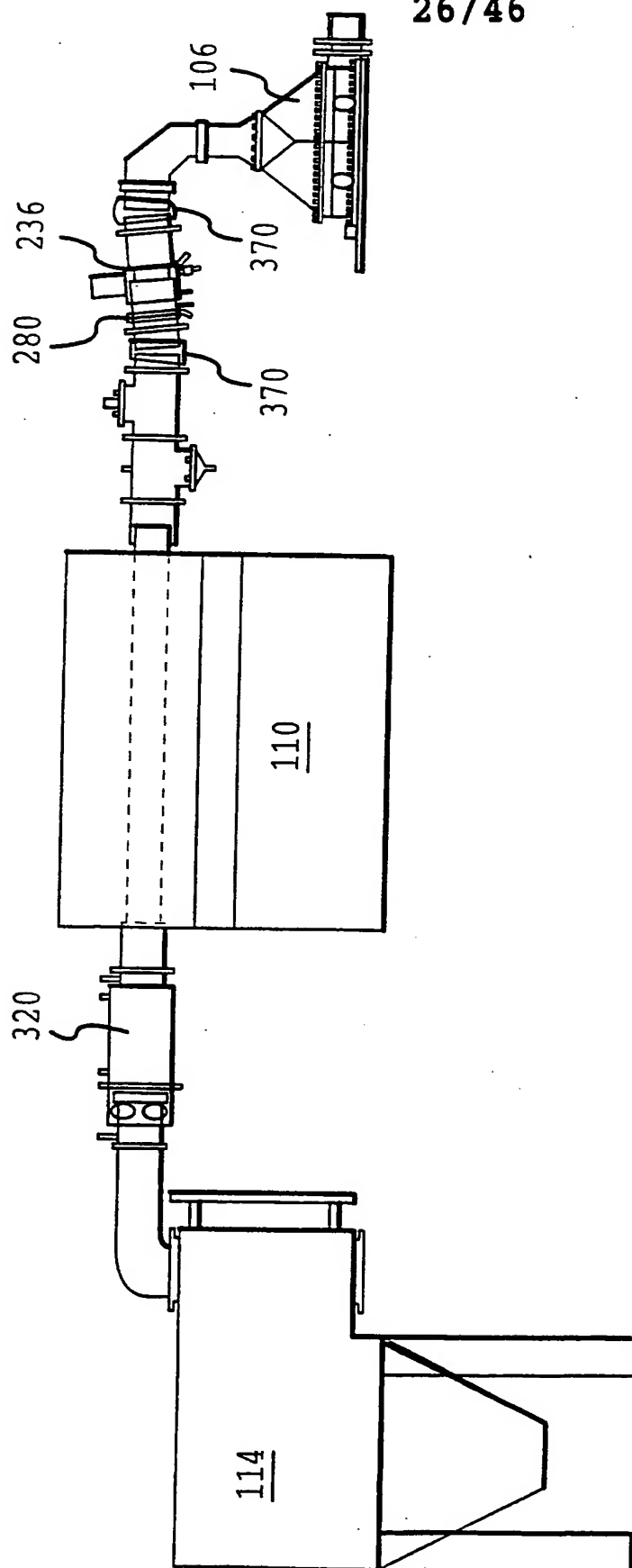


FIG.36

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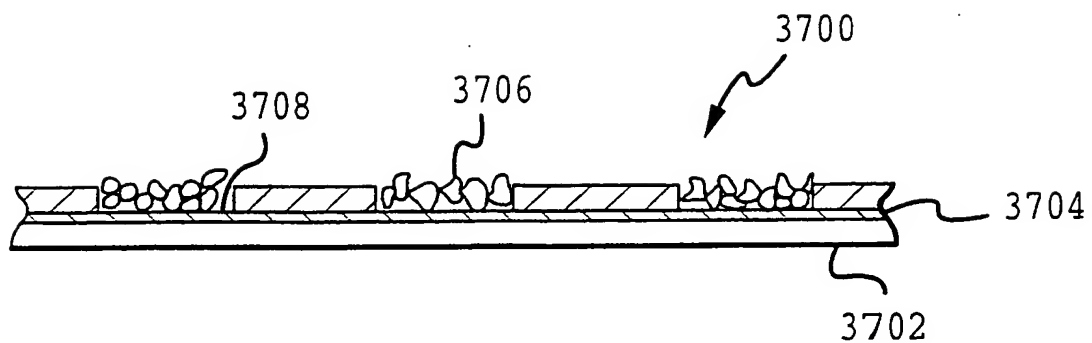


FIG.37

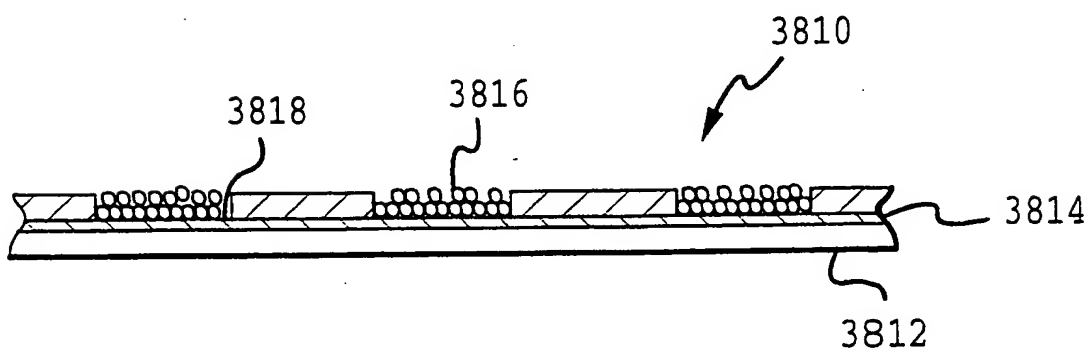


FIG.38

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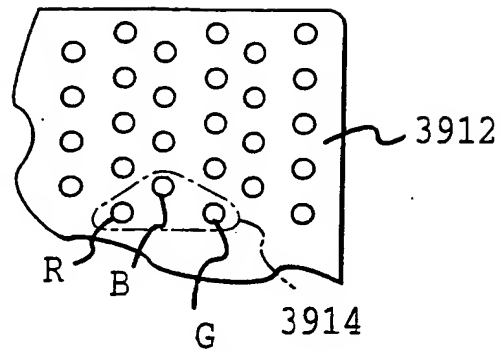


FIG. 40

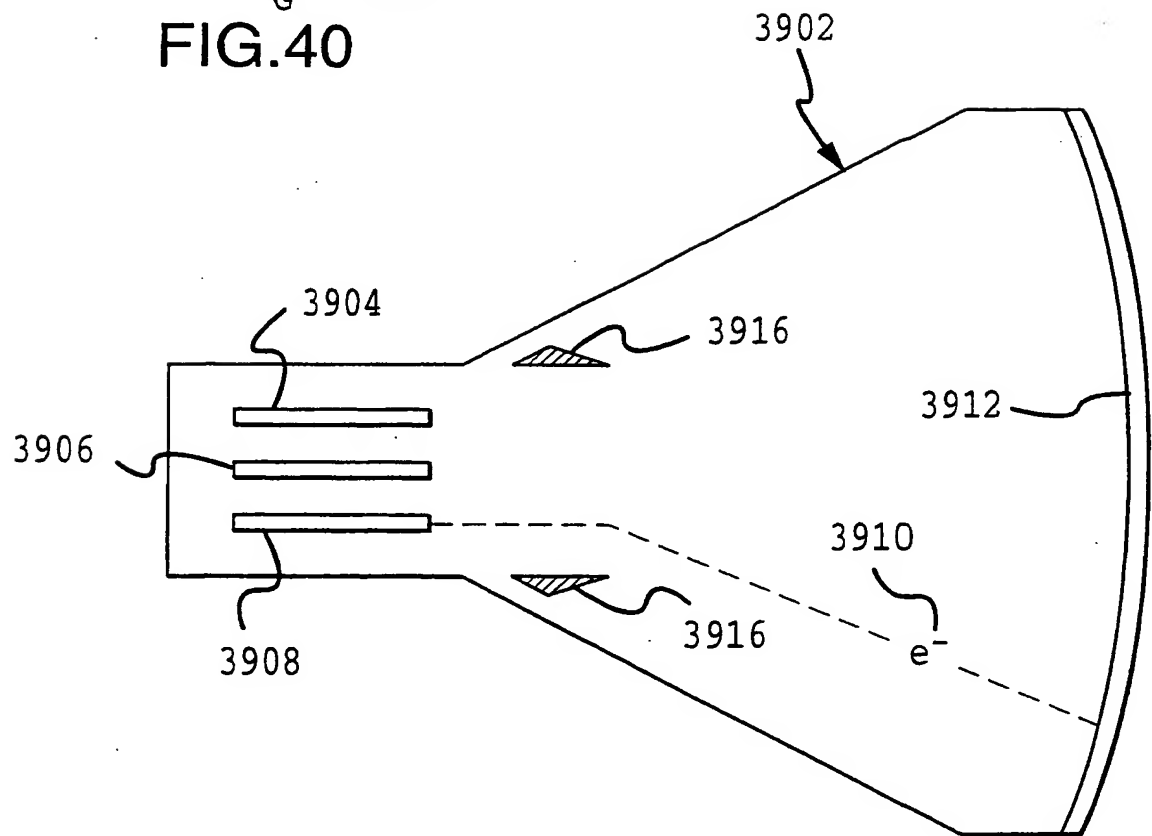


FIG. 39

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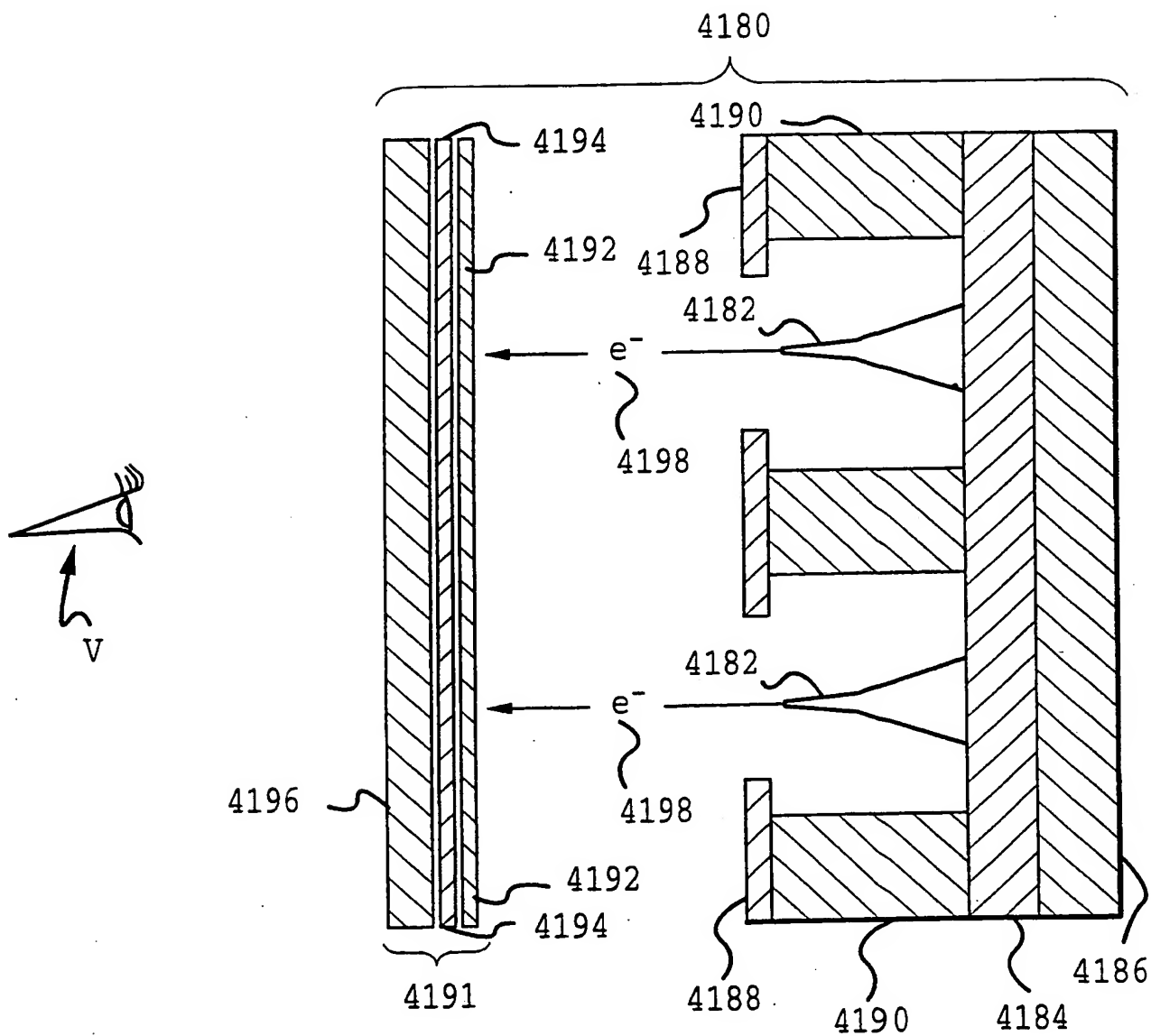


FIG.41

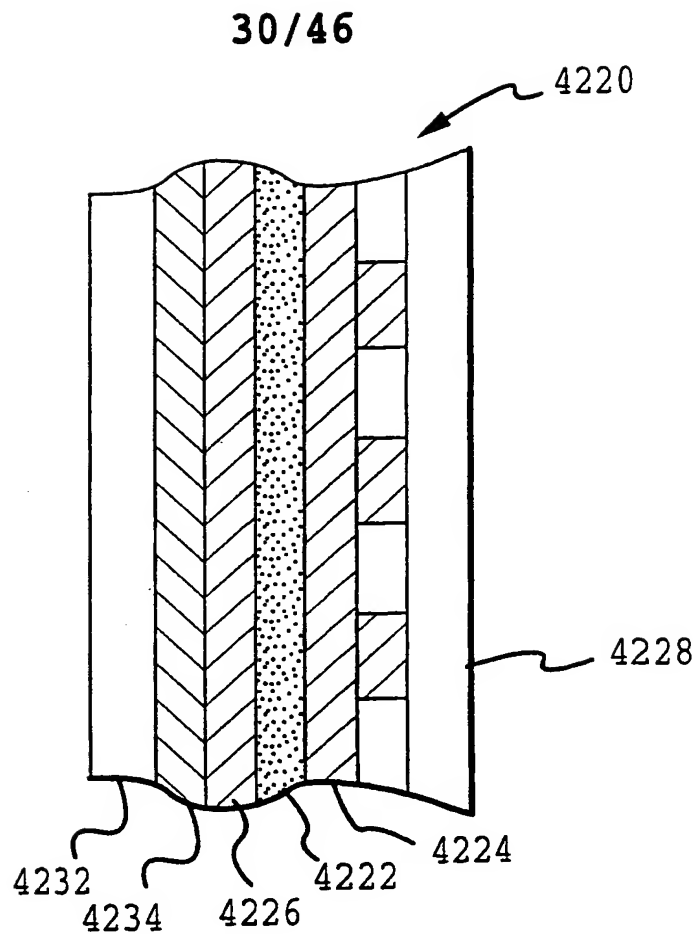


FIG. 42

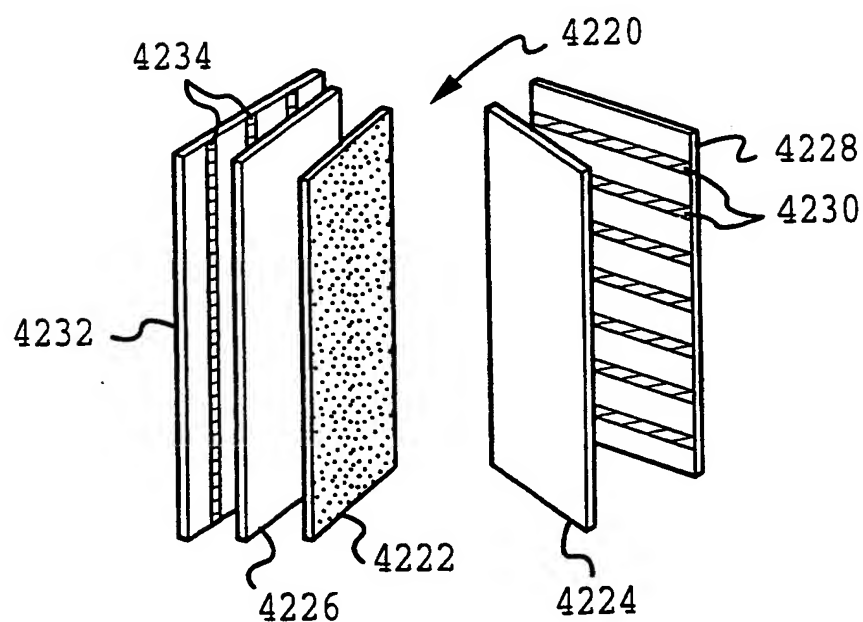


FIG. 43

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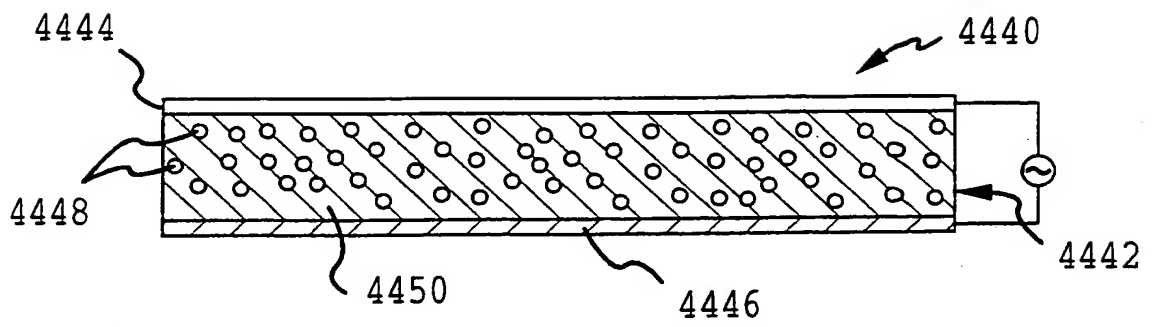


FIG.44

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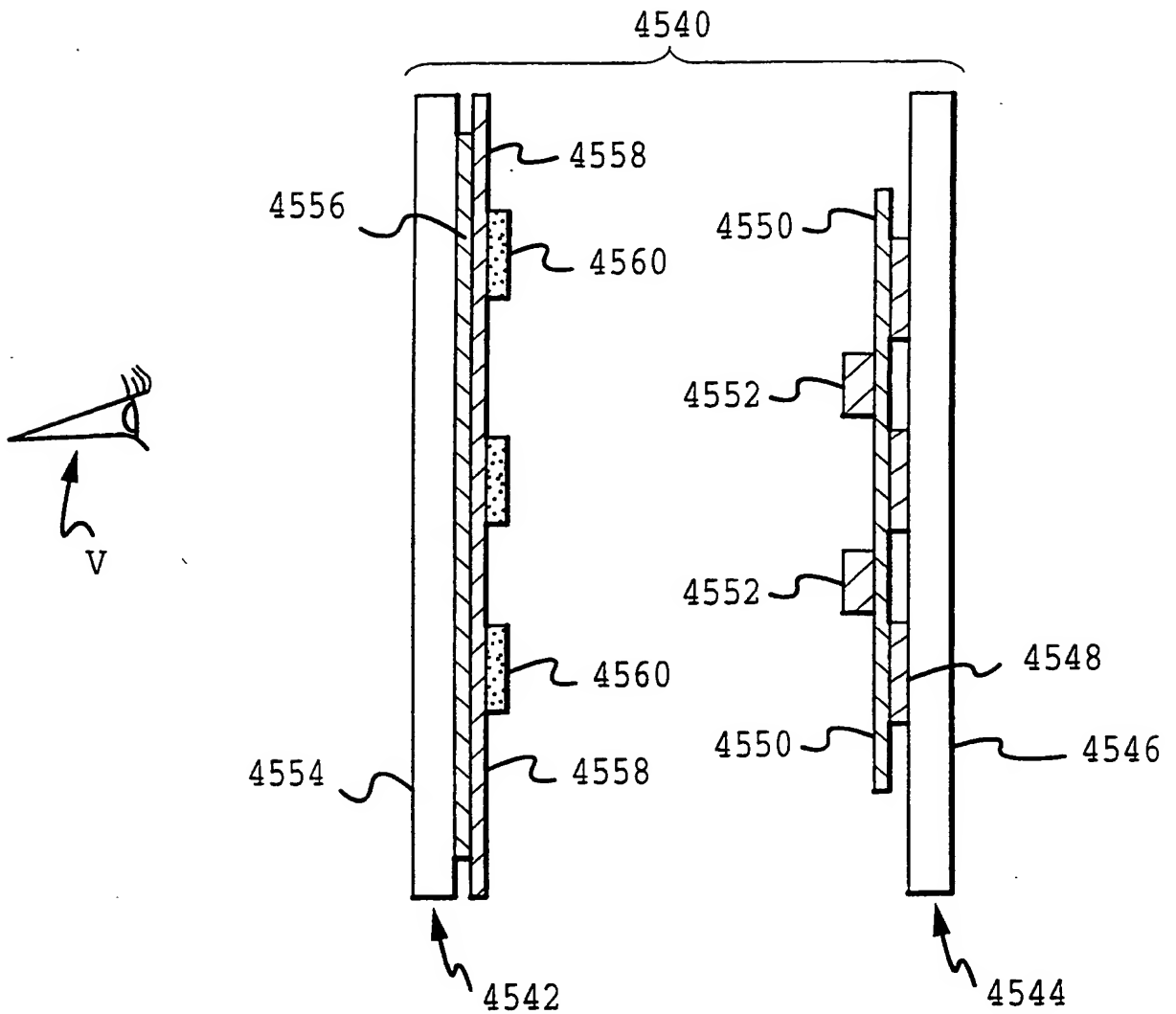


FIG.45

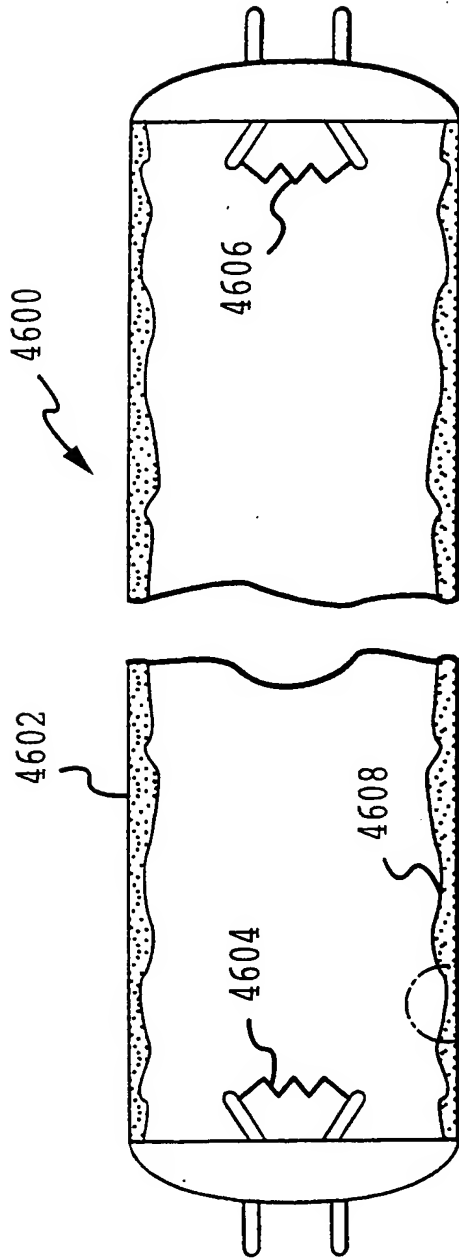


FIG. 46

FIG. 47

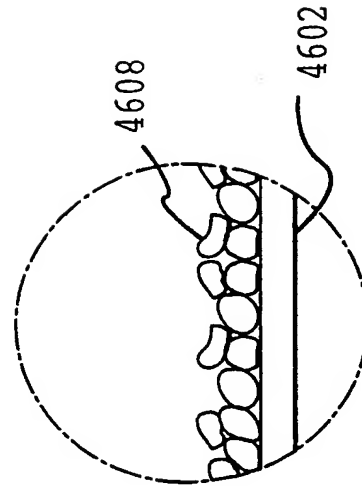


FIG. 47

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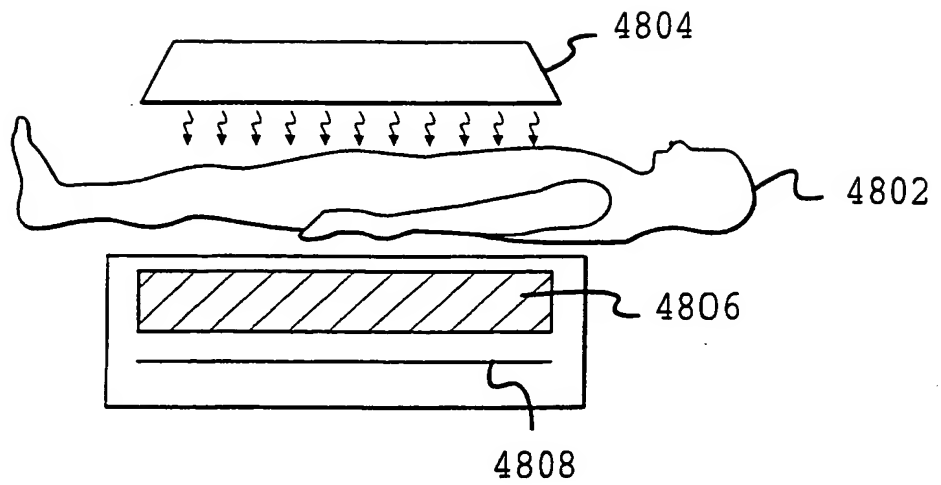


FIG. 48

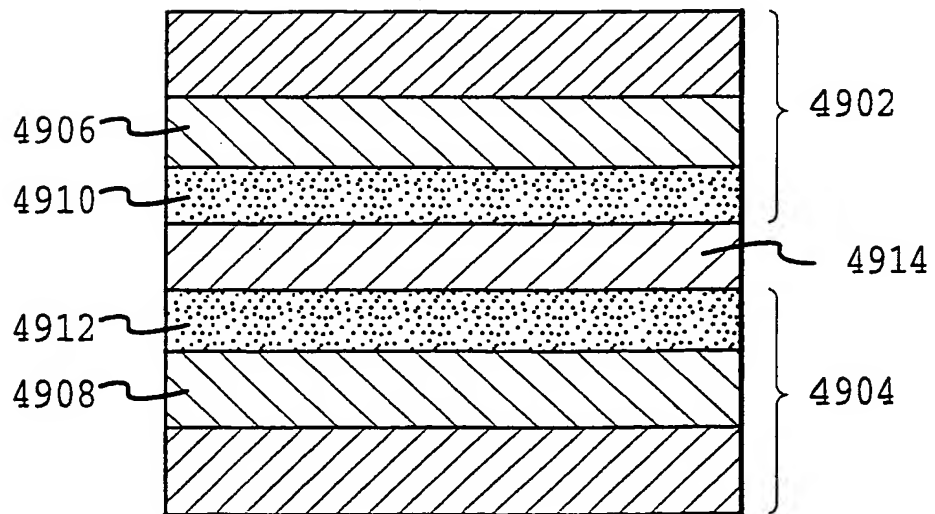


FIG. 49

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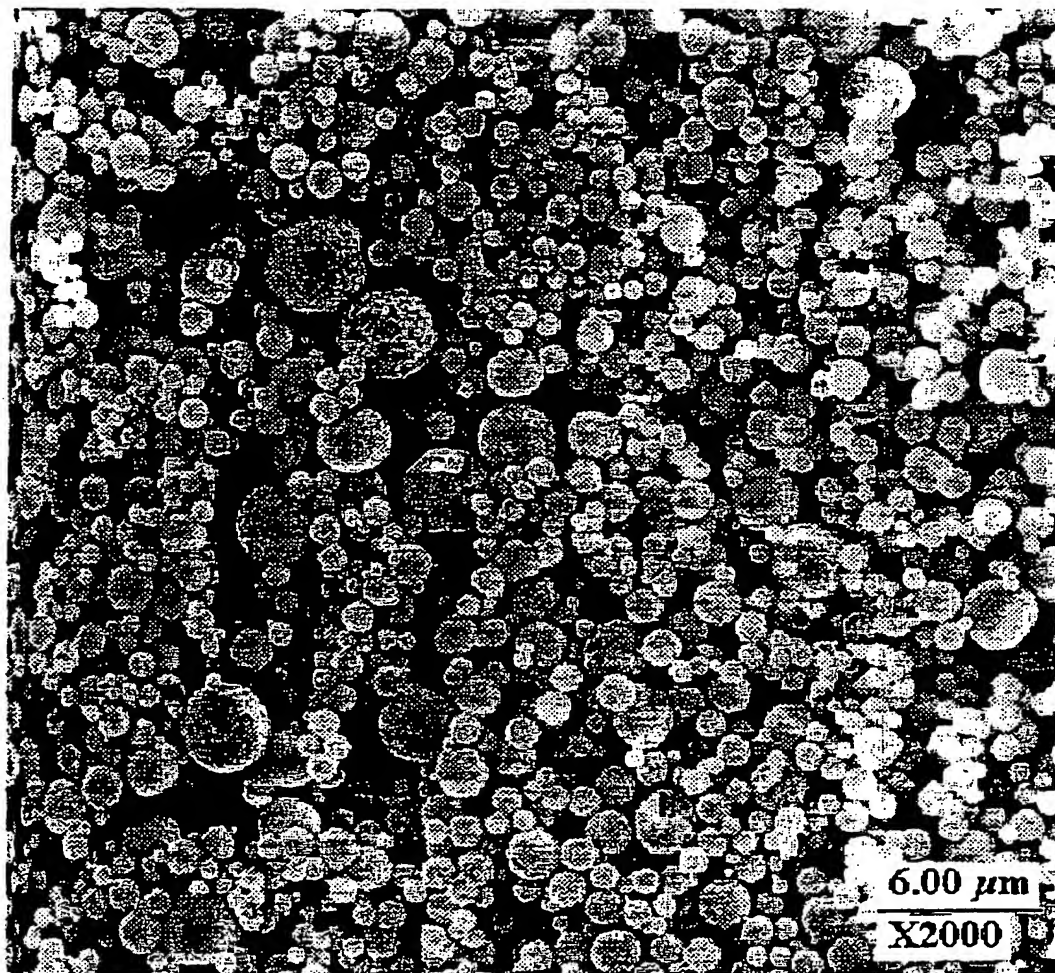


FIG.50

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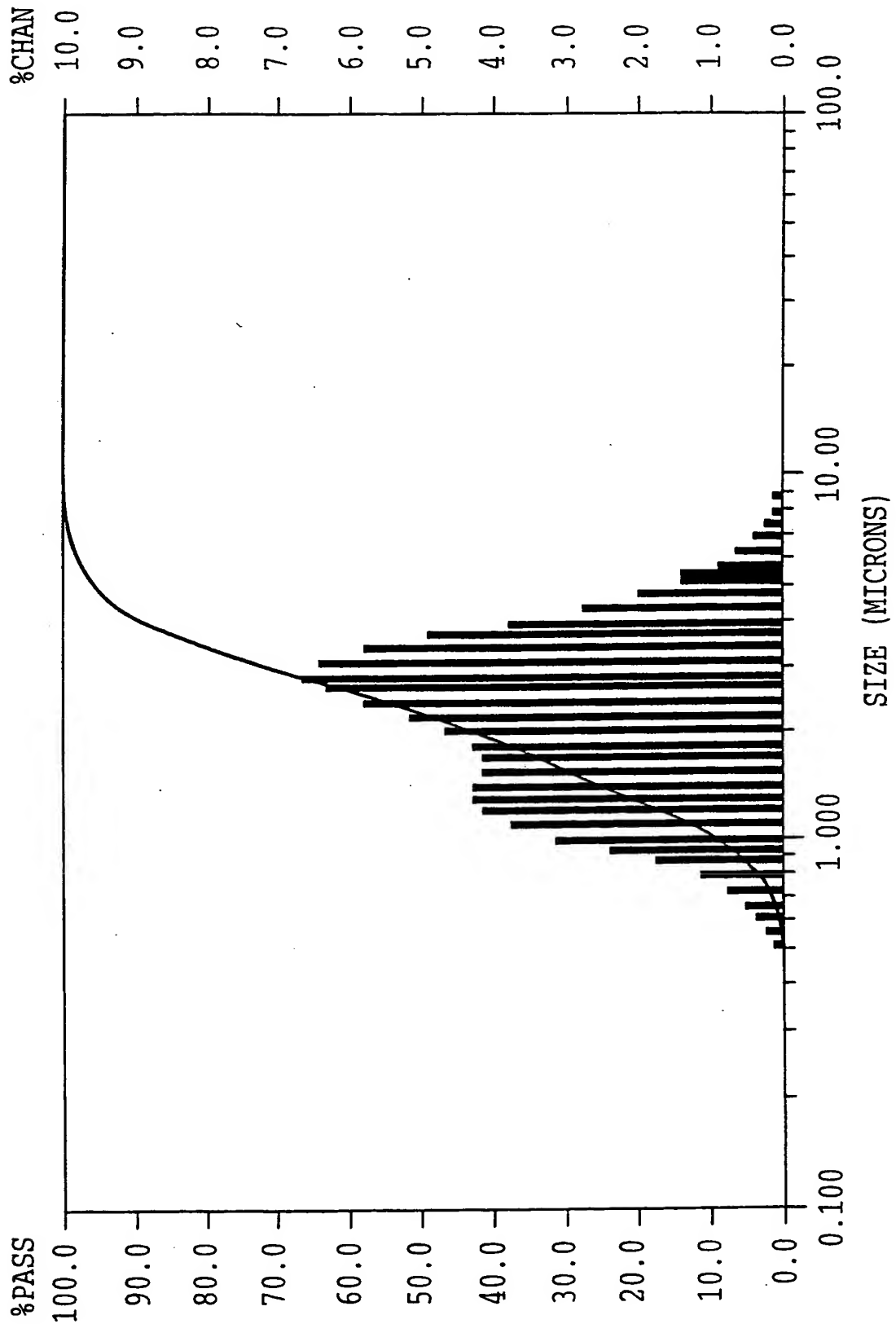


FIG.51

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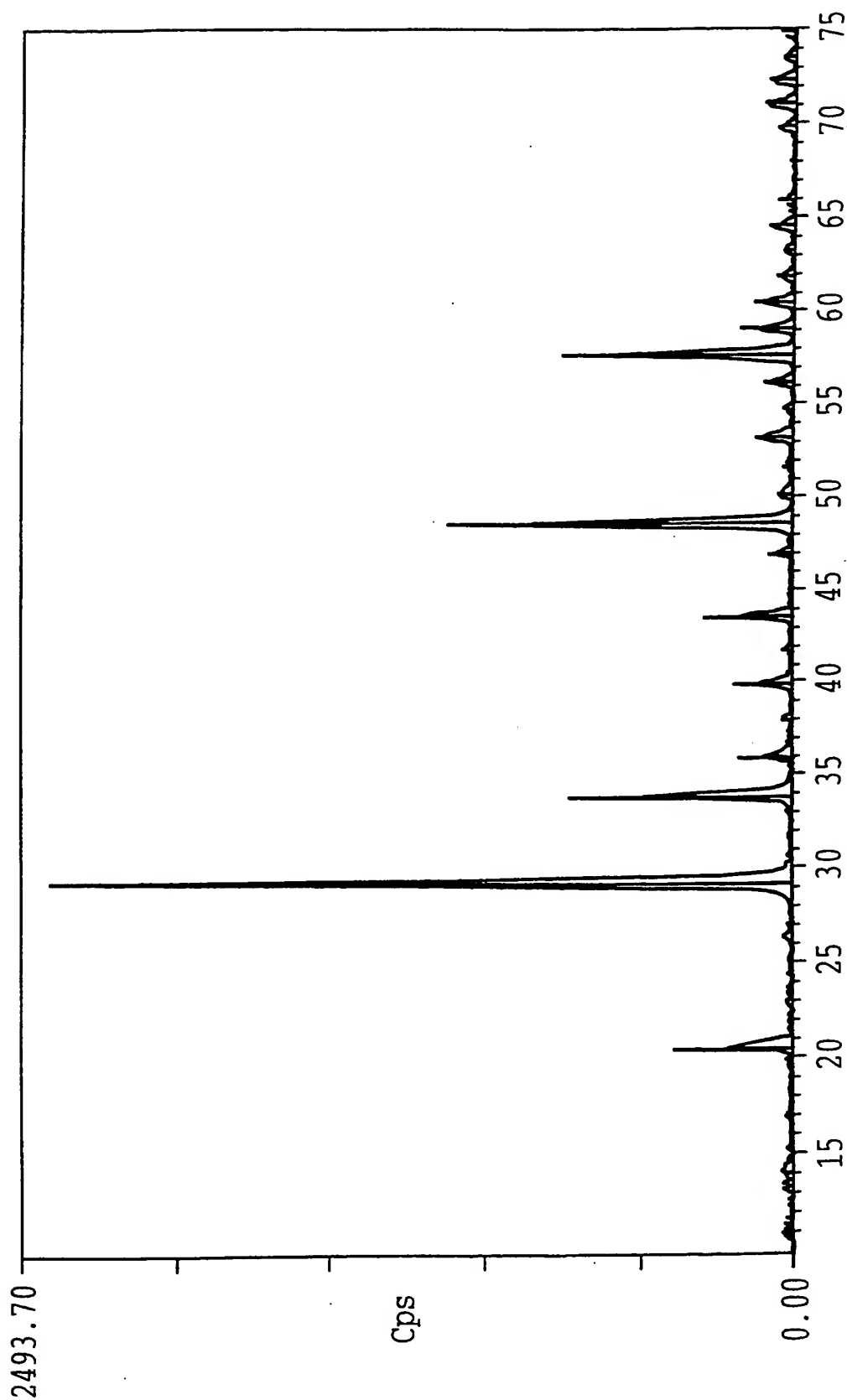


FIG.52

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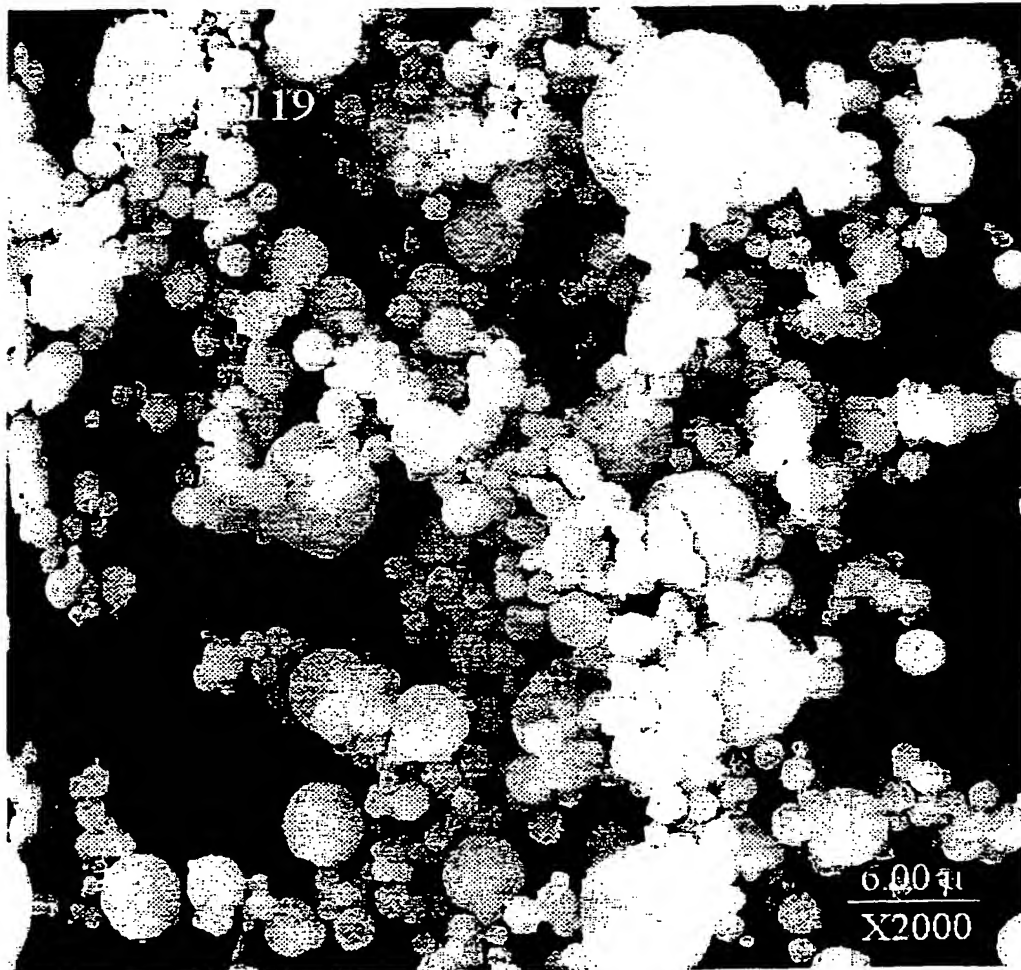


FIG.53

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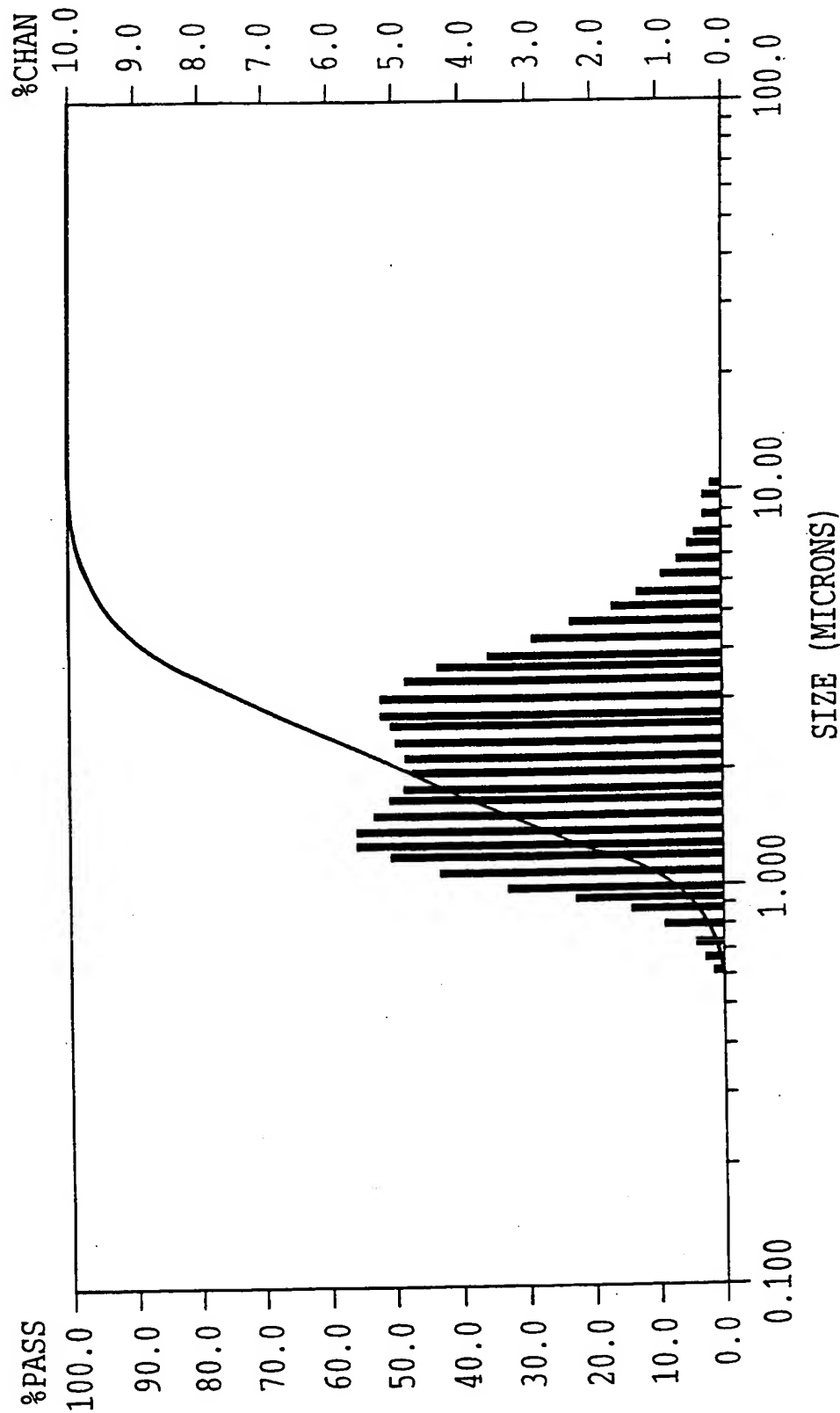


FIG.54

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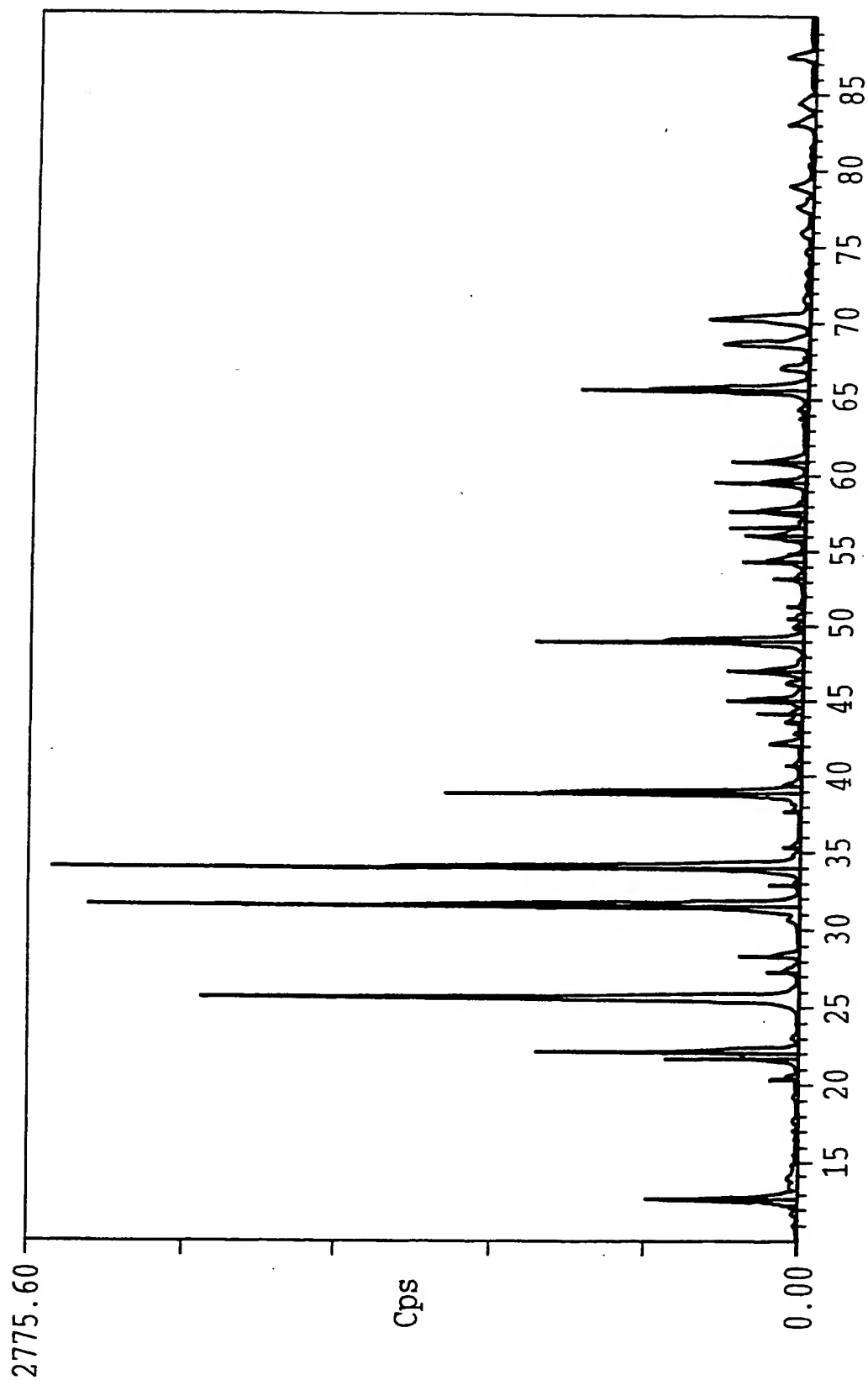


FIG.55

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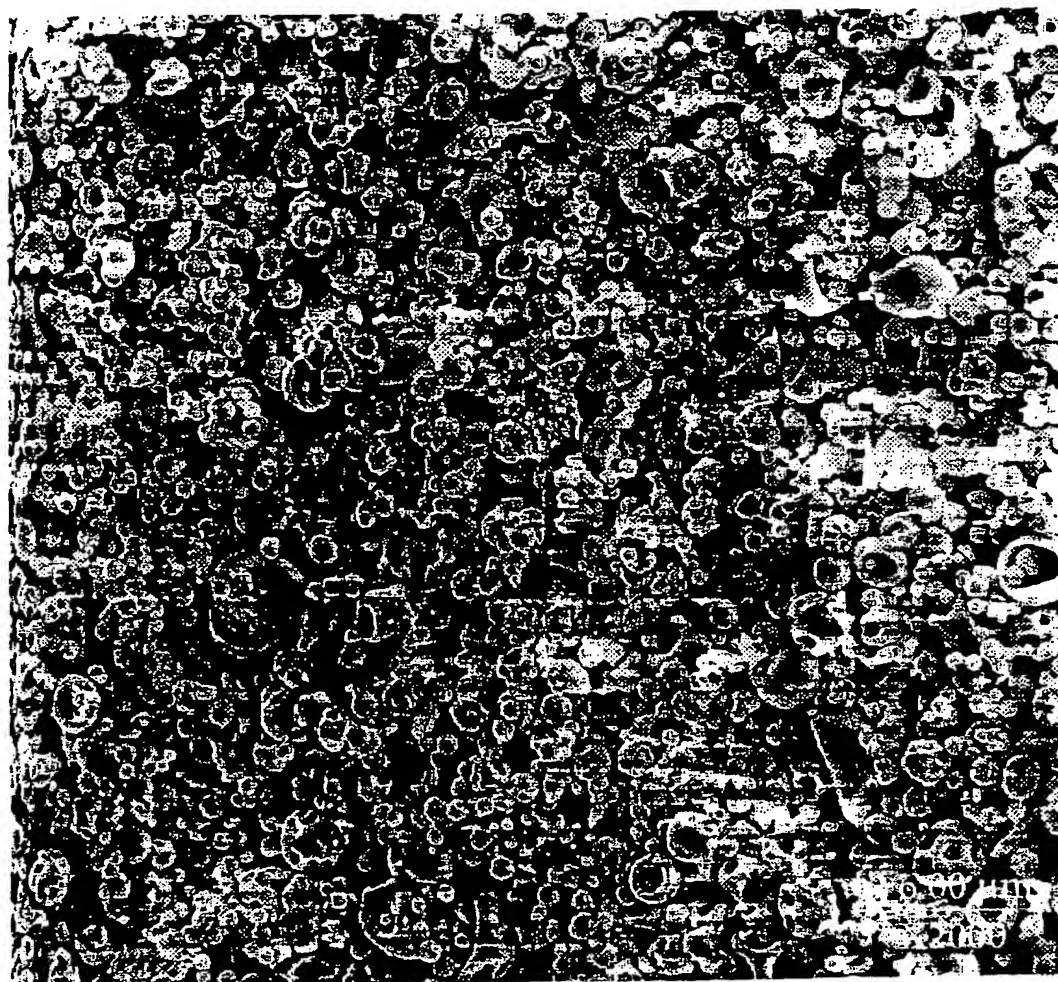


FIG.56

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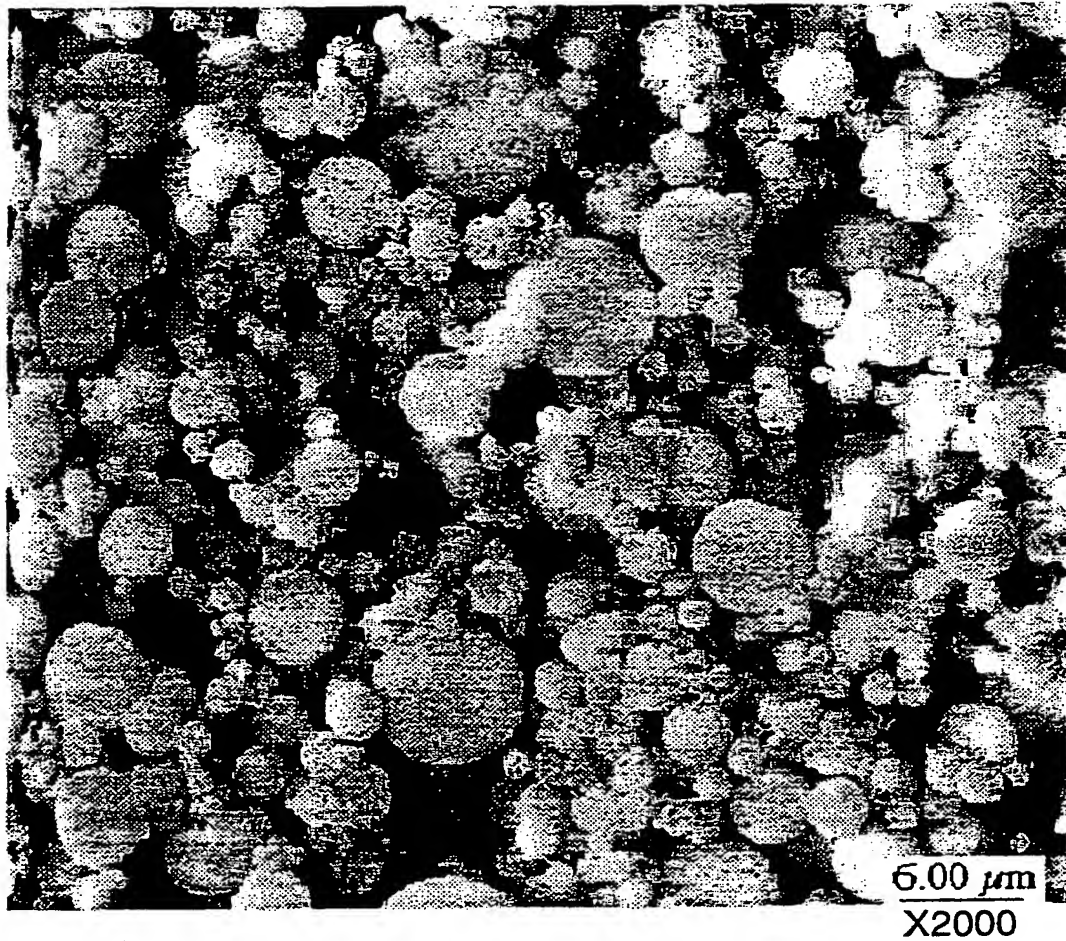


FIG.57

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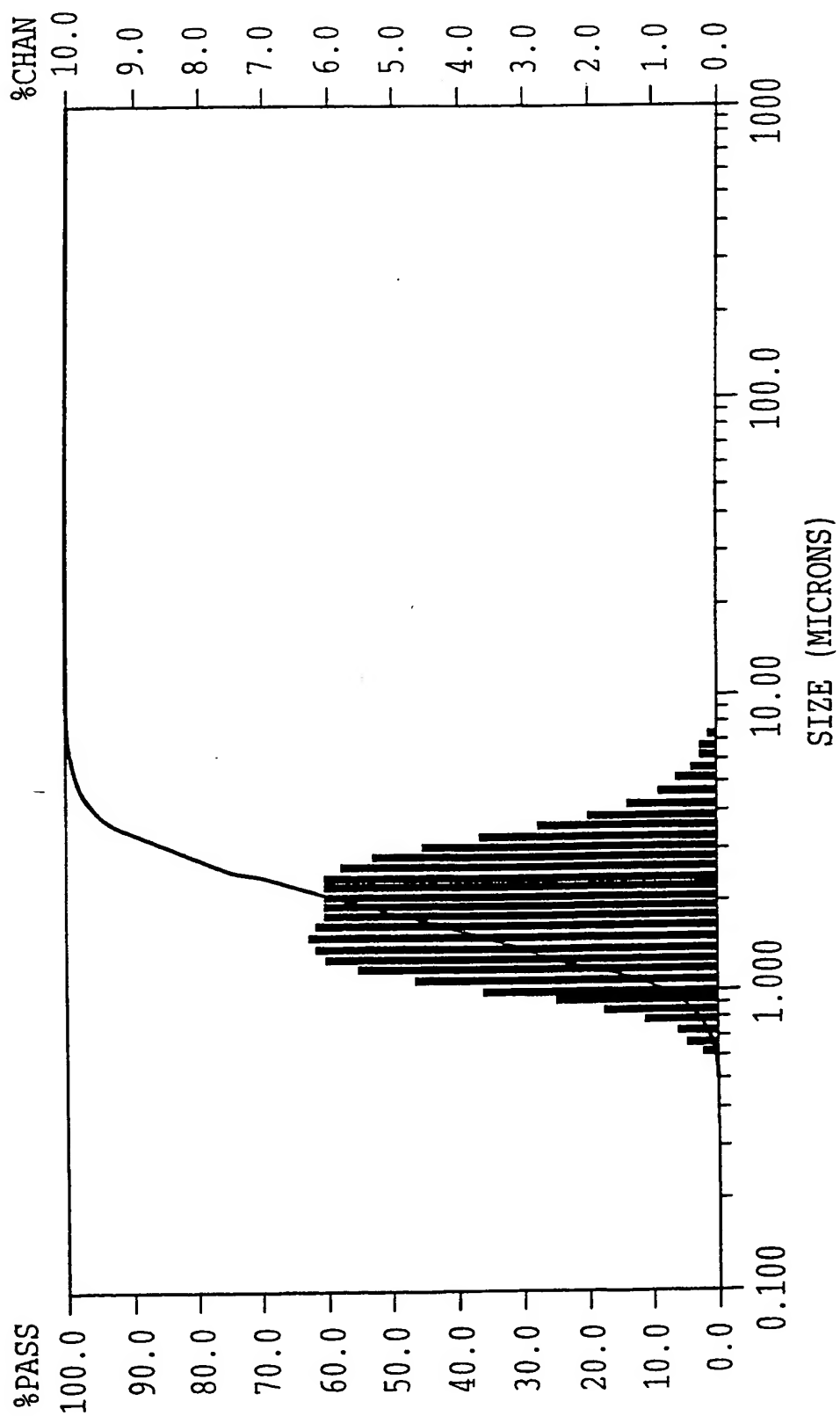


FIG.58

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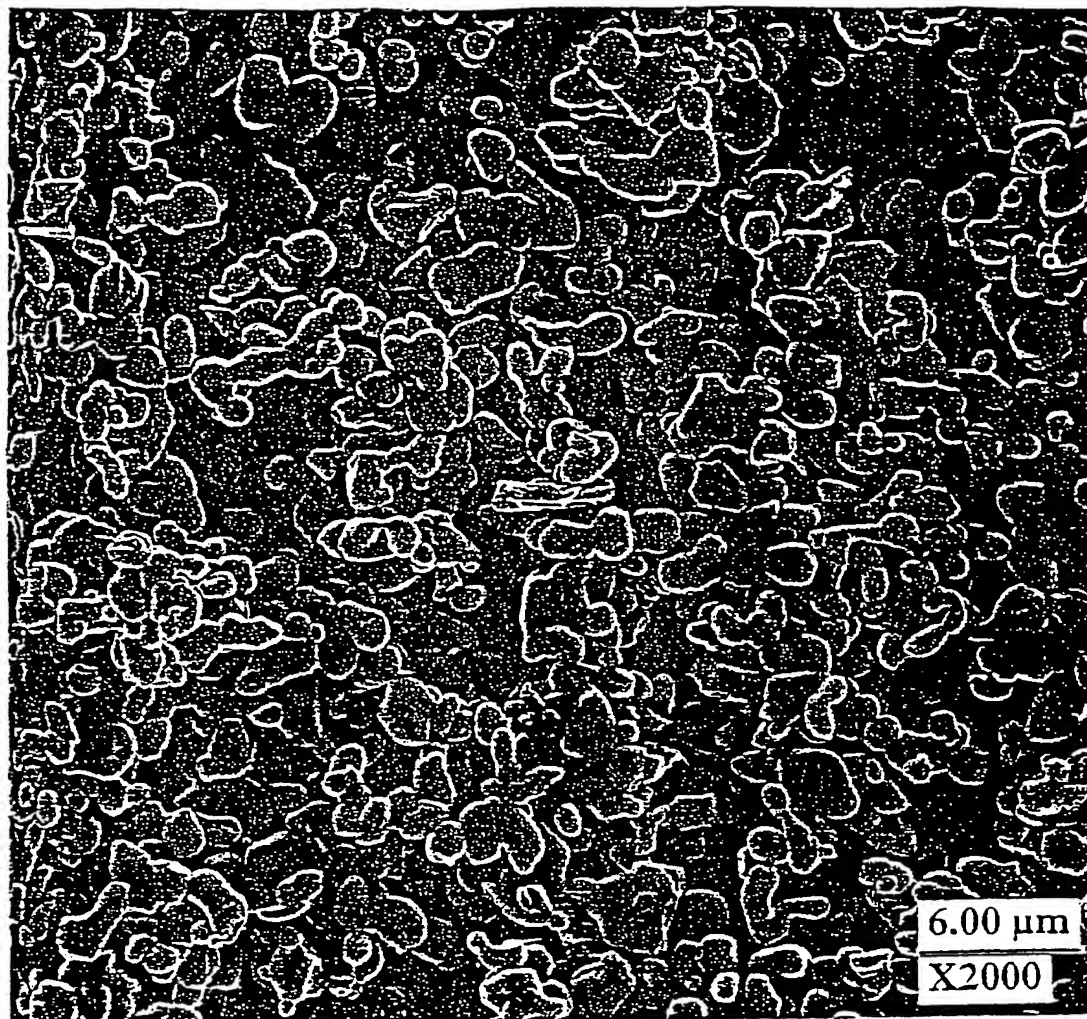


FIG.59

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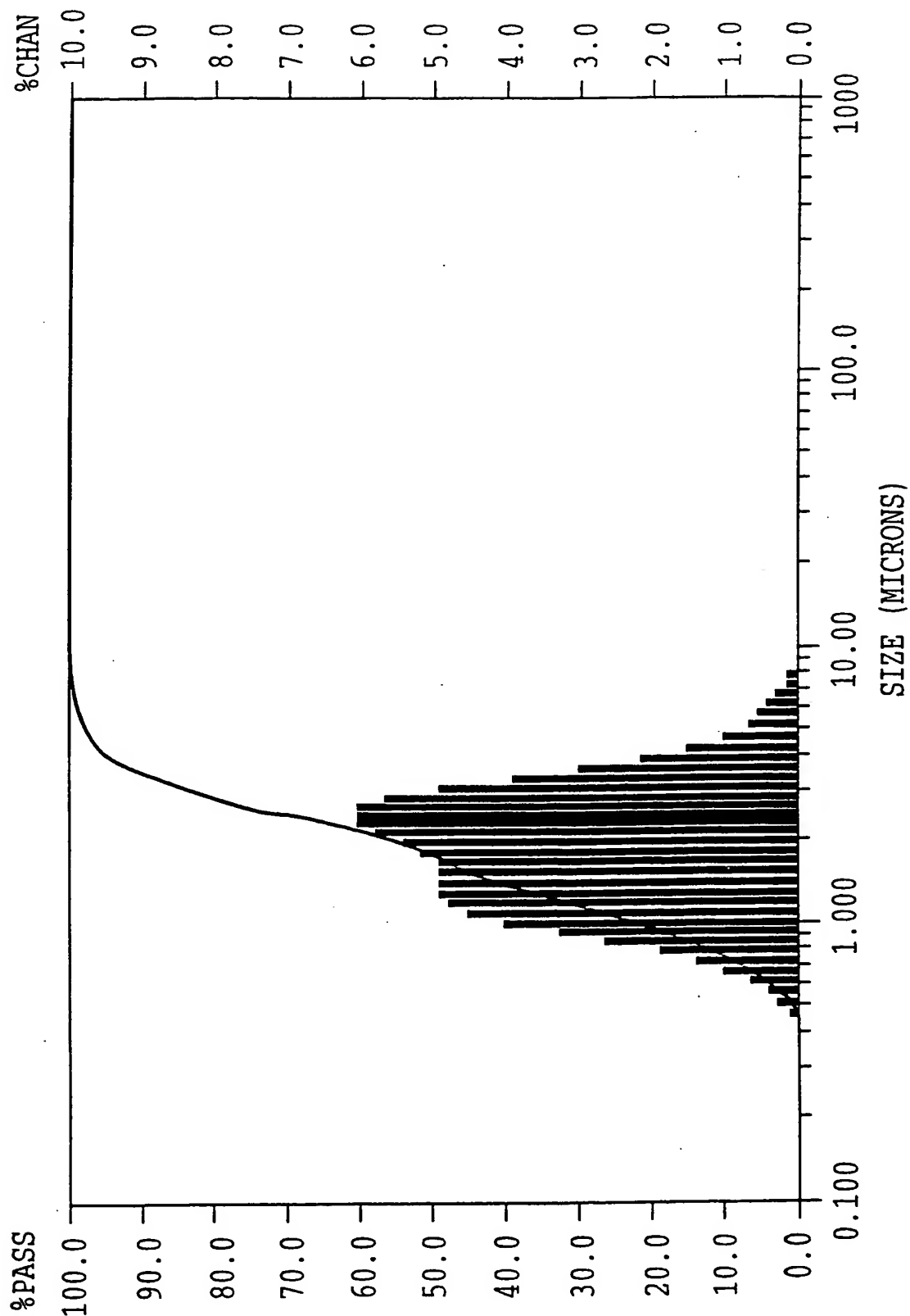


FIG.60

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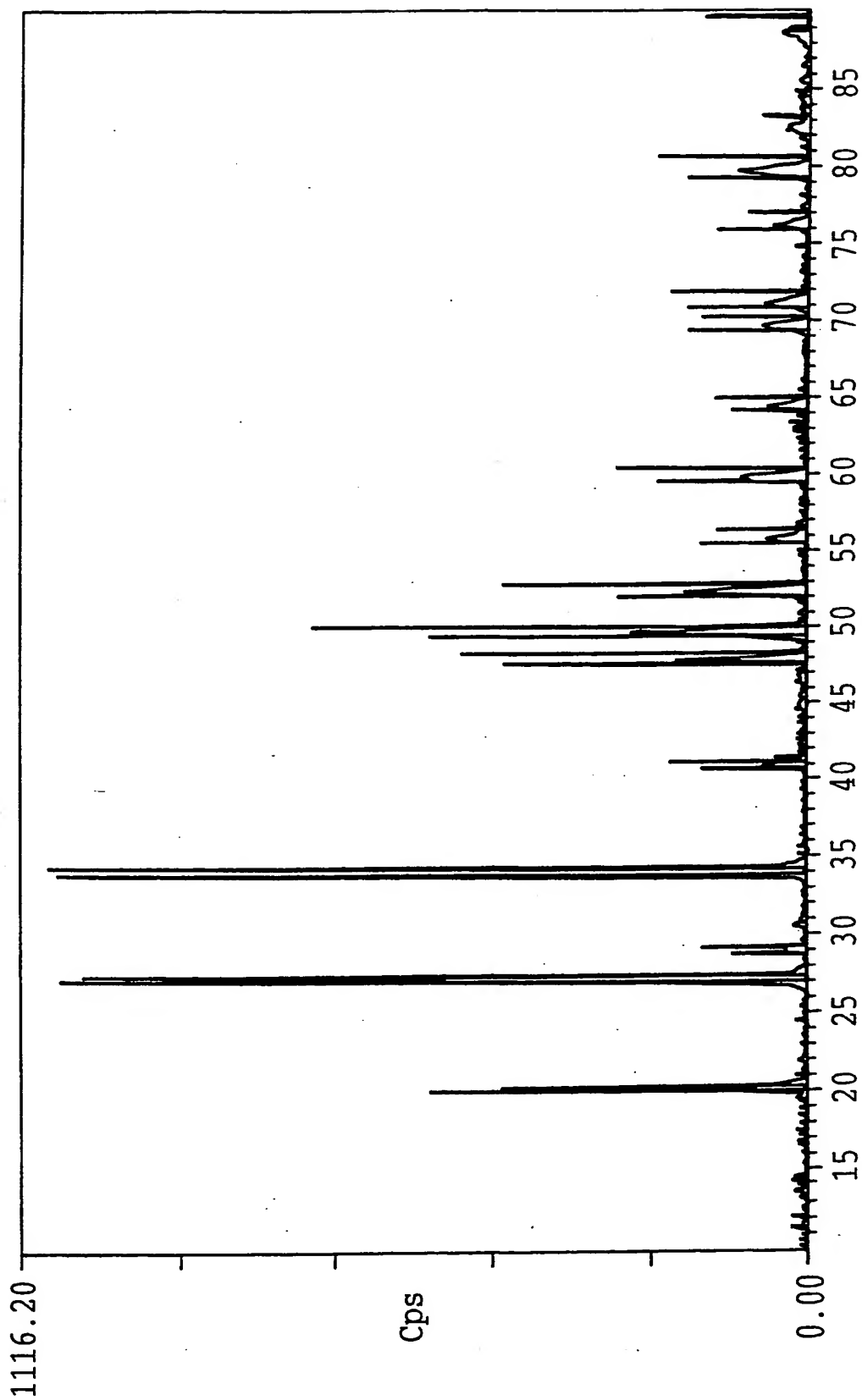


FIG.61

INTERNATIONAL SEARCH REPORT

International application No.

PCT/US99/19582

A. CLASSIFICATION OF SUBJECT MATTER

IPC(6) : Please See Extra Sheet.

US CL : Please See Extra Sheet.

According to International Patent Classification (IPC) or to both national classification and IPC

B. FIELDS SEARCHED

Minimum documentation searched (classification system followed by classification symbols)

U.S. : 252/301.4R, 301.4S, 301.4F, 301.6F, 301.6S; 313/463, 467, 468, 486, 486, 503, 582, 584; 250/361R; 283/85; 106/31.32, 31.64

Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched

Electronic data base consulted during the international search (name of data base and, where practicable, search terms used)

C. DOCUMENTS CONSIDERED TO BE RELEVANT

Category*	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
Y	US 3,923,682 A (DALE et al) 2 December 1975, examples.	230-232
Y	US 5,662,831 A (CHADHA) 2 September 1997, col. 1, ine 15-col. 7-30.	1-5, 19-22, 37-43, 62- 65, 69, 83- 90, 97-104, 108- 112, 114-120, 125-129, 135, 140- 144, 147- 150, 155- 159, 161- 165, 171- 180, 187, 188, 192- 196, 200

☒ Further documents are listed in the continuation of Box C. ☐ See patent family annex.

* Special categories of cited documents:	*T* later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the invention
A document defining the general state of the art which is not considered to be of particular relevance	*X* document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to involve an inventive step when the document is taken alone
E earlier document published on or after the international filing date	*Y* document of particular relevance; the claimed invention cannot be considered to involve an inventive step when the document is combined with one or more other such documents, such combination being obvious to a person skilled in the art
L document which may throw doubts on priority claim(s) or which is cited to establish the publication date of another citation or other special reason (as specified)	*G* document member of the same patent family
O document referring to an oral disclosure, use, exhibition or other means	
P document published prior to the international filing date but later than the priority date claimed	

Date of the actual completion of the international search

08 NOVEMBER 1999

Date of mailing of the international search report

29 NOV 1999

Name and mailing address of the ISA/US
Commissioner of Patents and Trademarks
Box PCT
Washington, D.C. 20231

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Authorized officer

C. MELISSA KOSLOW

Telephone No. (703) 308-0661

DEBORAH THOMAS
PARALEGAL SPECIALIST

INTERNATIONAL SEARCH REPORT

 International application No.
 PCT/US99/19582

C (Continuation). DOCUMENTS CONSIDERED TO BE RELEVANT

Category*	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
Y	US 5,644,193 A (MA. TSUDA et al) 1 July 1997, col. 1-15.	1-5,19-22, 37-43,62- 65,69,83- 90,97-104, 108-112, 114-120, 125-129, 135,140- 144,147- 150,155- 159,161- 165,171- 180,187, 188,192- 196,200
Y	US 5,037,577 A (YAMANOI et al) 6 August 1991, col. 1-10.	1-3,13,14, 19,20,62- 64,73,74,83- 89,97-103,108- 111, 114-118, 125-129, 139-143
Y	US 4,801,398 A(OHNO et al) 31 January 1989, col. 1-6.	1-3,13,14, 19,20,62- 64,73,74, 83-89,94,97- 103,108- 111,114- 118,125- 129,139- 143
Y	US 5,132,045 A (OSAKA et al) 21 July 1992, col. 1-col. 28.	62-84
Y	US 4,451,521 A (KAULE et al) 29 May 1984, col. 1-col. 10.	201-215
Y	US 4,883,970 A (PAGE) 28 November 1989, col. 2-col. 8.	53-55,60 -64,67- 68,83-84, 216- 218223-229

Form PCT/ISA/210 (continuation of second sheet)(July 1992)★

INTERNATIONAL SEARCH REPORT

International application No.

PCT/US99/19582

C (Continuation). DOCUMENTS CONSIDERED TO BE RELEVANT

Category*	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
Y	US 3,773,540 A (SHIDLOVSKY) 20 November 1973, col. 1-col. 6.	1-3,19,20, 85-89,98- 101,109- 111,114- 117,129, 140-145, 147
Y	US 5,413,736A (NISHISU et al) 9 May 1995, col. 1-col. 4.	1-5,19-22, 37-43,62- 69,83-90,97- 104,108- 112,114- 120-,125- 129,135, 140-144, 147-149, 155-165, 171-180, 187,188, 192-196, 200-203, 206-212, 215
Y ^P	US 5,811,924 A (OKUMURA et al) 22 September 1998, col. 1-28.	1-10,13-28,37- 43,46- 52,62-71,73- 78,80-94, 96-106,108- 123,125- 135,138- 152,154- 165,167- 188,192- 196,198- 215

Form PCT/ISA/210 (continuation of second sheet)(July 1992)★

INTERNATIONAL SEARCH REPORT

International application No.

PCT/US99/19582

A. CLASSIFICATION OF SUBJECT MATTER:

IPC (6):

C09K 11/00, 11/54, 11/56, 11/59, 11/64, 11/63, 11/78, 11/79, 11/80, 11/84; G01T 1/20; B24D 15/00, 15/10, C09D 11/00, 5/22; H01J 29/24, 1/63

A. CLASSIFICATION OF SUBJECT MATTER:

US CL :

252/301.4R, 301.4S, 301.4F, 301.6F, 301.6S; 313/463, 467, 468, 486, 486, 503, 582, .584; 250/361R; 283/85; 106/31.32, 31.64